

Damage-enhanced order reduction models for 2D woven composites based on data-driven multiscale mechanics

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Context

2D woven composites

- **periodic heterogeneous multiscale** materials that can be described by **RVEs**
 - mesoscale ($\sim 10^{-3}$ m) RVE: matrix + warp and weft yarns
 - microscale ($\sim 10^{-6}$ m) RVE of yarns: UD fibres + matrix
- **damage-enhanced nonlinear** behaviours for constituents
 - fibres: transversely isotropic linear elasticity
 - matrix: isotropic J_2 elasto-plasticity
- numerical modelling and simulation
 - **complex** geometrical description and material properties assignment
 - **high** computational cost (time and memory usage)

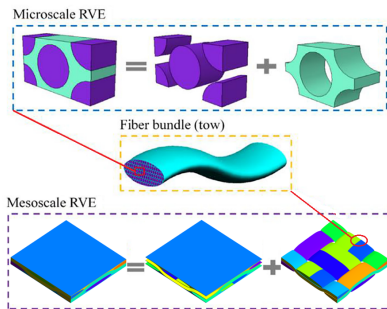


Figure: Multiscale description of a 2D woven composite [Wang et al. 2017]

Context

Mean-Field Homogenisation (MFH) for yarns

- **semi-analytical** solution for the extended Eshelby problem with multiple inclusions

$$\bar{\boldsymbol{\varepsilon}} = v_1 \boldsymbol{\varepsilon}_1 + v_0 \boldsymbol{\varepsilon}_0, \quad \bar{\boldsymbol{\sigma}} = v_1 \boldsymbol{\sigma}_1 + v_0 \boldsymbol{\sigma}_0$$

- Linear Comparison Composite (LCC)-based nonlinear framework

$$\Delta \boldsymbol{\varepsilon}_1 = \mathbb{B}^\varepsilon (\mathbf{I}, \mathbb{C}_1^{\text{LCC}}, \mathbb{C}_0^{\text{LCC}}) : \Delta \boldsymbol{\varepsilon}_0$$

- Mori-Tanaka (MT) model

$$\mathbb{B}^\varepsilon (\mathbf{I}, \mathbb{C}_1^{\text{LCC}}, \mathbb{C}_0^{\text{LCC}}) = \left\{ \mathbf{I} + \mathbb{S} (\mathbf{I}, \mathbb{C}_0^{\text{LCC}}) : \left[(\mathbb{C}_0^{\text{LCC}})^{-1} : \mathbb{C}_1^{\text{LCC}} - \mathbf{I} \right] \right\}^{-1}$$

Context

Full-order FE modelling and damage-enhanced MFH-based **nonlinear** Direct Numerical Simulation (DNS) ([Wu, Maillard, and Noels 2021] and [Wu et al. 2021])

- computational cost: 63189 elements; ~ 3 days
- still too high for applications of industrial complexity

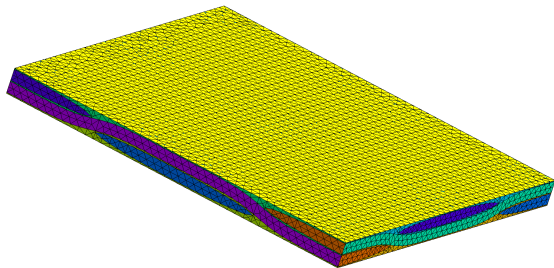


Figure: FE model of a 2D woven RVE at mesoscale

Problematic

Propose a **reduced-order** model (ROM) for 2D woven composites by embedding

- **micromechanics-based** homogenization methods
- **data-driven** model reduction techniques

to conduct accurately and efficiently **damaged-enhanced nonlinear multiscale** analysis

Outline

- 1 Introduction
- 2 Reduced-order models
- 3 Offline training and online evaluation
- 4 Application
- 5 Conclusions and outlooks

Outline

- 1 Introduction
- 2 **Reduced-order models**
 - Deep Material Network (DMN)
 - VM scheme
 - LVM scheme
 - VLM scheme
- 3 Offline training and online evaluation
- 4 Application
- 5 Conclusions and outlooks

Deep Material Network (DMN)

Deep Material Network (DMN) [Liu and Wu 2019]: a ROM for RVE

- mechanistic building blocks-based tree structure: homogenization + rotation
- efficient offline training in linear elastic regime: geometrical parameters fitting
- accurate online extrapolation in arbitrary unknown nonlinear inelastic regime: mechanical response prediction

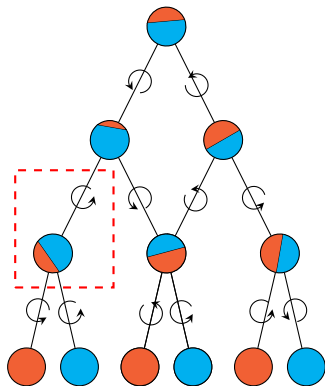
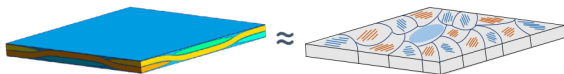


Figure: Illustration of a 3-layer DMN for a biphasic material

VM scheme

DMN-liked VM scheme [Wu, Adam, and Noels 2021]: 2D woven RVE \approx Short Fibre Reinforced Matrix (SFRM) pseudo-grains aggregate



- step 1: matrix + fibres \Rightarrow MFH \Rightarrow SFRM
- step 2: SFRMs \Rightarrow Voigt's rule of mixtures \Rightarrow 2D woven RVE
- parameters: $\chi^{\text{VM}} = \{v_i, \theta_i, \alpha_i \mid i = 1, \dots, N_s; \sum_{i=1}^{N_s} v_i = 1.0 - v_0\}$

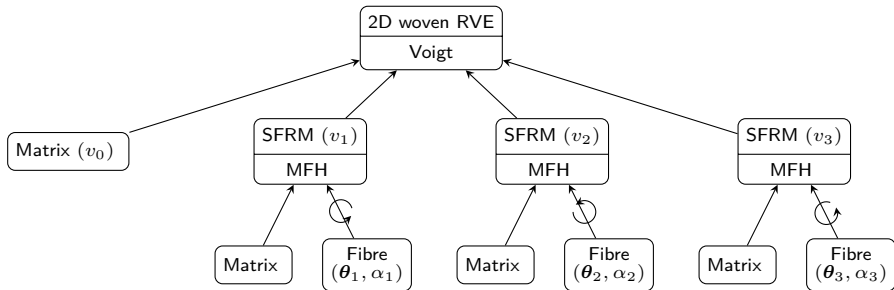
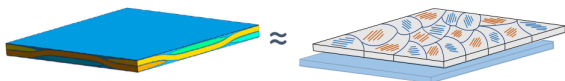


Figure: Illustration of the VM scheme ($N_s = 3$)

LVM scheme

DMN-liked LVM scheme [Wu, Adam, and Noels 2021]: 2D woven RVE \approx 2-ply laminate



- step 1: matrix + fibres \Rightarrow MFH \Rightarrow SFRM
- step 2: SFRMs \Rightarrow Voigt's rule of mixtures \Rightarrow SFRMs aggregate
- step 3: matrix ply + aggregate ply \Rightarrow Laminate theory \Rightarrow 2D woven RVE
- parameters: $\chi^{\text{LVM}} = \{v_i, \theta_i, \alpha_i \mid i = 1, \dots, N_s; \sum_{i=1}^{N_s} v_i = 1.0\}$

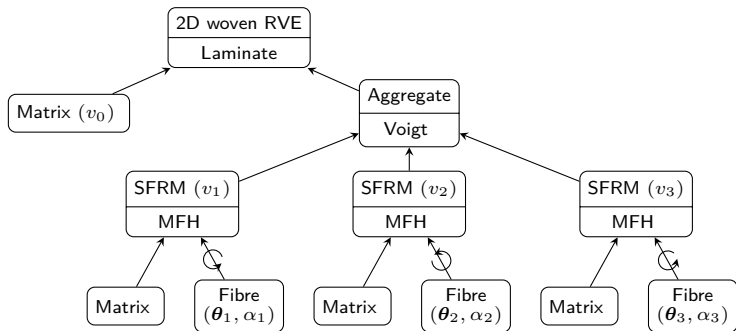
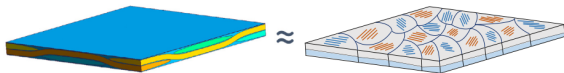


Figure: Illustration of the LVM scheme ($N_s = 3$)

VLM scheme

DMN-liked VLM scheme [Wu, Adam, and Noels 2021]: 2D woven RVE \approx 2-ply laminate pseudo-grains aggregate



- step 1: matrix + fibres \Rightarrow MFH \Rightarrow SFRM
- step 2: matrix ply + SFRM ply \Rightarrow Laminate theory \Rightarrow 2-ply laminate
- step 3: 2-ply laminate aggregate \Rightarrow Voigt's rule of mixtures \Rightarrow 2D woven RVE
- parameters: $\chi^{\text{VLM}} = \left\{ v_i^g, \theta_i^g, v_i^m, \theta_i^f, \alpha_i \mid i = 1, \dots, N_s; \sum_{i=1}^{N_s} v_i^g = 1.0; \sum_{i=1}^{N_s} v_i^g v_i^m = v_0 \right\}$

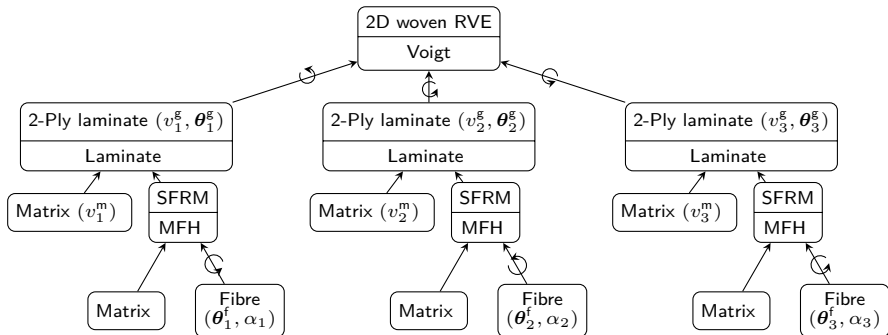


Figure: Illustration of the VLM scheme ($N_s = 3$)

Outline

- 1 Introduction
- 2 Reduced-order models
- 3 Offline training and online evaluation**
- 4 Application
- 5 Conclusions and outlooks

Offline training

Identification of χ^{opt} in linear elasticity by solving

$$\chi^{\text{opt}} = \arg \min_{\chi} L(\hat{\mathbb{C}}(\gamma), \bar{\mathbb{C}}(\chi|\gamma_s))$$

with

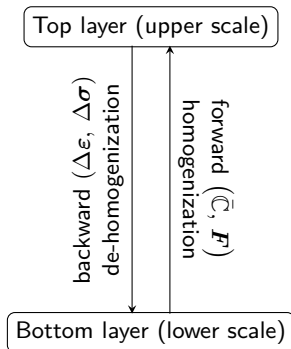
- $\hat{\mathbb{C}}(\gamma)$: homogenized **elasticity** tensor given by computational homogenization on a **full-order** model parametrized by $\gamma = (E_0, \nu_0, E_1^T, E_1^L, \nu_1^{LT}, \nu_1^{TT}, G_1^{LT})$
- $\bar{\mathbb{C}}(\chi|\gamma_s)$: homogenized **elasticity** tensor given by **VM**, **LVM** or **VLM** scheme parametrized by χ^{VM} , χ^{LVM} or χ^{VLM} for given γ
- loss function: ($G(\chi)$ is a volume fraction consistency function)

$$L(\hat{\mathbb{C}}, \bar{\mathbb{C}}(\chi)) = \frac{1}{n} \sum_{s=1}^n \frac{\|\hat{\mathbb{C}}(\gamma_s) - \bar{\mathbb{C}}(\chi|\gamma_s)\|}{\|\hat{\mathbb{C}}(\gamma_s)\|} + \frac{\lambda}{2} G(\chi)$$

Online evaluation

Extrapolation to **unknown** material and loading spaces: NR iterative resolution for nonlinear equations

- top-bottom backward de-homogenisation process
- bottom-top forward homogenisation process



Outline

- 1 Introduction
- 2 Reduced-order models
- 3 Offline training and online evaluation
- 4 Application**
 - Uniaxial tensile test and shear test
 - Local damage models calibration
 - Damage-enhanced uniaxial tensile test
- 5 Conclusions and outlooks

Uniaxial (cyclic) tensile test & In-plane cyclic shear test

- training results is available in [Wu, Adam, and Noels 2021]
- damage-free nonlinear predictions with
 - fibres: transversely isotropic linear elasticity
 - matrix: isotropic J_2 elasto-plasticity

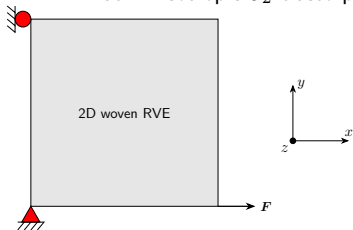


Figure: Uniaxial tensile test

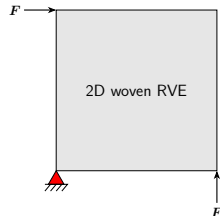


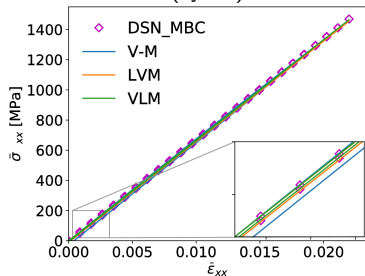
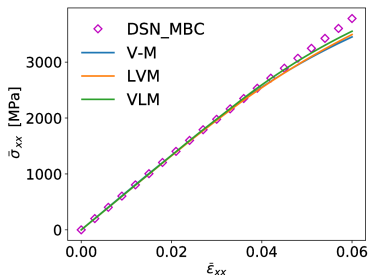
Figure: In-plane shear test

- comparison of the CPU time

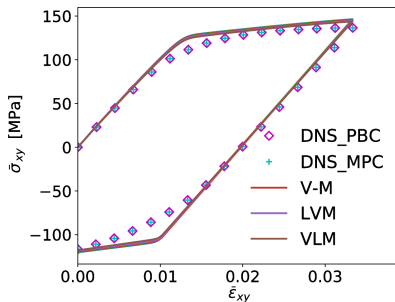
	Tensile	Shear
DNS	11 h	76 h
VM	30 s	22 s
LVM	35 s	25 s
VLM	35 s	22 s

Uniaxial (cyclic) tensile test & In-plane cyclic shear test

- comparison of the global responses for the uniaxial tensile (cyclic) test



- comparison of the global responses for the in-plane cyclic shear test



Local damage models calibration

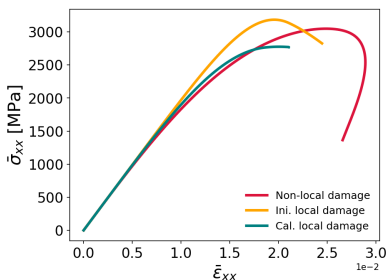
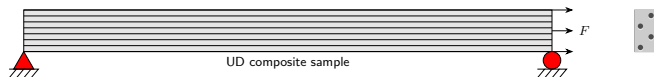
Non-local damage models are used for DNS

- phase field-like damage model for fibres, Lemaitre-like damage model for matrix [Wu et al. 2021]

Local damage models are used for VM, LVM and VLM

- no physical “characteristic length” between material nodes
- Weibull-like damage model for fibres, Lemaitre-like damage model for matrix [Wu, Maillard, and Noels 2021]

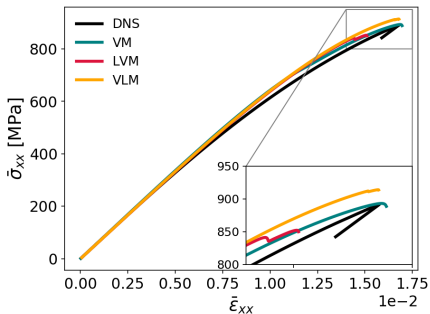
Calibration with an uniaxial tensile test on a UD composite sample



Damage-enhanced uniaxial tensile test - Load until failure

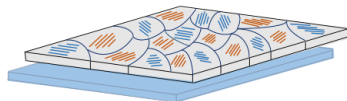
Comparison of the global responses

	Strength [GPa]
DNS	891.778
VM	892.981
rel. err. (%)	0.13
LVM	852.096
rel. err. (%)	4.45
VLM	913.868
rel. err. (%)	2.48



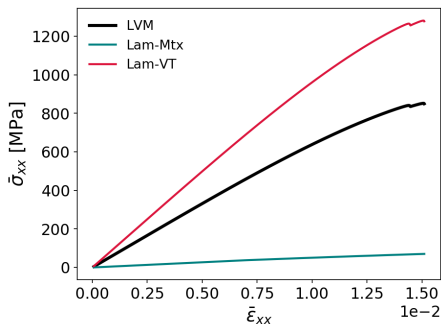
Uniaxial tensile test - Load until failure

LVM training results and responses of the two plies



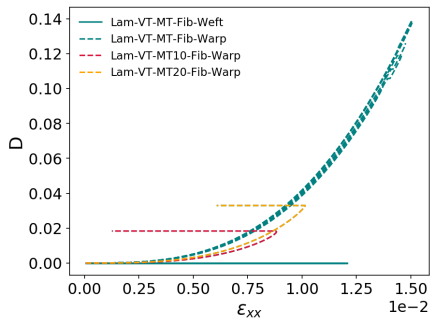
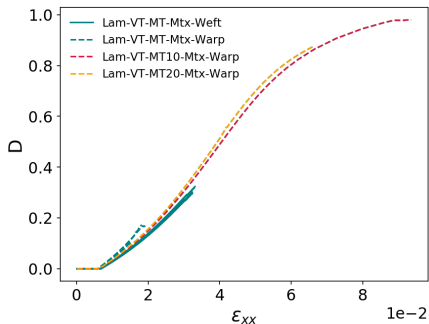
- Aggregate ply takes the majority load
- Grains 9, 10, 19 and 20 have a small α

Grain	α
1 (weft) & 2 (warp)	216.645
3 (weft) & 4 (warp)	258.721
5 (weft) & 6 (warp)	195.969
7 (weft) & 8 (warp)	181.194
9 (weft) & 10 (warp)	21.032
11 (weft) & 12 (warp)	247.153
13 (weft) & 14 (warp)	204.540
15 (weft) & 16 (warp)	123.377
17 (weft) & 18 (warp)	298.981
19 (weft) & 20 (warp)	23.852



Uniaxial tensile test - Load until failure

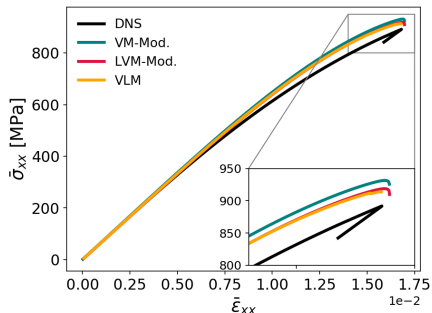
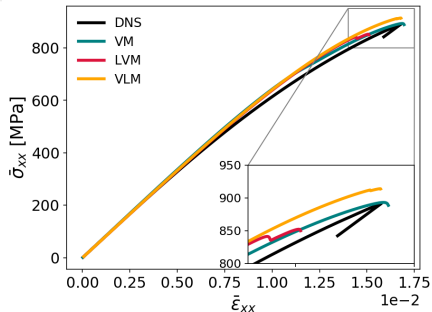
Matrix and fibres **damage** evolution in the grains of the aggregate ply:



Uniaxial tensile test - Load until failure

- Deactivation of matrix damage in
 - VM: grains 2 and 3
 - LVM: grains 9, 10, 19 and 20
- Comparison of the global responses:

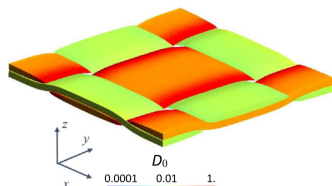
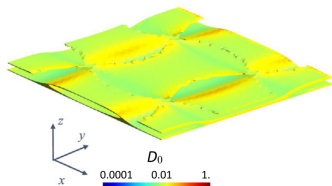
	Strength [GPa]	CPU Time
DNS	891.778	64 h
VM	892.981	
rel. err. (%)	0.13	19 s
VM-Mod.	931.430	
rel. err. (%)	4.45	
LVM	852.096	
rel. err. (%)	4.45	6 mins
LVM-Mod.	918.646	
rel. err. (%)	3.01	
VLM	913.868	17 mins
rel. err. (%)	2.48	



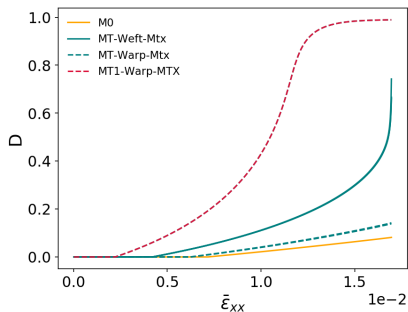
Uniaxial tensile test - Load until failure

Comparison of matrix damage state

- DNS

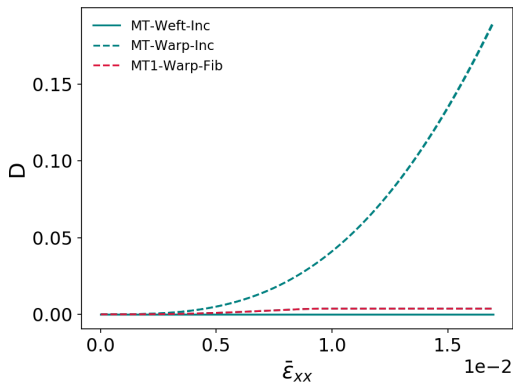
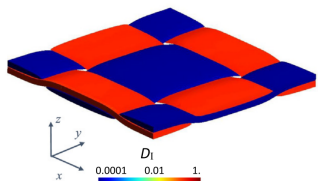


- VM



Uniaxial tensile test - Load until failure

Comparison of fibres damage state (DNS vs. VM)



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- 1 Introduction
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Conclusions and outlooks

Conclusions

- Construction of **DMN-liked** and **micromechanics-based damage-enhanced** ROMs (VM, LVM and VLM) for 2D woven composites
- Efficient offline learning only in **linear elasticity**
- Accurate online extrapolation to **arbitrary unknown nonlinear inelastic** material laws and loading spaces

Outlooks

- Improvement of the accuracy by considering more suitable **damage** models
- Application for **3D** woven composites and for **other types** of composites

Thanks for your attention !



Liu, Z. and C. Wu (2019). "Exploring the 3D architectures of deep material network in data-driven multiscale mechanics". In: *Journal of the Mechanics and Physics of Solids* 127, pp. 20–46.



Sun, Z. et al. (2020). "Numerical analysis of out-of-plane thermal conductivity of C/C composites by flexible oriented 3D weaving process considering voids and fiber volume fractions". In: *Journal of Materials Research* 35.14, pp. 1888–1897.



Wang, L. et al. (2017). "Progressive failure analysis of 2D woven composites at the meso-micro scale". In: *Composite structures* 178, pp. 395–405.



Wu, L., L. Adam, and L. Noels (2021). "Micro-mechanics and data-driven based reduced order models for multi-scale analyses of woven composites". In: *Composite Structures* 270, p. 114058.



Wu, L., E. Maillard, and L. Noels (2021). "Tensile failure model of carbon fibre in unidirectionally reinforced epoxy composites with mean-field homogenisation". In: *Composite Structures* 273, p. 114270.



Wu, L. et al. (2021). "Per-phase spatial correlated damage models of UD fibre reinforced composites using mean-field homogenisation; applications to notched laminate failure and yarn failure of plain woven composites". In: *Computers & Structures* 257, p. 106650.

Local damage evolution (VM, LVM or VLM)

$$D_1 = \varphi + \varrho \sqrt{\frac{\varphi(1-\varphi)}{N}}$$

with

- $\varphi(\hat{\sigma}_1^{zz}) \sim$ Weibull distribution, with $\hat{\sigma}_1 = \mathbb{C}_1^{\text{el}} : \varepsilon_1$
- ϱ one realisation of $P \sim \mathcal{N}(0, 1)$, and N the number of fibres in a bundle

Non-local damage evolution (full-order model)

$$D_1(x, t) = 1 - (1 - d_1(x, t))^n$$

with

- $d_1(x, t)$ given by the phase-field model
- $n \in [2, 3]$ shape parameter

Local (VM, LVM or VLM) and **non-local** (full-order model) damage evolution

$$D_0 = \frac{D_0^{\max}}{1 - \frac{1}{1 + \exp\{(s_0 p_{c_0})\}}} \left(\frac{1}{1 + \exp\{(-s_0(\chi_0 - p_{c_0}))\}} - \frac{1}{1 + \exp\{(s_0 p_{c_0})\}} \right)$$

with

- D_0^{\max} the saturation damage level
- s_0 and p_{c_0} two material parameters
- $\chi_0(t)$ guarantee of the irreversibility
 - **local**: $\chi_0(t) = \max_{\tau \in [0, t]} (p_0)$ with $\dot{p}_0 = \sqrt{\frac{2}{3} \dot{\epsilon}_0^{\text{pl}} : \dot{\epsilon}_0^{\text{pl}}}$
 - **non-local**: $\chi_0(\mathbf{x}, t) = \max_{\tau \in [0, t]} (\tilde{p}_0)$ with $\tilde{p}_0 - \nabla \cdot (c_0 \cdot \nabla \tilde{p}_0) = p_0$

Information paths for a parent node with N_d direct child nodes

- top-bottom **strain** distribution:

$$\Delta \boldsymbol{\varepsilon} = \begin{cases} \sum_{i=1}^{N_d} v_i^d \Delta \boldsymbol{\varepsilon}_i^d, & \text{for MFH-based and 2-ply laminate-based node} \\ \{\Delta \boldsymbol{\varepsilon}_i^d\}_{i=1, \dots, N_d}, & \text{for Voigt's mixtures-based node} \end{cases}$$

- bottom-top **stress** and **algorithmic operator** collection:

$$\boldsymbol{\sigma} = \sum_{i=1}^{N_d} v_i^d \boldsymbol{\sigma}_i^d$$
$$\mathbb{C}^{\text{alg}} = \mathbb{C}^{\varepsilon\varepsilon} = \frac{d\boldsymbol{\sigma}}{d\boldsymbol{\varepsilon}} = \sum_{n=1}^{N_d} v_n^d \mathbb{C}_n^{\text{alg}, d} : \frac{d\boldsymbol{\varepsilon}_n^d}{d\boldsymbol{\varepsilon}}$$

Appendix B.1 - Nonlinear analysis for MFH-based node

Linearisation of the **composite** parent node by a damage-enhanced **incremental-secant-based Linear Comparison Composite (LCC)**

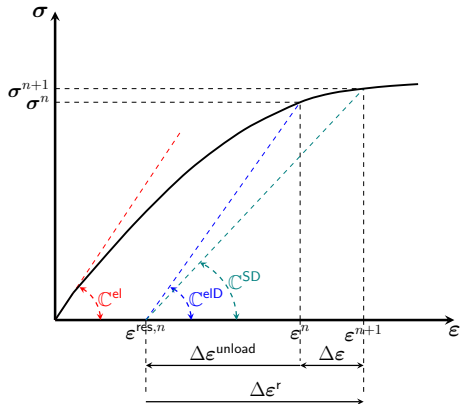
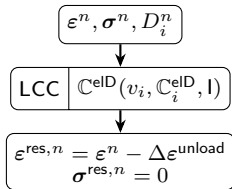
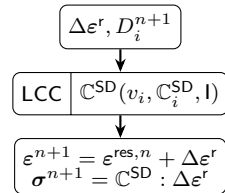


Figure: Composite parent node linearisation

- unloading: (i for fibre and matrix)



- reloading: (i for fibre and matrix)



Appendix B.1 - Nonlinear analysis for MFH-based node

Linearisation of the fibre child node by a damage-enhanced transversely isotropic elastic material with $E_1^{zD} = (1 - D_1)E_1^z$ and $\nu_1^{zxD} = (1 - D_1)\nu_1^{zx}$

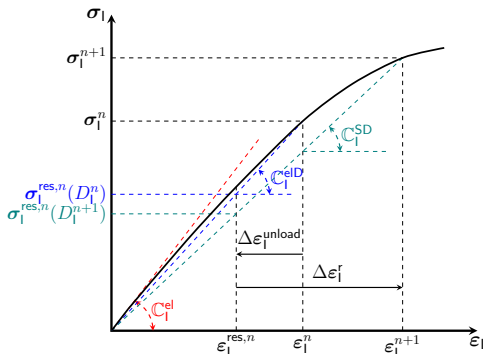
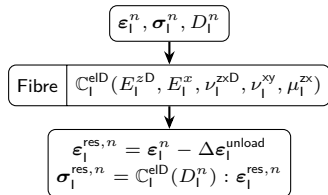
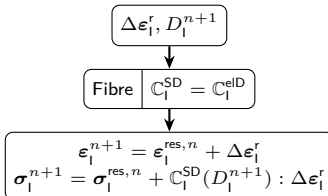


Figure: Fibre child node linearisation

• unloading:



• reloading:

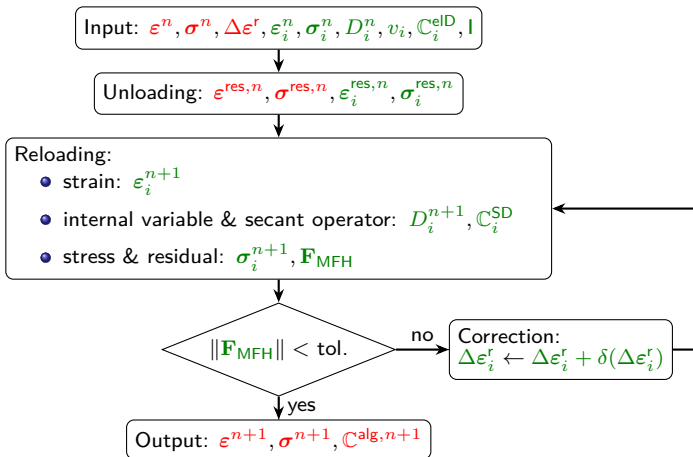


Appendix B.1 - Nonlinear analysis for MFH-based node

Nonlinear equation for the MFH problem: (\mathbb{S} the so-called Eshelby tensor)

$$\mathbf{F}_{\text{MFH}} = \mathbb{C}_0^{\text{SD}} : \left[\Delta \boldsymbol{\varepsilon}_i^r - \frac{1}{v_0} \mathbb{S}^{-1}(\mathbf{I}, \mathbb{C}_0^{\text{SD}}) : (\Delta \boldsymbol{\varepsilon}_i^r - \Delta \boldsymbol{\varepsilon}^r) \right] - \mathbb{C}_i^{\text{SD}} : \Delta \boldsymbol{\varepsilon}_i^r$$

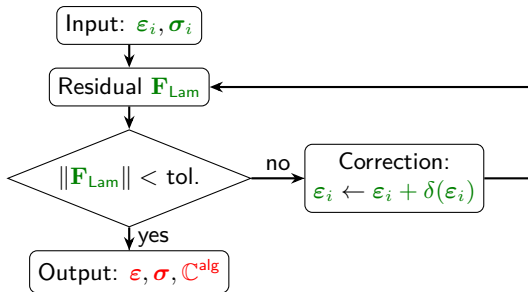
NR iterative resolution: (i for fibre or matrix)



Nonlinear equation to satisfy for the 2-ply laminate problem:

$$\mathbf{F}_{\text{Lam}} = \begin{bmatrix} (\sigma_A^{\text{zz}} - \sigma_B^{\text{zz}}) + (\varepsilon_A^{\text{xx}} - \varepsilon_{\text{Lam}}^{\text{xx}}) \\ (\sigma_A^{\text{xz}} - \sigma_B^{\text{xz}}) + (\varepsilon_A^{\text{yy}} - \varepsilon_{\text{Lam}}^{\text{yy}}) \\ (\sigma_A^{\text{yz}} - \sigma_B^{\text{yz}}) + (\varepsilon_A^{\text{xy}} - \varepsilon_{\text{Lam}}^{\text{xy}}) \end{bmatrix} = \mathbf{0}$$

NR iterative resolution: (i for A-ply or B-ply)



No nonlinear equation to be solved for the Voigt's mixtures problem

VM			
Grain	θ	v	α
1 (matrix)	-	0.3544	-
2 (weft) & 3 (warp)	[0/90, 89.99982, 0]	0.0204	6.469
4 (weft) & 5 (warp)	[0/90, 89.99986, 0]	0.2737	151.095
6 (weft) & 7 (warp)	[0/90, 89.99985, 0]	0.0286	105.591

Appendix C.2 - Offline training results for LVM

Grain	LVM		
	θ	v	α
1 (weft) & 2 (warp)	[0/90, 82.432, 0]	0.0314	216.645
3 (weft) & 4 (warp)	[0/90, 90.648, 0]	0.2618	258.721
5 (weft) & 6 (warp)	[0/90, 90.651, 0]	0.0770	195.969
7 (weft) & 8 (warp)	[0/90, 93.161, 0]	0.0241	181.194
9 (weft) & 10 (warp)	[0/90, 116.875, 0]	0.0176	21.032
11 (weft) & 12 (warp)	[0/90, 92.995, 0]	0.0179	247.153
13 (weft) & 14 (warp)	[0/90, 80.612, 0]	0.0105	204.540
15 (weft) & 16 (warp)	[0/90, 102.166, 0]	0.0073	123.377
17 (weft) & 18 (warp)	[0/90, 88.039, 0]	0.0312	298.981
19 (weft) & 20 (warp)	[0/90, 66.960, 0]	0.0211	23.852

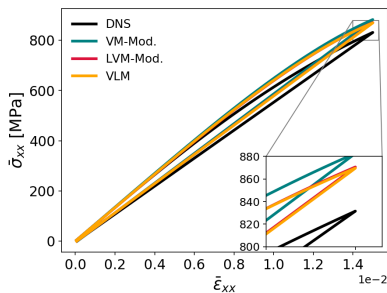
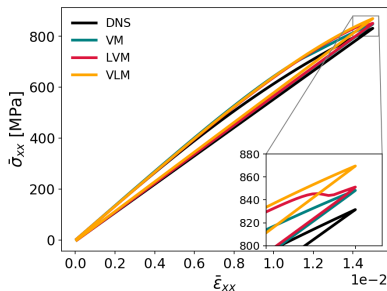
Appendix C.3 - Offline training results for VLM

Grain	VLM				
	θ^f	θ^g	v^m	v^g	α
1 (weft) & 2 (warp)	[0, 89.963, 0]	[0/90, 0.479, 0]	0.2291	0.2721	152.599
3 (weft) & 4 (warp)	[0, 179.575, 0]	[0/90, -89.584, 0]	0.5216	0.0204	542.078
5 (weft) & 6 (warp)	[0, 89.698, 0]	[0/90, 8.230, 0]	0.5514	0.0614	285.184
7 (weft) & 8 (warp)	[0, 90.256, 0]	[0/90, -6.628, 0]	0.5073	0.0681	207.968
9 (weft) & 10 (warp)	[0, 90.108, 0]	[0/90, -2.391, 0]	0.4424	0.0755	189.845
11 (weft) & 12 (warp)	[0, 90.095, 0]	[0/90, -59.110, 0]	0.9995	0.0024	367.619

Appendix D.1 - Uniaxial tensile test - Load-Unload

Comparison of the global responses:

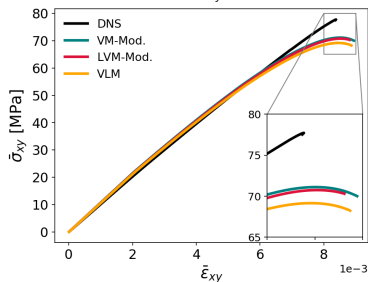
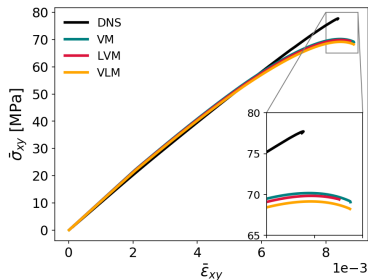
	Strength [GPa]	CPU Time [mins]
DNS	831.481	1204
VM	848.241	0.05
rel. err. (%)	2.02	
VM-Mod.	882.647	1
rel. err. (%)	6.15	
LVM	851.166	1
rel. err. (%)	2.37	
LVM-Mod.	870.576	1
rel. err. (%)	4.70	
VLM	869.539	1
rel. err. (%)	4.58	



Appendix D.2 - In-plane shear test - Load until failure

Comparison of the global responses: (predictions less accurate)

	Strength [GPa]	CPU Time [mins]
DNS	77.721	1435
VM	70.179	
rel. err. (%)	9.70	2
VM-Mod.	71.112	
rel. err. (%)	8.50	
LVM	69.838	
rel. err. (%)	10.14	37
LVM-Mod.	70.747	
rel. err. (%)	8.97	
VLM	69.155	
rel. err. (%)	11.02	9



Appendix D.1 - In-plane shear test - Load-Unload

Comparison of the global responses: (predictions less accurate)

	Strength [GPa]	CPU Time [mins]
DNS	76.906	1149
VM	70.048	
rel. err. (%)	8.92	
VM-Mod.	70.865	0.2
rel. err. (%)	7.85	
LVM	69.677	
rel. err. (%)	9.40	4
LVM-Mod.	70.472	
rel. err. (%)	8.37	
VLM	69.004	2
rel. err. (%)	10.27	

