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Abstract

Dichotomous high-Ti and low-Ti magmas are ubiquitous in large igneous provinces (LIPs). These magmas often form economically critical Fe–Ti oxide ores in layered intrusions via mechanisms that remain debated. To constrain the evolution of high-Ti and low-Ti basalts during fractionation, we performed stepwise equilibrium crystallization experiments at atmospheric pressure. We specifically aimed to quantify the influences of starting composition and oxygen fugacity (f_{O_2}) on phase stability, phase compositions, and the onset of silicate liquid immiscibility during cooling. Both types of magma crystallize similar phase assemblages at QFM (quartz–fayalite–magnetite thermodynamic equilibrium) and QFM + 2: olivine, clinopyroxene, plagioclase, Fe–Ti oxides, and whitlockite. Tridymite crystallizes late in experiments at QFM + 2. The starting composition exerts a strong influence on phase and melt compositions. High CaO and Al_2O_3 contents in the melt favor the early crystallization of plagioclase and enhance FeO enrichment before Fe–Ti oxide saturation. f_{O_2} affects the composition and stability of Fe–Ti oxides, and high f_{O_2} conditions may promote melt differentiation into the calc-alkaline field. Silicate liquid immiscibility occurs in both compositional trajectories, producing Fe-rich melt globules in equilibrium with Si-rich melts. Strong iron enrichment is not necessary for immiscibility to develop; unmixing also occurs in Fe depleted compositions. We propose a new parameterization to map the binodal surface in temperature-composition space that successfully fits the two-liquid field in experiments and natural immiscible compositions. Our results indicate that Fe–Ti oxide ores in layered intrusions associated with LIPs form by the segregation of Fe-rich melts and/or the accumulation of early crystallized Fe–Ti oxides during fractionation.

Keywords (separated by '- ')

Flood basalt - Large igneous province - Immiscibility - Phase equilibria - Liquid line of descent

Footnote Information

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2 Experimental study of high-Ti and low-Ti basalts: liquid lines 3 of descent and silicate liquid immiscibility in large igneous provinces

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7 Abstract

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9 economically critical Fe–Ti oxide ores in layered intrusions via mechanisms that remain debated. To constrain the evolu-
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24 **Keywords** Flood basalt · Large igneous province · Immiscibility · Phase equilibria · Liquid line of descent

25 Introduction

26 Large igneous provinces (LIPs) are regions compris-
27 ing extremely large volumes of mafic/ultramafic magmas
28 erupted or emplaced in the upper crust within a relatively
29 short time (~1–5 Myr; Bryan and Ernst 2008). As the largest
30 volcanic episodes on Earth and occurred in both oceanic and
31 continental crust, they are thought to result from hot mantle
32

plumes causing high degrees of melting at the base of the
lithosphere (Campbell and Griffiths 1990; Campbell 2005). 33

LIPs generally contain large volumes of moderately 34
evolved, phenocryst-poor, lavas (< 8 wt% MgO) asso- 35
ciated with rare picritic dikes, and, locally, ultramafic/ 36
mafic layered intrusions. They usually show a geochemi- 37
cal dichotomy between high-Ti and low-Ti compositions 38
(Cox et al. 1967), which are discriminated based on their 39
 TiO_2 contents or ratios of Ti to highly incompatible trace 40
elements (e.g., Ti/Y, Peate et al. 1992). These contrasting 41
magma compositions may reflect lithologically distinct 42
mantle sources and/or the degree to which the sources 43
are enriched in incompatible elements (Marsh et al. 2001; 44
Bryan and Ernst 2008; Ernst 2014; Heinonen et al. 2022). 45
However, recent studies have scrutinized this classifica- 46
tion, because the melt evolution during cooling, especially 47
with respect to TiO_2 , depends strongly on oxygen fugac- 48
ity (fO_2), pressure, and temperature, and can produce a 49

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continuum of lava compositions between high-Ti and low-Ti endmembers (e.g., Kamenetsky et al. 2012). In addition, differences between the fO_2 conditions of high-Ti and low-Ti magmas, attributed to different degrees of crustal assimilation (e.g., Xu et al. 2003), influence phase equilibria, especially the crystallization of Fe–Ti oxides. Therefore, a low-Ti composition could theoretically evolve into a high-Ti composition if Fe–Ti oxide saturation is delayed (Hou et al. 2011).

To better understand the compositional diversity of LIPs, it is important to investigate the differentiation of high-Ti and low-Ti magmas and systematically evaluate the roles of primary magma composition and fO_2 in determining phase relations and differentiation processes. Fe–Ti oxides ore deposits are commonly preserved in LIPs and are geochemically associated with high-Ti lavas (e.g., Pang et al. 2010; Zhang et al. 2018; Fan et al. 2013). The formation of these deposits remains controversial, they form by either the density-driven accumulation of Fe–Ti oxides during fractional crystallization (Charlier et al. 2010; Pang et al. 2008), or their crystallization from a segregated immiscible Fe-rich melt (Zhou et al. 2005, 2013; Fischer et al. 2016). However, tracking the onset of silicate immiscibility has relied heavily on relatively few experiments (e.g., Charlier and Grove 2012) and thermodynamic modelling (Ghiorso and Carmichael 1980; Ghiorso et al. 1983). Unfortunately, no valid predictive model for the development of immiscibility currently exists, hindering our ability to validate or invalidate whether of Fe–Ti oxide ore formation requires silicate liquid immiscibility.

In this study, we use a stepwise experimental approach to closely reproduce the fractional crystallization of high-Ti and low-Ti basalts under various fO_2 conditions at atmospheric pressure. Our results show that the different starting compositions, i.e. primary magma variability, strongly affect phase and melt compositions. The early crystallization of plagioclase in the low-Ti composition may enhance FeO enrichment in the melt and crystallinity at a given degree of fractionation. Oxygen fugacity has a profound effect on the stability, saturation temperature and chemistry of Fe–Ti oxides, in turn affecting the differentiation of the melt with respect to FeO and MgO as well as other major elements. We demonstrate that the residual melts of high-Ti and low-Ti basalts at both low and high fO_2 conditions develop silicate liquid immiscibility during cooling and fractionation. We propose a new parameterization to describe the binodal surface of the immiscible pairs and thus the temperature-composition space of the two-liquid field. We apply our model in the ~260 Ma Emeishan LIP, China, and show that the development of immiscibility in high-Ti lavas may contribute to the formation of Fe–Ti oxide ores.

Experimental strategy and analytical methods

Starting materials and experimental conditions

The high-Ti and low-Ti starting materials were selected based on a compiled database of lava compositions from the Emeishan LIP. Geochemical and structural features of the Emeishan province, such as kilometer-scale uplift, indicate its mantle plume origin (e.g., Chung and Jahn 1995; Xu et al. 2001). We consider the Emeishan province to be representative of world-wide LIPs; for example, they share similar bulk rock geochemistry with the Karoo LIP, Southern Africa (Fig. 1). Bulk-rock compositions of Emeishan rocks are classified as high-Ti and low-Ti using a threshold Ti/Y ratio of 500 (Xu et al. 2001; Xiao et al. 2004). Nonetheless, a continuum of bulk-rock Ti/Y ratios and TiO_2 contents are observed for both provinces (Fig. 1a, b). We excluded compositions with > 15 wt% MgO to avoid drastic bulk-rock compositional changes due to olivine \pm spinel accumulation. Therefore, the bulk-rock compositions in our database should close to the true liquid compositions. Finally, we selected high-Ti and low-Ti experimental starting compositions (HT1 and LT1, respectively; Table 1) by averaging the major oxide compositions of each group at ~14 wt% MgO, corresponding to the estimated primary melt for the Emeishan LIP (Xu et al. 2020).

The starting materials for experiments are synthetic powders prepared from mixtures of high-purity oxides and silicates (SiO_2 , TiO_2 , Al_2O_3 , MnO, MgO, Fe_2O_3 , $CaSiO_3$, $AlPO_4$, Na_2SiO_3 , $K_2Si_4O_9$, Cr_2O_3) in appropriate proportions. The $CaSiO_3$, Na_2SiO_3 , and $K_2Si_4O_9$ silicates were produced by mixing appropriate proportions of carbonates and SiO_2 , then reacted and decarbonated for 5 days at 1100 °C for $CaSiO_3$ and at 750 °C for Na_2SiO_3 , and $K_2Si_4O_9$. The silicate compounds were then checked for homogeneity, no silica relics were observed. The products were weighed to ensure complete decarbonization and then grinded in ethanol in an agate mortar for 1 h.

We employed a three-step experimental approach to simulate fractional crystallization of the high-Ti and low-Ti starting compositions at fO_2 conditions of QFM (quartz–fayalite–magnetite thermodynamic equilibrium) and QFM + 2 (Table 2). The corresponding fO_2 conditions at different temperature were calculated following O'Neill (1987), and the obtained fO_2 values are consistent with the estimated fO_2 range of different LIPs (e.g., Bai et al. 2019; Freise et al. 2009; Cao and Wang 2022). Additional experiments, referring as 'step 4' in Table 1, were performed to explore the potential development of silicate liquid immiscibility at the lowest experimental temperature. In step 1 (runs ending in 's1' in Table 2), compositions

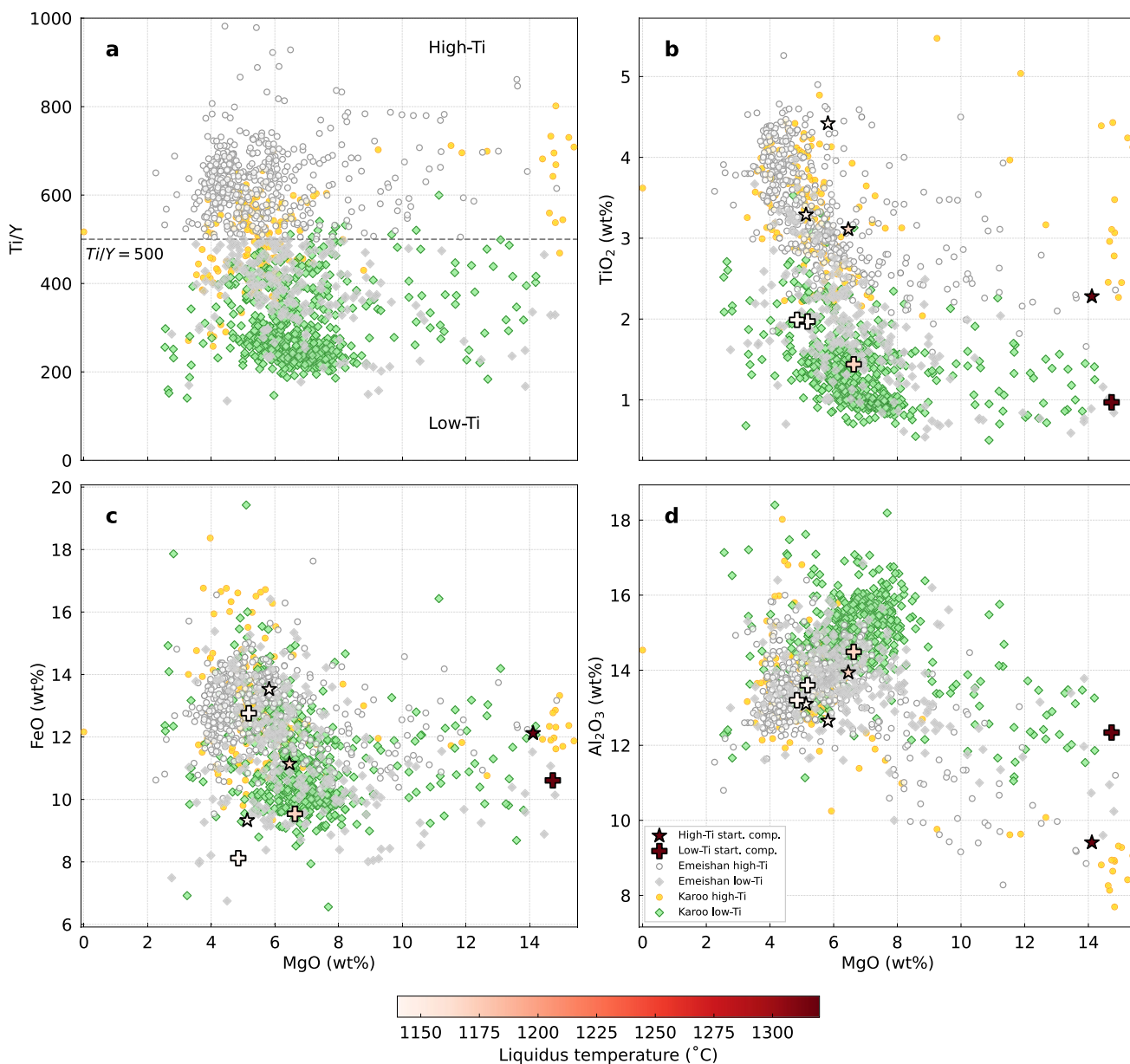


Fig. 1 Compositions of our starting materials compared to high-Ti and low-Ti lavas from the Emeishan and Karoo LIPs. The dashed line in **a** indicates $Ti/Y=500$, the criterion used by Xu et al. (2001) to classify Emeishan lavas as high-Ti or low-Ti. Symbols indicating starting compositions according to their liquidus temperature. The Emeishan lavas are from a compiled database (Chung and Jahn 1995;

Song et al. 2001; Xu et al. 2001; Xiao et al. 2004; Zhou et al. 2005, 2008; Zhang et al. 2006; Wang et al. 2007; Qi et al. 2008; Hanski et al. 2010; He et al. 2010; Kamenetsky et al. 2012; Tao et al. 2015; Arguin et al. 2016; Ren et al. 2017). Data for Karoo LIP are from the GEOROC database (<https://georoc.mpch-mainz.gwdg.de/georoc/>)

152 HT1 and LT1 were used for isothermal experiments at
 153 temperatures ranging from 1330 °C (super-liquidus) to
 154 1160 °C. These compositions typically attained ~50%
 155 crystallinity at 1170 °C. Then, we prepared synthetic
 156 compositions HT2 and LT2 (Table 1) corresponding to
 157 the residual melts produced at 1170 °C under QFM + 2 in
 158 step 1 (runs 18As1 and 18Bs1). We prepared only these
 159 two new starting materials, because the compositions of

the residual liquids at QFM are almost identical to those at
 QFM + 2 due to the absence of Fe–Ti oxide crystallization
 in the step 1 temperature interval (see below). These new
 compositions were used in step 2 (runs ending in ‘s2’ in
 Table 2) for isothermal experiments at temperatures rang-
 ing from 1175 to 1120 °C. Due to the significant influ-
 ence of fO_2 on the TiO_2 and FeO contents of the melts
 in this temperature range, particularly after saturation of

Table 1 Compositions (wt%) of starting materials in this and other studies

Sample no	SiO ₂	TiO ₂	Al ₂ O ₃	FeO _t	MnO	MgO	CaO	Na ₂ O	K ₂ O	P ₂ O ₅	Cr ₂ O ₃	Total	Ca#	Al#
This study														
HT1 (step 1)	48.79	2.28	9.41	12.12	0.18	14.11	10.74	2.03	0.51	0.20	0.13	100.12	0.84	0.16
HT2 (step 2)	49.91	3.11	13.93	11.14	0.16	6.46	12.37	2.55	0.42	0.22	0.02	100.29	0.83	0.22
HT31 (step 3 and 4)	48.43	4.42	12.65	13.53	0.19	5.82	10.40	2.13	0.81	0.23	0.02	98.63	0.83	0.21
HT32 (step 3 and 4)	54.63	3.29	13.10	9.33	0.21	5.13	9.19	2.82	0.83	0.19	0.04	98.76	0.77	0.19
LT1 (step 1)	49.09	0.97	12.34	10.61	0.17	14.73	11.31	1.32	0.76	0.10	0.14	99.54	0.90	0.20
LT2 (step 2)	53.68	1.44	14.49	9.54	0.19	6.63	11.91	1.96	0.95	0.16	0.02	99.97	0.86	0.21
LT31 (step 3 and 4)	53.89	1.97	13.60	12.76	0.17	5.18	9.42	1.43	1.19	0.18	0.01	99.80	0.87	0.20
LT32 (step 3 and 4)	59.90	1.99	13.20	8.12	0.19	4.85	8.00	1.67	1.58	0.16	0.01	99.67	0.83	0.18
Snyder et al. (1993)														
4-3	47.4	2.27	14.2	14.34	0.20	6.13	9.04	2.97	0.70	0.22	–	99.2	0.75	0.23
4-83C	46.2	3.83	13.4	17.33	0.22	4.66	7.58	3.56	1.03	0.35	–	99.5	0.68	0.22
Toplis and Carroll (1995)														
SC1	48.8	2.90	14.90	13.10	–	6.50	10.90	2.70	0.30	–	–	100.10	0.80	0.23

Ca# = Ca/(Na + Ca) and Al# = Al/(Al + Si)

168 Fe–Ti oxides, four starting materials with distinct FeO and
 169 TiO₂ contents were then synthesized as third-step compo-
 170 sitions (HT31, LT31, HT32, and LT32; Table 1). These
 171 compositions correspond to the residual melt composi-
 172 tions produced at 1140 °C at QFM and QFM + 2 in step
 173 2 (runs 11As2, 11Bs2, 12As2, and 12Bs2; Table 2). Step
 174 3 experiments were performed with a 1 °C/h cooling rate
 175 from above the liquidus temperature (1160 °C) to final
 176 temperatures ranging from 1080 to 1040 °C.

177 An additional set of ten experiments (step 4) was per-
 178 formed with an identical cooling protocol and temperature
 179 range as step 3. Step 4 experiments were performed with the
 180 same four compositions as in step 3 but under different *f*O₂
 181 conditions (see Table 2). For example, the high *f*O₂ starting
 182 materials (HT32 and LT32) of both series were used for
 183 additional experiments at more reduced conditions (QFM).
 184 This set of experiments was performed to test the develop-
 185 ment of liquid immiscibility in a variety of melt composi-
 186 tions under changing *f*O₂. The compositions of all starting
 187 materials are shown in Fig. 1.

188 **Experimental methods**

189 All experiments were performed in a GERO HTRV
 190 70-250/18 vertical tube furnace at University of Liège, Bel-
 191 gium and in a Nabertherm HTRV 50/150/17 vertical tube
 192 furnace at KU Leuven, Belgium. Approximately 50 mg of
 193 the starting powders were suspended on a 0.2-mm-wide Pt
 194 loop and heated to a super-liquidus temperature for 10 min
 195 in a muffle furnace to produce glass. The starting glass was
 196 then suspended in the hot spot of the vertical furnace (~5 cm
 197 of the furnace where the temperature is within ± 1 °C).
 198 Experimental temperatures were measured using a S-type

(Pt–Pt₉₀Rh₁₀) thermocouple calibrated against the melting
 points of Ag and Au. To minimize Fe loss from the start-
 ing materials, the Pt loops were pre-saturated in Fe by elec-
 troplating following the method of Grove (1981). Oxygen
 fugacity was buffered at QFM and QFM + 2 by flushing the
 furnace with mixtures of high-purity CO and CO₂ using
 Bronkhorst gas-flow controllers operating at a flow rate of
 0.12 cm/s.

199 Isothermal experiments (steps 1 and 2) were equilibrated
 200 for 24–96 h at target temperatures between super-liquidus
 201 (1320–1330 °C) and 1120 °C. To produce crystals large
 202 enough for reliable analyses in lower temperature step 3 and
 203 step 4 experiments, we used a three-step approach: experi-
 204 ments were (1) held above the liquidus (1160 °C) for ~24 h,
 205 (2) cooled to 1080–1040 °C at a rate of 1 °C/h, and then (3)
 206 held at the final temperature for an additional 176–240 h.
 207 Four additional isothermal experiments (192 h durations)
 208 were performed at 1060 °C and QFM + 2 to compare their
 209 phase assemblages with the cooling experiments to check
 210 how the reliability of the cooling experiments and to ver-
 211 ify whether the cooling experiments attained full thermo-
 212 dynamic equilibrium. All experiments were quenched by
 213 dropping the charges in water. Details on experimental con-
 214 ditions, run products, and phase proportions are reported
 215 in Table 2.

224 **Analytical methods**

225 The experimental charges were analyzed using a JEOL
 226 JXA-8530F wavelength dispersive spectrometer cou-
 227 pled to a field emission gun electron microprobe at Uni-
 228 versity of Münster, Germany and at the Department of
 229 Material Engineering, KU Leuven, Belgium. We used a

Table 2 Experimental conditions, phase assemblages, and phase proportions

Run no	Start. comp	T (°C)	Duration (h)	fO ₂	Method	Phase assemblage	Phase proportions ^a	Σf ²	ΔFe (%)	ΔNa (%)
Super-liquidus										
00A/00B	HT1/LT1	1330	24	QFM	Isothermal	Gl	-	-	-	-
22A/22B	HT1/LT1	1330	24	QFM + 2	Isothermal	Gl	-	-	-	-
2As2/10Bs2	HT2/LT2	1175	24	QFM	Isothermal	Gl	-	-	-	-
7As2/7Bs2	HT2/LT2	1175	24	QFM + 2	Isothermal	Gl	-	-	-	-
H104/H204/L104/L204	HT31/HT32 /LT31/LT32	1140	24	QFM + 2	Isothermal	Gl	-	-	-	-
Liquid lines of descent experiments (step 1–3)										
High-Ti experiments										
02As1	HT1	1320	24	QFM	Isothermal	Gl, Oi	98:02:00	0.47	0	2.36
01As1	HT1	1302	24	QFM	Isothermal	Gl, Oi	95:05:00	0.8	0	10.57
04As1	HT1	1280	48	QFM	Isothermal	Gl, Oi, Spl	92:07:01	0.17	0	11.51
03As1	HT1	1260	48	QFM	Isothermal	Gl, Oi, Spl	90:10:00	0.22	0	35.84
05As1	HT1	1240	48	QFM	Isothermal	Gl, Oi, Spl	88:12:00	0.31	0	8.02
06As1	HT1	1200	48	QFM	Isothermal	Gl, Oi, Spl, Aug	82:15:0:3	0.12	0	10.12
15As1	HT1	1180	72	QFM	Isothermal	Gl, Oi, Spl, Aug	74:16:0:10	0.24	0	7.45
17As1	HT1	1170	72	QFM	Isothermal	Gl, Oi, Spl, Aug	72:17:0:11	0.19	0.39	5.44
14As1	HT1	1160	72	QFM	Isothermal	Gl, Oi, Spl, Aug	67:17:0:16	0.05	0	9.89
1As2	HT2	1160	72	QFM	Isothermal	Gl, Aug, Pl	81:09:10	0.18	13.06	5.44
11As2	HT2	1140	72	QFM	Isothermal	Gl, Aug, Pl	61:20:19	0.06	1.1	0
3As2	HT2	1120	72	QFM	Isothermal	Gl, Aug, Pl, Mag	33:30:30:7	0.06	0	2.97
H105	HT31	1160–1080	176	QFM	1 °C/h	Gl, Aug, Pl, Ilm, Mag	36:30:25:4:5	0.44	0	1.31
H103	HT31	1160–1060	225	QFM	1 °C/h	Gl, Aug, Pl, Ilm, Mag, Wht (tr)	29:33:28:5:5:tr	0.05	0	7.61
H106	HT31	1160–1040	240	QFM	1 °C/h	Si-Gl, Fe-Gl, Aug, Pl, Ilm, Mag, Wht (tr)	26:0:34:28:6:6:tr	0.14	0	9.42
12As1	HT1	1320	24	QFM + 2	Isothermal	Gl	100	0	0	0
07As1	HT1	1302	24	QFM + 2	Isothermal	Gl, Oi	97:03:00	0.11	0	1.81
08As1	HT1	1280	48	QFM + 2	Isothermal	Gl, Oi, Spl	94:06:00	0.11	0	2.82
09As1	HT1	1260	48	QFM + 2	Isothermal	Gl, Oi, Spl	92:08:00	0.02	0	5.98
10As1	HT1	1240	48	QFM + 2	Isothermal	Gl, Oi, Spl	89:11:00	0.18	0	6.57
11As1	HT1	1200	48	QFM + 2	Isothermal	Gl, Oi, Spl	86:14:00	0.34	0	0
16As1	HT1	1180	72	QFM + 2	Isothermal	Gl, Oi, Spl, Aug	77:15:0:8	0.17	0	0
18As1	HT1	1170	72	QFM + 2	Isothermal	Gl, Oi, Spl, Aug	71:13:1:14	0.04	1.12	2.39
13As1	HT1	1160	72	QFM + 2	Isothermal	Gl, Oi, Spl, Aug	66:15:0:19	0.06	0	1.4
5As2	HT2	1160	72	QFM + 2	Isothermal	Gl, Aug, Pl, Mag	75:11:12:2	0.04	0	0
12As2	HT2	1140	72	QFM + 2	Isothermal	Gl, Aug, Pl, Mag, P _{sb}	49:26:20:3:1	0.1	0	0
6As2	HT2	1120	96	QFM + 2	Isothermal	Gl, Aug, Pl, Mag (NA), P _{sb}	-	-	-	-

Table 2 (continued)

Run no	Start. comp	T (°C)	Duration (h)	fO ₂	Method	Phase assemblage	Phase proportions ^a	Σr ²	ΔFe (%)	ΔNa (%)
H201	HT32	1160–1060	225	QFM + 2	1 °C/h	Gl, Aug, Pl, Mag, Psb, Wht (tr)	43:31:27:4:4:tr	0	0	0
H207	HT32	1160–1040	240	QFM + 2	1 °C/h	Si-Gl (NA), Fe-Gl (NA), Aug, Pl, Mag, Psb, Trd (tr), Wht (tr)	–	–	–	–
Low-Ti experiments										
02Bs1	LT1	1320	24	QFM	Isothermal	Gl, Ol	96:04:00	1.17	0	4.59
01Bs1	LT1	1302	24	QFM	isothermal	Gl, Ol	94:06:00	0.39	0	0
04Bs1	LT1	1280	48	QFM	Isothermal	Gl, Ol, Spl	92:08:00	0.49	0	0
03Bs1	LT1	1260	48	QFM	Isothermal	Gl, Ol, Spl	88:11:00	0.13	0.46	8.01
05Bs1	LT1	1240	48	QFM	Isothermal	Gl, Ol, Spl, Aug	86:14:0:0	0.25	0	0
06Bs1	LT1	1200	48	QFM	Isothermal	Gl, Ol, Spl, Aug	83:17:0:0	0.09	1.64	0
15Bs1	LT1	1180	72	QFM	Isothermal	Gl, Ol, Spl, Aug	74:17:0:9	0.93	0	0
17Bs1	LT1	1170	72	QFM	Isothermal	Gl, Ol, Aug, Pl	50:21:16:13	0.03	2.48	5.73
14Bs1	LT1	1160	72	QFM	Isothermal	Gl, Ol, Aug, Pl	42:21:19:18	0.03	1.89	0
8Bs2	LT2	1160	72	QFM	Isothermal	Gl, Aug, Pl	78:11:11	0.45	0	8.06
11Bs2	LT2	1140	72	QFM	Isothermal	Gl, Aug, Pl, Pig	65:18:18:0	1.56	0	0.48
9Bs2	LT2	1120	72	QFM	Isothermal	Gl (NA), Aug, Pl, Mag (NA), Pig	–	–	–	–
L105	LT31	1160–1080	176	QFM	1 °C/h	Gl, Aug, Pl, Mag	60:20:17:3	1.28	2.78	0
L103	LT31	1160–1060	225	QFM	1 °C/h	Gl, Aug, Pl, Mag, Pig, Wht (tr)	47:27:22:4:0:tr	0.22	0	5.7
L106	LT31	1160–1040	240	QFM	1 °C/h	Si-Gl, Fe-Gl, Aug, Pl, Mag, Wht (tr)	36:0:31:27:6:tr	0.9	0	3.99
12Bs1	LT1	1320	24	QFM + 2	Isothermal	Gl	100	3.59	0	4.39
07Bs1	LT1	1302	24	QFM + 2	Isothermal	Gl, Ol	99:01:00	0.92	0	0
08Bs1	LT1	1280	48	QFM + 2	Isothermal	Gl, Ol, Spl	96:04:00	0.39	0	0
09Bs1	LT1	1260	48	QFM + 2	Isothermal	Gl, Ol, Spl	94:06:00	0.22	0.9	0
10Bs1	LT1	1240	48	QFM + 2	Isothermal	Gl, Ol, Spl	91:09:00	0.23	0	0
11Bs1	LT1	1200	48	QFM + 2	Isothermal	Gl, Ol, Spl	87:12:00	0.09	0	0
16Bs1	LT1	1180	72	QFM + 2	Isothermal	Gl, Ol, Spl, Aug, Pl	69:13:0:13:5	0.11	0	0
18Bs1	LT1	1170	72	QFM + 2	Isothermal	Gl, Ol, Spl, Aug, Pl	44:14:0:24:17	0.07	9.09	0
13Bs1	LT1	1160	72	QFM + 2	Isothermal	Gl, Ol, Spl, Aug, Pl	33:14:1:29:23	0.06	6.39	4.77
5Bs2	LT2	1160	72	QFM + 2	Isothermal	Gl, Aug, Pl, Mag	70:15:14:1	0.01	0	1.56
12Bs2	LT2	1140	72	QFM + 2	Isothermal	Gl, Aug, Pl, Mag, Psb	51:26:21:2:0	0.27	0	0.18
6Bs2	LT2	1120	96	QFM + 2	Isothermal	Gl, Aug, Pl, Mag, Opx	36:31:29:4:0	0.5	0	2.11
L201	LT32	1160–1060	225	QFM + 2	1 °C/h	Gl, Aug, Pl, Mag, Trd (tr), Opx	56:21:18:5:tr:0	0.5	0	8.09
L207	LT32	1160–1040	240	QFM + 2	1 °C/h	Si-Gl (NA), Fe-Gl (NA), Aug, Pl, Mag, Trd (tr), Wht (tr)	–	–	–	–

Table 2 (continued)

Run no	Start. comp	T (°C)	Duration (h)	fO ₂	Method	Phase assemblage	Phase proportions ^a	Σr ²	ΔFe (%)	ΔNa (%)
Additional experiments for testing immiscibility (step 4)										
H205	HT32	1160–1080	176	QFM	1 °C/h	Gl, Aug, Pl, Ilm	59:21:20:1	1.64	0	20.4
H203	HT32	1160–1060	225	QFM	1 °C/h	Gl, Aug, Pl, Ilm, Mag, Opx, Wht (tr)	45:26:25:3:0:0:tr	0.92	0	20.56
H206	HT32	1160–1040	240	QFM	1 °C/h	Si-Gl, Fe-Gl, Aug, Pl, Ilm, Mag, Wht (tr)	35:0:32:28:5:0:tr	0.35	0	18.4
H101	HT31	1160–1060	225	QFM+2	1 °C/h	Si-Gl, Fe-Gl (tr), Aug, Pl, Mag, Psb, Wht (tr)	27:tr:31:27:7:tr	0.05–	0.27	34.65
H107	HT31	1160–1040	240	QFM+2	1 °C/h	Si-Gl (NA), Fe-Gl (NA), Aug, Pl, Mag, Psb, Trd (tr), Wht (tr)	–	–	–	–
L205	LT32	1160–1080	176	QFM	1 °C/h	Gl, Aug, Pl	87:13:00	5.55	0	7.37
L203	LT32	1160–1060	225	QFM	1 °C/h	Gl, Aug, Pl, Pig, Wht (tr)	66:18:16:0:tr	2.86	0	1.72
L206	LT32	1160–1040	240	QFM	1 °C/h	Si-Gl, Fe-Gl, Aug, Pl, Mag, Pig, Wht (tr)	55:0:24:20:2:tr	0.3	5.89	0
L101	LT31	1160–1060	225	QFM+2	1 °C/h	Si-Gl, Fe-Gl (tr), Aug, Pl, Mag, Psb, Trd (tr)	42:tr:25:23:10:0:tr	0.28	0	24.25
L107	LT31	1160–1040	240	QFM+2	1 °C/h	Si-Gl (NA), Fe-Gl (NA), Aug, Pl, Mag, Psb, Trd, Wht (tr)	–	–	–	–
Additional experiments for testing equilibrium										
H108	HT31	1060	192	QFM+2	Isothermal	Gl (NA), Aug (NA), Pl (NA), Mag (NA)	–	–	–	–
H208	HT32	1060	192	QFM+2	Isothermal	Gl (NA), Aug (NA), Pl (NA), Mag (NA)	–	–	–	–
L108	LT31	1060	192	QFM+2	Isothermal	Gl (NA), Aug (NA), Pl (NA), Mag (NA)	–	–	–	–
L208	LT32	1060	192	QFM+2	Isothermal	Gl (NA), Aug (NA), Pl (NA), Mag (NA), Psb (NA)	–	–	–	–

^aNA indicates phases are too small to analysis. tr indicates trace amount

230 15 kV accelerating voltage and beam currents of 10 nA
 231 for glasses and 20 nA for minerals. Glasses were analyzed
 232 with a defocused beam (5–10 μm diameter) and miner-
 233 als with a focused beam (~ 1 μm diameter). We measured
 234 Si, Ti, Al, Fe, Mn, Mg, Ca, Na, K, P, and Cr concentra-
 235 tions with on-peak and background counting times of 10
 236 and 5 s, respectively. Na, Al, K, and Cr were analyzed
 237 first to minimize migration. Natural and synthetic pri-
 238 mary standards (reported in the Appendix) were used,
 239 and the CITZAF online data correction package was used
 240 for matrix correction of all analyses (Armstrong 1995).
 241 To ensure accuracy and precision as well as consistency
 242 between analytical sessions, we measured the following
 243 secondary standards at regular intervals (every 80–100
 244 points): VG-2 (NMNH 111240-52) for glass, San Carlos
 245 olivine (NMNH 111312-44) for olivine, kakanui augite
 246 for pyroxene (NMNH 122142), County Lake plagioclase
 247 (NMNH 115900) for plagioclase, ilmenite (NMNH 96189)
 248 for Fe–Ti oxides. Experimental phase compositions gener-
 249 ally have ~98% accuracy for major elements. The complete
 250 dataset of glass and mineral analyses and primary and sec-
 251 ondary standards is reported in the Appendix.

252 **Attainment of equilibrium**

253 In isothermal experiments (steps 1 and 2), we gauged the
 254 attainment of equilibrium mainly based on compositional
 255 and textural characteristics. Furthermore, the run durations
 256 presented herein are considerably longer than those in pre-
 257 vious experimental studies (e.g., Toplis and Carroll 1995;
 258 Charlier and Grove 2012). Melts and minerals in experi-
 259 mental charges show homogeneous backscattered electron
 260 intensities (Fig. 2), except pyroxenes, which are known
 261 to show apparent sector zoning with prominent Al₂O₃
 262 concentrations (Neave and Putirka 2017). Euhedral crystal
 263 morphologies indicate that equilibrium was attained.
 264 Fe–Mg partition coefficients between olivine and melt
 265 (K_D^{Fe-Mg} see Compositional details section) in our experi-
 266 ments are $\sim 0.30 \pm 0.02$, in good agreement with previously
 267 reported values for similar compositions (0.30 ± 0.03 ; e.g.,
 268 Toplis and Carroll 1995), further indicating that our experi-
 269 ments attained thermodynamic equilibrium.

270 To assess the degree of equilibrium attained in steps
 271 3 and 4 cooling experiments, we compared those experi-
 272 ments with isothermal experiments performed at 1060 °C
 273 and QFM + 2. The cooling experiments produced larger
 274 crystals but had identical phase assemblages (melt, augite,
 275 plagioclase, and magnetite) as the isothermal experiments
 276 (Fig. S1). This confirms that our 1 °C/h cooling protocol
 277 did not hinder the attainment of equilibrium. Nevertheless,
 278 we report only the compositions of crystal rims in direct
 279 contact with melt.

Phase proportion estimates

280 Modal phase proportions were estimated using a non-nega-
 281 tive least squares regression algorithm implemented using
 282 the SciPy module in Python ([https://github.com/eazzon/
 283 MassBalanceCal](https://github.com/eazzon/MassBalanceCal)), in which we balanced the starting compo-
 284 sitions with the average compositions of all analyzed phases
 285 in each experimental charge. The squared sum of the residu-
 286 als ($\sum r^2$) was typically <0.5. A few experiments, mainly
 287 cooling experiments with zoned pyroxenes and/or zoned pla-
 288 gioclase resulted in $\sum r^2 > 1$. Fe loss (ΔFe) was calculated
 289 by adding a pure Fe phase in the mass balance calculation
 290 (Li et al. 2020). Our results show a maximum 13.1% Fe loss
 291 in only one run; minor to no Fe loss was detected in the rest
 292 of the experiments. We used a similar procedure to evaluate
 293 potential Na loss due to volatility in the atmosphere of the
 294 furnace (Tormey et al. 1987; Table 2). Calculated values
 295 were generally low (<5–10%) but reached 35% in some
 296 low-temperature, near-solidus experiments. However, such
 297 Na loss estimations are strongly affected by the accuracy of
 298 microprobe Na analyse due to potential Na migration under
 299 the electron beam (Voigt et al. 2017), particularly in low-
 300 temperature experiments, where both residual melt pools
 301 and plagioclase crystals are small (<2–3 μm) inhibiting
 302 analyses with large beam diameters. Such cases affect both
 303 least square calculations and measurement of elements in
 304 plagioclase. Consequently, Na losses reported in Table 2, A01
 305 and especially the high values, should be taken with caution.

Experimental results

Phase equilibria and immiscibility textures

307 Figure 3 shows phase relations in the liquid lines of descent
 308 experiments (steps 1–3) and step 4 experiments as a function
 309 of temperature and fO_2 (QFM and QFM + 2). In both high-Ti
 310 and low-Ti liquid lines of descent experiments (steps 1–3),
 311 solid phases crystallizing upon cooling are olivine, Cr-spi-
 312 nel, augite, plagioclase, trace of low-Ca pyroxene (pigeonite
 313 and/or orthopyroxene), magnetite, ilmenite, pseudobrookite,
 314 whitlockite, and tridymite.

315 Olivine is the first liquidus phase in the primitive com-
 316 positions HT1 and LT1 (1320–1160 °C). Cr-spinel joined
 317 the liquidus at around 1280–1260 °C, followed by the crys-
 318 tallization of augite at around 1240–1180 °C, and plagioc-
 319 lase at around 1180–1160 °C. These temperature ranges
 320 depend on both, the starting composition [low-Ti (LT1) vs
 321 high-Ti (HT1)] and fO_2 (QFM vs QFM + 2). Compared to
 322 experiments at QFM, olivine and augite saturation is delayed
 323 by ~20–40 °C at QFM + 2 in both high-Ti and low-Ti com-
 324 positions. Plagioclase saturation is weakly dependent on fO_2
 325 (1180 °C at QFM + 2 vs 1170 °C at QFM), but plagioclase in
 326
 327

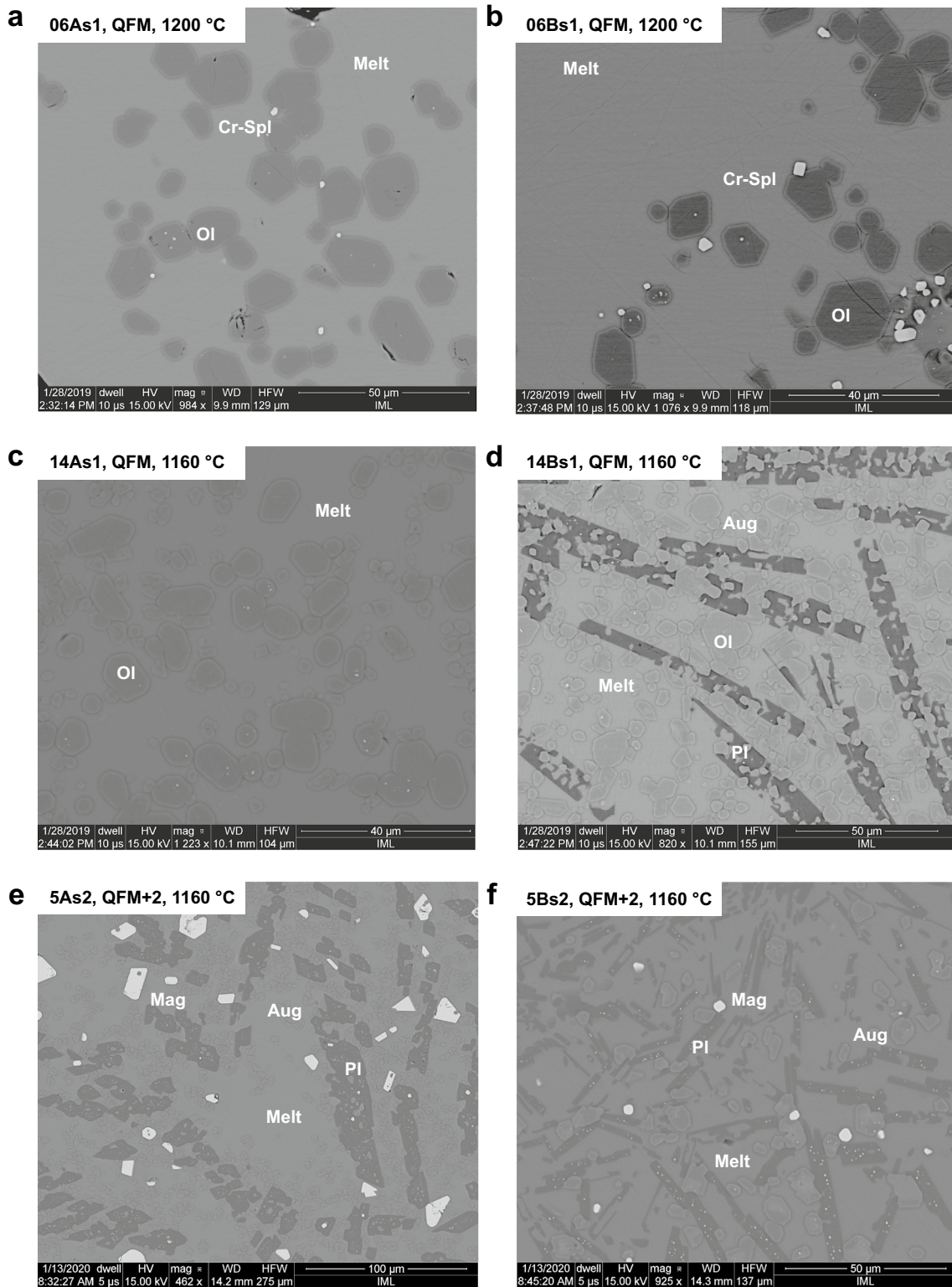


Fig. 2 Back-scattered electron images of selected isothermal experiments. *Ol* olivine; *Cr-Spl* Cr-spinel; *Aug* augite; *Pl* plagioclase; *Mag* magnetite

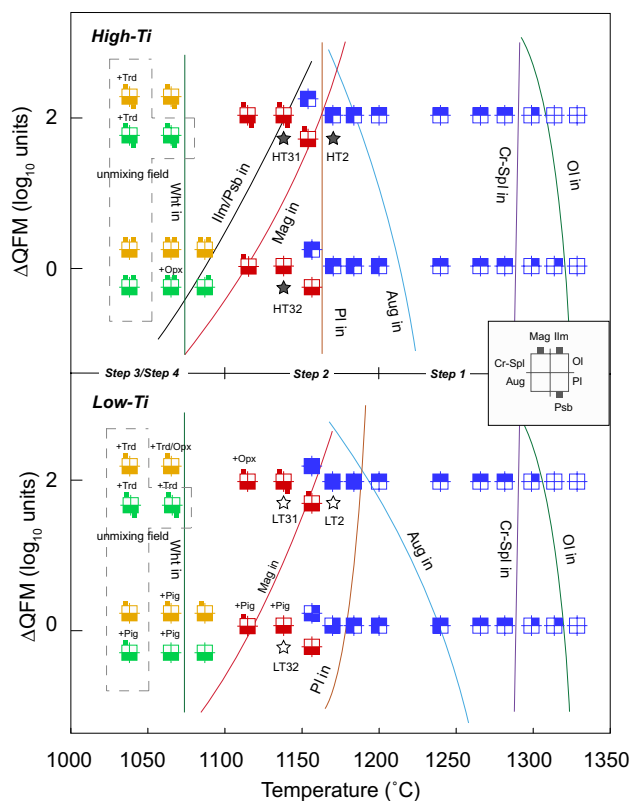


Fig. 3 Phase relations as a function of temperature and fO_2 . Blue, red and orange symbols indicate steps 1–3, respectively, in the liquid lines of descent of high-Ti and low-Ti melts (step 1–3); green symbols indicate step 4 experiments testing the development of immiscibility. Gray dashed lines enclose experiments that produced immiscible melts. *Oliv* olivine; *Cr-Spl* Cr-spinel; *Aug* augite; *Pl* plagioclase; *Opx* orthopyroxene; *Pig* pigeonite; *Mag* magnetite; *Ilm* ilmenite; *Psb* pseudobrookite; *Trid* tridymite; *Whl* whitlockite

the low-Ti composition begins crystallizing 20 °C hotter than in the high-Ti composition. The cotectic proportion of plagioclase relative to mafic minerals (olivine ± clinopyroxene) is ~ 1:3 when olivine is present and changes to 1:1 when olivine disappears. Trace amount of pigeonite and orthopyroxene are observed at low temperature (1120–1040 °C) in low-Ti experiments at QFM and QFM + 2. No low-Ca pyroxene phase is observed in high-Ti experiments during steps 1–3. Fe–Ti oxides join the liquidus after olivine ± augite ± plagioclase ± low-Ca pyroxene. The liquidus temperature of Fe–Ti oxides is strongly related to fO_2 and melt composition. At QFM, magnetite is the first Fe–Ti oxide phase in both high-Ti and low-Ti experiments appearing at ~ 1120 °C. Ilmenite is stable in high-Ti compositions below ~ 1080 °C but does not crystallize in low-Ti experiments. At QFM + 2, the magnetite liquidus in low-Ti and high-Ti experiments increases 1160 °C, followed by pseudobrookite at 1140 °C. The relative stability of Fe–Ti oxides as a function of fO_2 is consistent with observations in natural rocks (Mullen and McCallum 2013). Whitlockite (too small for chemical

analyses) follows Fe–Ti oxides at ~ 1060 °C, regardless of melt composition and fO_2 . Subsequently, tridymite appears in both the high-Ti and low-Ti experiments at 1040 °C under QFM and at 1060 °C under QFM + 2, because the residual melt reaches markedly higher SiO_2 contents (see below).

In step 3 and 4 experiments (Table 1), liquid immiscibility between Fe-rich and Si-rich melts develops at 1040 °C in all compositions at both QFM and QFM + 2 (i.e., compositions HT31, HT32, LT31 and LT32). In step 4 experiments at QFM + 2, using starting materials HT31 and LT31, representing residual melts from both series equilibrated at QFM (i.e., runs H101 and L101; Fig. S1, Table 2), immiscibility develops at a slightly higher temperature (1060 °C). This may be due to the high total Fe-contents of the starting compositions and/or the higher $Fe^{3+}/\Sigma Fe$ at QFM + 2 (Hou et al. 2018). In experiments showing silicate liquid immiscibility, we observe sharp interfaces between the conjugate melts (Figs. 4, S1).

In experiments at QFM and QFM + 2 quenched at 1040 °C, Fe-rich melt droplets range in size from nanometers to several micrometers and are widely distributed within the Si-rich melt. In the interstices between plagioclase crystals, aggregates of Fe-rich melt form irregular melt pools. Fe-rich melt droplets wet plagioclase surfaces with relatively high wetting angles, and are attached to augite and Fe–Ti oxides with lower wetting angles. Whitlockite is usually observed around augite and Fe–Ti oxides, or as feathery fringe microstructures around grain boundaries, as described by Honour et al. (2019a, b). Experiments at QFM + 2 and quenched at 1060 °C (H101 and L101; Table 2) show small Fe-rich droplets (< 0.5 μm) concentrated near or attached to plagioclase.

Melt compositions

The experimental melts produced in steps 1–3 range from basaltic to rhyolitic in composition (Fig. S2). Major element concentrations in homogeneous melts, as well as those in immiscible conjugate melts produced in steps 3 and 4 are shown as a function of temperature in Fig. 5. Due to the small size of Fe-rich droplets, we could only confidently measure the compositions of immiscible melt pairs produced at 1040 °C and QFM (steps 3 and 4), and Si-rich melts at 1060 °C and QFM + 2 (step 4).

Upon cooling, the high-Ti and low-Ti melts show similar compositional trends, because they crystallize similar mineral assemblages. Melt MgO contents are linearly related with experimental temperature. Melt Al_2O_3 and P_2O_5 contents are insensitive to experimental fO_2 conditions. Melt CaO and total alkali ($Na_2O + K_2O$) contents are weakly dependent on fO_2 . Melt CaO concentrations at QFM are slightly higher than those at QFM + 2. In both high-Ti and low-Ti experiments, melt CaO content increases steadily

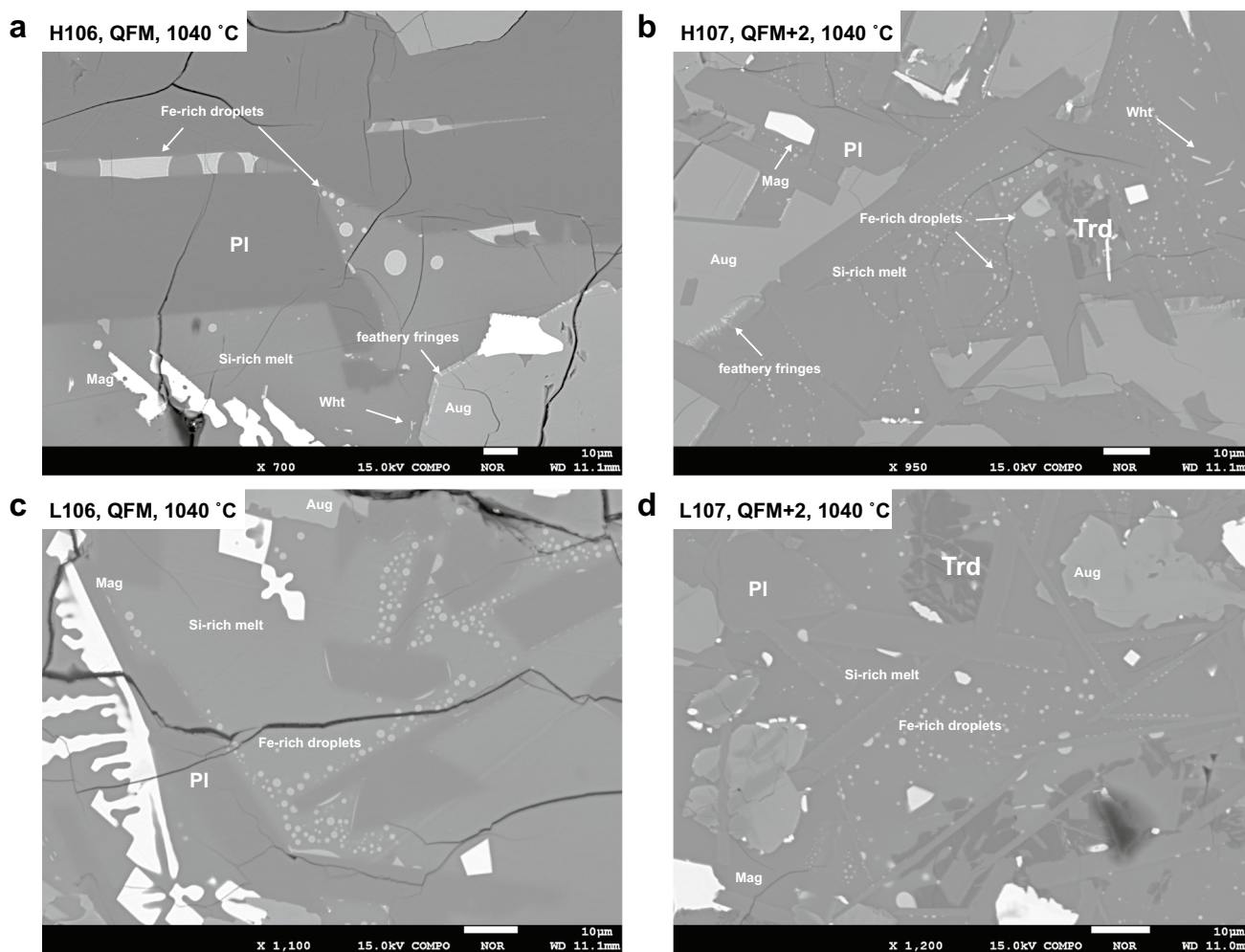


Fig. 4 Back-scattered electron images of selected experimental products showing silicate liquid immiscibility. **a, c** Experiments at QFM showing Fe-rich droplets in Si-rich melt. **b, d** Experiments at

QFM + 2 showing smaller Fe-rich droplets within the Si-rich melt and attached to plagioclase crystals. Abbreviations are as in Fig. 3

399 during early crystallization, reaching a maximum of ~13.5
400 wt% upon augite saturation at ~1200–1240 °C, then decreasing
401 gradually to ~4.0 wt%. Similarly, Al₂O₃ contents in
402 both the high-Ti and low-Ti melts steadily rise to maximum
403 values of 14.1 wt% at 1160 °C and 15.5 wt% at 1180 °C,
404 respectively, corresponding to the initiation of plagioclase
405 crystallization, and then gradually decrease. The total alkali
406 and P₂O₅ concentrations steadily increase with decreasing
407 temperature. Although whitlockite crystallizes at the lowest
408 experimental temperatures, we do not observe depletion of
409 P₂O₅ in the residual melt.

410 Due to its critical control on the stability of Fe–Ti oxides,
411 *f*O₂ has more profound effects on melt TiO₂, FeO, and SiO₂
412 contents. TiO₂ content is strongly controlled by the crys-
413 tallization of ilmenite and pseudobrookite, and less so by
414 magnetite. The TiO₂ content of the high-Ti melt reaches a
415 maximum of 4.4 wt% at 1140 °C at QFM, and 3.6 wt% at
416 1160 °C at QFM + 2, then decreases upon crystallization

of Fe–Ti oxides. Similarly, the TiO₂ content of the low-Ti
417 melts reaches a maximum of 2.6 wt% at 1080 °C at QFM
418 before dropping slightly due to magnetite crystallization and
419 a maximum of 2.0 wt% at 1140 °C at QFM + 2, correspond-
420 ing to the crystallization of pseudobrookite.

421 The FeO content of the melt evolved differently in the
422 high-Ti and low-Ti experimental series. In the high-Ti
423 experiments, the FeO content of the melt at QFM slightly
424 decreases from ~13.0 to ~12.0 wt% from 1320 to 1160 °C,
425 increases to 14.1 wt% at 1140 °C then decreases when
426 magnetite crystallizes at 1120 °C. At QFM + 2, the melt
427 FeO content remains constant (at ~12.5 wt%) from 1320 to
428 1160 °C, does not obviously increase just before magnetite
429 saturation (12.7 wt%), then decrease sharply to <10.0
430 wt% at 1060 °C. In the low-Ti experiments, the FeO con-
431 tent of the melt at QFM decreases from 11.5 to 9.7 wt%
432 from 1320 to 1175 °C, increases to 12.6 wt% at 1140 °C,
433 then decreases when magnetite crystallizes at 1120 °C. At
434

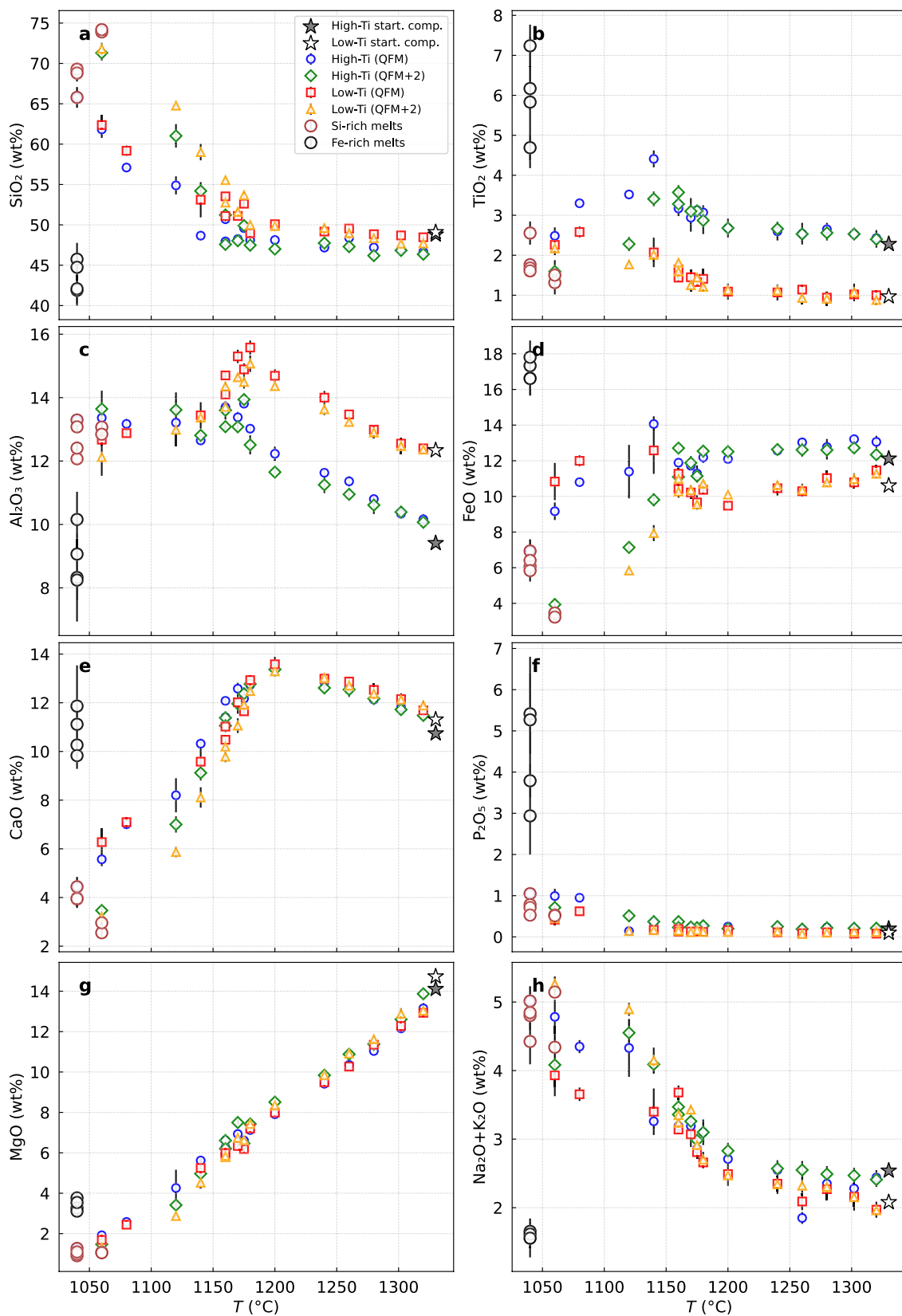


Fig. 5 Binary diagrams of selected major element concentrations in the melt as a function of temperature

435 QFM + 2, the melt FeO content slightly decreases from
436 liquidus to 1160 °C, then decreases sharply the magnetite
437 crystallization.

438 Oxygen fugacity also strongly affects the SiO₂ content in
439 the melt. In both the high-Ti and low-Ti melts, SiO₂ content
440 increases slowly but steadily from 1320 to 1140–1160 °C,
441 then more drastically increases below ~ 1140 °C due to
442 Fe–Ti oxide crystallization. In both series, the SiO₂ contents
443 of the melt at QFM + 2 increase more steeply with decreasing
444 temperature than those at QFM, probably due to higher
445 proportions of Fe–Ti oxide crystallization at more oxidizing
446 conditions and higher bulk crystallinity.

447 The compositions of immiscible melt globules produced
448 at low temperatures (1040–1060 °C in steps 3 and 4) from
449 the high-Ti and low-Ti melts are plotted against melt SiO₂
450 content in Fig. 6, alongside with the homogeneous melts
451 produced at higher temperatures. Major and minor elements
452 show different degrees of enrichment/depletion in the
453 two immiscible melts. Si-rich melts (65.8–74.2 wt% SiO₂,
454 3.2–6.9 wt% FeO) are enriched in Al₂O₃ and total alkalis,
455 whereas their conjugate Fe-rich melts (41.9–45.7 wt% SiO₂,
456 16.6–17.8 wt% FeO) are enriched in TiO₂, CaO, and P₂O₅.

457 By plotting the degree of polymerization of the silicate
458 melts (NBO/T, [the ratio of non-bridging oxygens to
459 tetrahedrally coordinated network-forming cations, i.e.,
460 T = Si + Al + P + Ti; Mysen 1983) against experimental
461 temperature (Fig. 7), we observe that the immiscible melts
462 produced in this study plot near the binodal surface defined by
463 Charlier and Grove (2012), although the binodal apex might
464 be at a slightly higher temperature here (~ 1060 °C at NBO/T
465 ≈ 0.5). Furthermore, the compositions of the last homogeneous
466 melts produced before attaining immiscibility in both the
467 high-Ti and low-Ti series are compositionally similar to the
468 first (i.e. highest-temperature) immiscible Si-rich melts. This
469 is because both compositional trends intersect the binodal
470 nearly tangentially and towards low NBO/T values as previously
471 reported in other tholeiitic settings (Charlier et al.
472 2011, 2013; Namur et al. 2011; 2012a).

473 Olivine

474 Olivine compositions are reported in the Appendix and
475 shown in Fig. 8. Olivine is present in experiments performed
476 at 1320–1160 °C (Table 2). The forsterite content of olivine
477 [Fo = molar Mg/(Mg + Fe)] does not clearly differ between
478 the high-Ti and low-Ti series. In the high-Ti experiments, olivine
479 Fo content ranges from Fo₈₇ to Fo₇₉ at QFM and from
480 Fo₈₉ to Fo₈₂ at QFM + 2 (Fig. 8a). This difference is likely
481 due to the higher Fe³⁺/ΣFe of the equilibrium melt. In low-Ti
482 experiments, olivine Fo content varies from Fo₈₉ to Fo₇₈
483 at QFM and from Fo₉₁ to Fo₈₃ at QFM + 2. The relationship
484 between olivine Fo content and the Mg# of the coexisting
485 liquid (= molar Mg/[Mg + Fe²⁺] accounting for ferrous iron

in the liquid; Kress and Carmichael 1991) is comparable for
both high-Ti and low-Ti compositions (Fig. 8b). Olivine-melts
K_D^{Fe–Mg} values range between 0.26 and 0.33, consistent
with previous studies (Roeder and Emslie 1970; Toplis
2005; Blundy et al. 2020).

Pyroxene

The high-Ti and low-Ti compositions crystallize augite,
and trace amounts of low-Ca pyroxene, orthopyroxene
and pigeonite (Fig. 9a). Augite compositional ranges are
similar in both magma series: Wo_{37–45}En_{40–51}Fs_{4–20} and
Wo_{41–47}En_{40–50}Fs_{6–13} at QFM and QFM + 2, respectively,
in high-Ti experiments, and Wo_{28–43}En_{38–52}Fs_{6–30} and
Wo_{38–45}En_{44–51}Fs_{5–12} at QFM and QFM + 2, respectively, in
low-Ti experiments (Fig. 9a). We calculated pyroxene ferric
and ferrous iron contents using the stoichiometric method
of Lindsley (1983) and then pyroxene Mg# accordingly.
Pyroxene Mg# is positively correlated with temperature,
and pyroxenes at QFM + 2 have generally higher Mg# than
those at QFM (Fig. 9b). Pyroxene-melt K_D^{Fe–Mg} values are
0.27 ± 0.08 in both the high-Ti and low-Ti compositions
(Fig. 9c), consistent with those reported by Toplis and Carroll
(1995) for tholeiitic basalts.

Plagioclase

Plagioclase compositions are reported in the Appendix
and shown in Fig. 10. Plagioclase compositions (An content = molar Ca/[Ca + Na]) range from An₇₂ to An₅₈ at QFM
and from An₆₇ to An₅₇ at QFM + 2 in high-Ti experiments,
and from An₇₈ to An₇₀ at QFM and from An₇₆ to An₇₃ at
QFM + 2 in low-Ti experiments. In low-temperature experiments
(< 1080 °C), plagioclase crystals are typically small
(< 2–3 μm) and locally zoned, inhibiting accurate measurement
of their rim composition; therefore, the reported An contents
for those experiments are probably slightly elevated because they
include both core and rim compositions. Plagioclase in low-Ti
experiments start crystallizing earlier (20 °C hotter) than in
high-Ti experiments and from a melt with higher CaO- and lower
Na₂O-contents (i.e., higher Ca/Na ratio; Fig. 10b). The An content
at the plagioclase liquidus is also higher in low-Ti experiments.

Cr-spinel and Fe–Ti oxides

Cr-spinel and Fe–Ti oxide compositions are reported in the
Appendix. Cr-spinel Cr# (= molar Cr/[Cr + Al]) ranges from
0.70 to 0.67 at QFM and from 0.68 to 0.53 at QFM + 2 in
high-Ti experiments, and from 0.65 to 0.51 at QFM and from
0.59 to 0.45 at QFM + 2 in low-Ti experiments. Cr# is not
correlated with temperature.

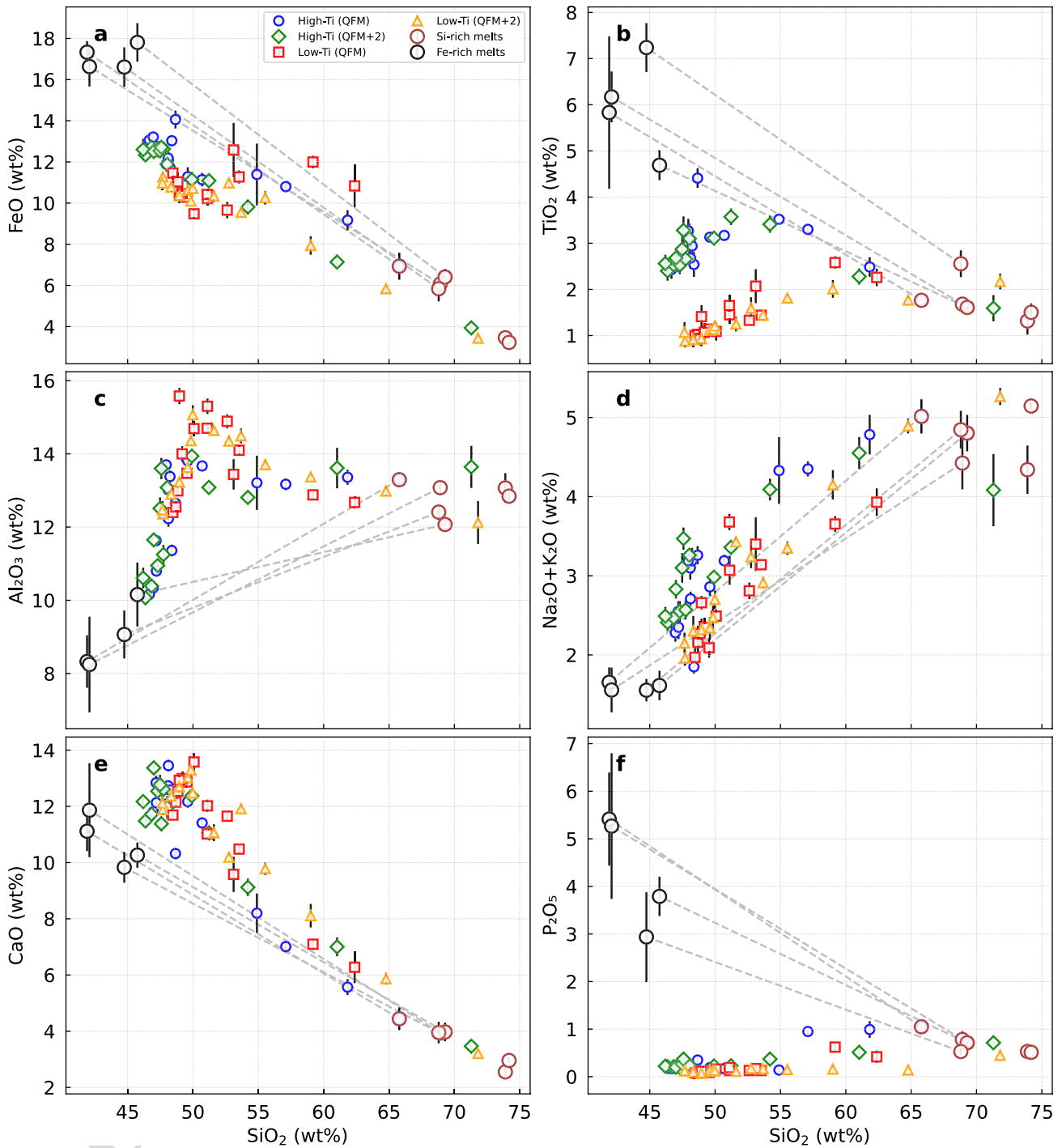


Fig. 6 Major element compositional variations in the melt as a function of melt SiO₂ content (wt%) for high-Ti and low-Ti melts. Immiscible melt pairs are connected by dashed lines

532 Endmember compositions of Fe–Ti oxides (i.e., X_{Mag} ,
 533 X_{Hem} , and X_{Psb} for the magnetite–ulvöspinel here ‘mag-
 534 netite’), hematite–ilmenite (‘ilmenite’), and pseudobrook-
 535 ite–ferropseudobrookite (‘pseudobrookite’) were recal-
 536 culated following the method of Andersen et al. (1993) (Fig.
 537 S3). At a given fO_2 , the recalculated magnetite (X_{Mag}) and

pseudobrookite (X_{Psb}) components are not correlated with
 temperature. Pseudobrookite was only observed in QFM + 2
 experiments, with compositions $X_{Psb} = 0.69–0.66$ and 0.60
 in high-Ti and low-Ti experiments, respectively. Magnetite
 has compositions of $X_{Mag} = 0.90–0.83$ and $0.92–0.79$ in
 high-Ti and low-Ti experiments at QFM + 2, respectively,

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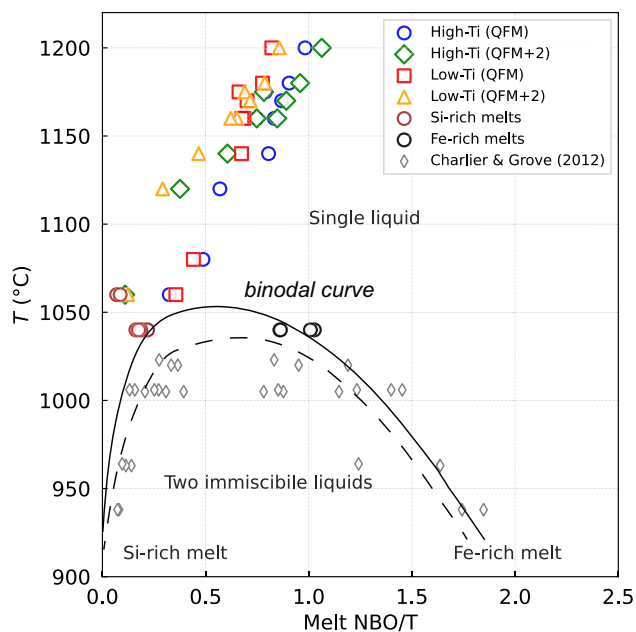


Fig. 7 The degree of melt polymerization (expressed as NBO/T, the ratio of non-bridging oxygens to tetrahedrally coordinated cations, $T = \text{Si} + \text{Al} + \text{P} + \text{Ti}$) versus temperature (< 1200 °C). Immiscible melts from Charlier and Grove (2012) are shown for comparison. The solid and dashed curve are the estimated binodal curve on which our experiments lie and the binodal curve from Charlier and Grove (2012), respectively

544 whereas at QFM, but lower respective compositions of
 545 $X_{\text{Mag}} = 0.50\text{--}0.43$ and $0.54\text{--}0.48$ at QFM. Ilmenite was only
 546 observed in high-Ti experiments at QFM, with compositions
 547 of $X_{\text{Hem}} = 0.25\text{--}0.19$ that correlate with temperature. Both
 548 MgO content in ilmenite and Al_2O_3 content in magnetite
 549 decrease with decreasing temperature.

Discussion

Effect of primary magma composition on phase relations and differentiation

As expected, due to their different compositions, our high-Ti and low-Ti melts cooled to crystallize plagioclase and Fe–Ti oxides with different compositions and liquidus temperatures. Nonetheless, both melts have similar MgO and Cr_2O_3 contents and bulk Ca/Al ratios, and thus saturate clinopyroxene of similar compositions at similar liquidus temperatures. The most obvious difference between the high-Ti and low-Ti compositions is the timing of plagioclase saturation. The Ca- and Al-enriched low-Ti melt is closer to the forsterite-anorthite cotectic in the simplified CMAS system than the Na- and Fe-enriched high-Ti melt (Fig. S4; Presnall et al. 1978; Neave et al. 2019). The former meets the cotectic at a lower diopside content and further away from the eutectic point, and therefore, saturates plagioclase at a higher temperature (Fig. S4). Plagioclase An content is controlled by the melt $\text{Ca}\# (= \text{Ca}/[\text{Na} + \text{Ca}])$ and $\text{Al}\# (= \text{Al}/[\text{Al} + \text{Si}])$ and Ca/Al ratio (Panjasawatwong et al. 1995; Namur et al. 2012b; Neave and Namur 2022,) which are all higher in primitive low-Ti melts than in equivalent high-Ti melts. Consequently, low-Ti melts tend to crystallize plagioclase with higher An content than high-Ti melts (Fig. 10).

The distinct melt compositions also play a role in the saturation of Fe–Ti oxides, although this effect is less pronounced than that of $f\text{O}_2$. Although our high-Ti and low-Ti melts have indistinguishable magnetite liquidus temperatures at a given $f\text{O}_2$ condition (1120 °C at QFM, 1160 °C at QFM + 2), both are significantly higher than the magnetite liquidus in experiments of Toplis and Carroll (1995; 1096 °C

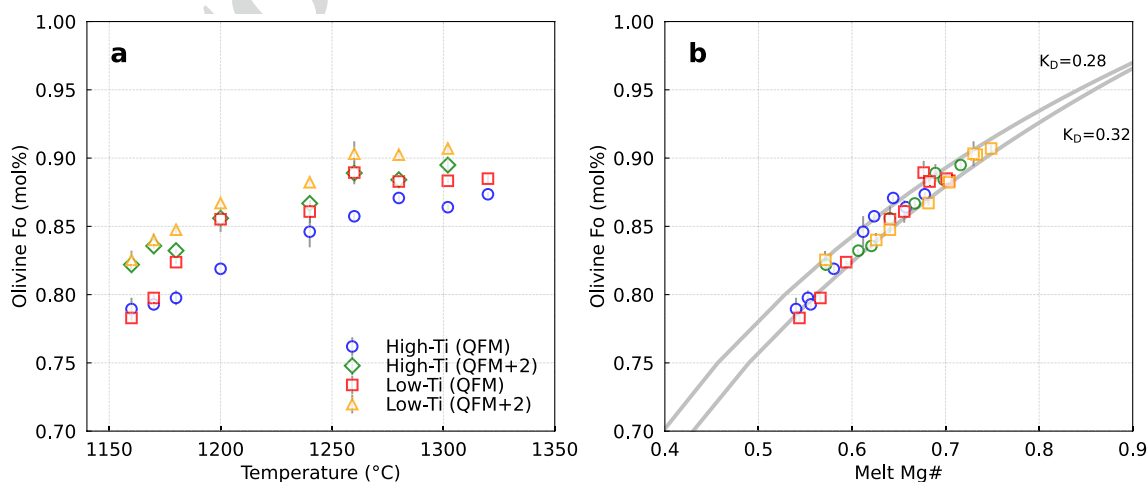


Fig. 8 Olivine forsterite contents as a function of **a** temperature and **b** Mg# of the coexisting melt. The Fe^{2+} content in the melt was calculated following Kress and Carmichael (1991). The grey curves in **b** indicate olivine-melt $K_D^{\text{Fe-Mg}} = 0.30 \pm 0.02$

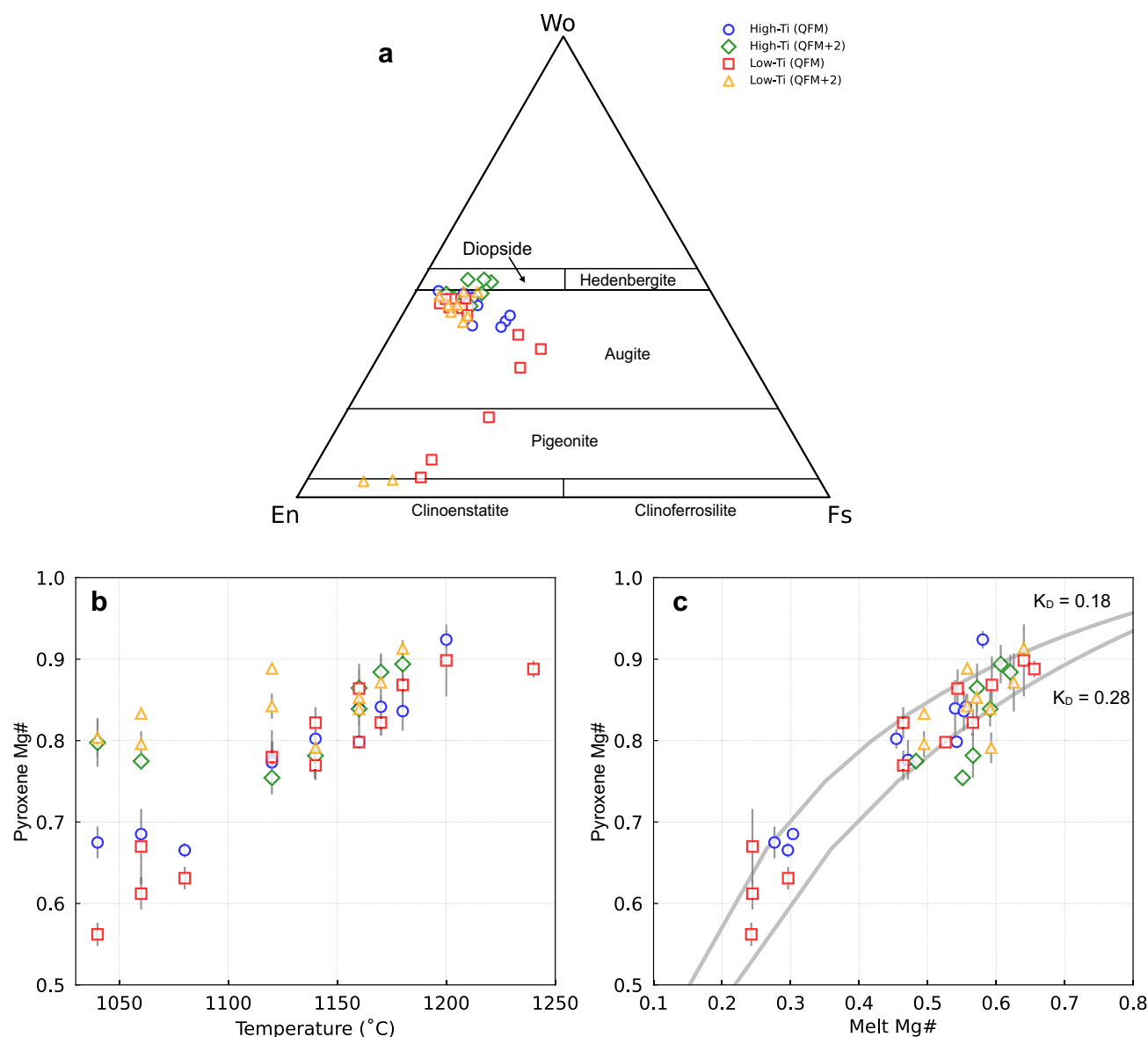


Fig. 9 Experimental pyroxene compositions. **a** Wollastonite (Wo)-enstatite (En)-ferrosilite (Fs) ternary diagram showing the compositions of experimental pyroxenes. **b** Pyroxene Mg# variations as a

function of temperature. **c** Pyroxene Fe/Mg equilibrium diagram; gray curves indicate the equilibrium field from Toplis and Carroll (1995; i.e., pyroxene-melt $K_D^{Fe-Mg} = 0.23 \pm 0.05$)

581 at QFM) and Juster et al. (1989; 1135 °C at QFM + 2.9;
 582 1120 °C at QFM + 1.4). Similarly, two compositions in the
 583 experiments of Snyder et al. (1993) have different magnetite
 584 liquid: 1130 °C at QFM in a relatively FeO and TiO₂-rich
 585 composition (17.3 wt% FeO, 3.8 wt% TiO₂) and 1109 °C
 586 at QFM in a relatively FeO and TiO₂-depleted composition
 587 (14.3 wt% FeO, 2.3 wt% TiO₂). Toplis et al. (1994) proposed
 588 that the progressive addition of phosphorus may destabilize
 589 magnetite under oxidizing conditions; however, is inconsis-
 590 tent with our experiments and those of Snyder et al. (1993),
 591 perhaps implying that the magnetite liquidus is controlled
 592 by a complex dependence on melt FeO, TiO₂, P₂O₅ and SiO₂

593 contents and *f*O₂ (Ariskin and Barmina 1999). The ilmenite
 594 crystallization is also affected by the melt TiO₂ content
 595 (Toplis and Carroll 1995); indeed, ilmenite did not saturate
 596 in the low-Ti melts, which only crystallized pseudobrookite
 597 at QFM + 2.

598 The most representative feature of tholeiitic magmas
 599 prior to the saturation of Fe–Ti oxides is the progressive
 600 enrichment of FeO in the evolved melt (less than ~6 wt%
 601 MgO; e.g., Grove and Bryan 1983; Toplis and Carroll 1995;
 602 Juster et al. 1989; Snyder et al. 1993). However, compared
 603 to the experiments of Toplis and Carroll (1995) and Snyder
 604 et al. (1993), our high-Ti and low-Ti melts do not reach high

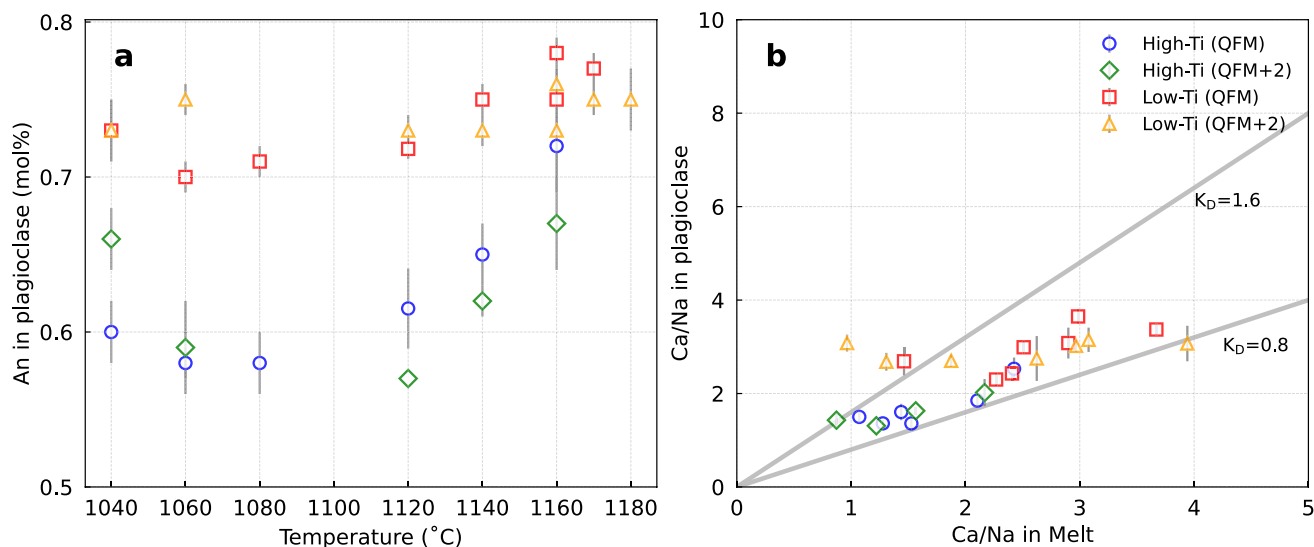


Fig. 10 **a** Experimental plagioclase An content variations as a function of temperature. **b** Ca/Na exchange between plagioclase and melt; gray lines indicate the equilibrium field from Toplis and Carroll (1995; i.e., plagioclase-melt $K_D^{\text{Ca-Na}} = 0.8\text{--}1.6$)

605 FeO enrichments (up to ~18 wt%; Toplis and Carroll 1995)
 606 at similar fO_2 . This is because we saturate Fe–Ti oxides at
 607 much higher temperatures, and likely also due to the sig-
 608 nificantly higher FeO concentrations in their starting com-
 609 positions and the higher cotectic proportions of plagioclase
 610 in their cumulates during crystallization. For instance, our
 611 starting compositions for step 2 contain ~6.5 wt% MgO and
 612 only ~10 wt% FeO, concentrations that were generated by
 613 the protracted fractionation of mafic olivine \pm clinopyroxene,
 614 compared to 13–15 wt% FeO at similar MgO concentrations
 615 in the bulk compositions of Toplis and Carroll (1995) and
 616 Snyder et al. (1993). Furthermore, the cotectic proportion of
 617 plagioclase relative to mafic minerals (olivine \pm clinopyrox-
 618 ene) in the experiments of Toplis and Carroll (1995) is ~3:1
 619 when only olivine and plagioclase are present, but progres-
 620 sively changes to 2:1 when clinopyroxene crystallizes, and
 621 the proportion is roughly 2:1 in the experiments of Snyder
 622 et al. (1993). In contrast, our high-Ti and low-Ti melts crys-
 623 tallize plagioclase and augite in the proportion 1:1.

624 Beside FeO enrichment, TiO_2 enrichment in tholeiitic
 625 melts is also observed before Fe–Ti oxides crystallized.
 626 Because Ti is relatively incompatible in tholeiitic melts, the
 627 maximum TiO_2 enrichment in the melt is related to the TiO_2
 628 content of the parental melt and the liquidus temperatures of
 629 Fe–Ti oxides. This is illustrated by the fact that high-Ti melts
 630 have higher TiO_2 concentration than low-Ti melts before
 631 Fe–Ti oxide saturation (i.e., ilmenite and pseudobrookite).
 632 We note that no significant TiO_2 depletion is observed in our
 633 low-Ti melts at QFM due to the absence of ilmenite crystal-
 634 lization. This supports the notion that a low-Ti composition
 635 could evolve to a high-Ti composition if Fe–Ti oxide satura-
 636 tion is delayed late during differentiation (Hou et al. 2011).

Effect of fO_2 on phase relations and differentiation path

637
 638
 639 Changing fO_2 significantly affects Fe–Ti oxide phase rela-
 640 tions but has only minor effects on silicates. Consistent
 641 with previous experiments showing that increasing fO_2
 642 expands the stability field of magnetite (Toplis and Carroll
 643 1995; Snyder et al. 1993; Ariskin and Barmina 1999), the
 644 magnetite liquidus in our experiments increases by ~40 °C
 645 between QFM and QFM + 2 in both the high-Ti and low-
 646 Ti compositions. Increasing fO_2 also increases X_{Mag}
 647 component of magnetite and causes Al_2O_3 -bearing pseudo-
 648 brookite to form rather than ilmenite. These changes in
 649 the stability of Fe–Ti oxides at different fO_2 will thus have
 650 significant effect on melt FeO and SiO_2 contents. Figure 11
 651 shows the melt FeO/MgO ratio as a function of SiO_2 con-
 652 tent. Because the high-Ti and low-Ti melts follow a simi-
 653 lar differentiation trend in terms of MgO concentration
 654 (i.e., $dMgO/dT \approx 0.04$ wt%/°C), the observed change of
 655 FeO/MgO with SiO_2 content in our experimental melts
 656 is directly related to the degree of FeO depletion under
 657 different fO_2 conditions. Both high-Ti and low-Ti melts
 658 follow a tholeiitic trend at QFM, but a calc-alkaline trend
 659 at QFM + 2 due to the early crystallization of high X_{Mag}
 660 magnetite. We note, however, that adding water in natural
 661 systems may have a similar effect by delaying plagioclase
 662 crystallization (Mandler et al. 2014; Almeev et al. 2012).
 663 For comparison, experiments on FeO-rich melts coexisting
 664 only with low X_{Mag} magnetite at QFM-1 to QFM + 1 (Top-
 665 lis and Carroll 1995) evolve uniquely within the tholeiitic
 666 field (Fig. 11).

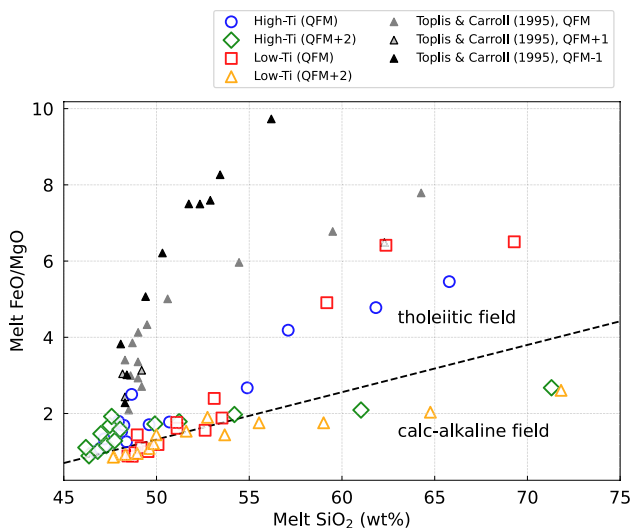


Fig. 11 The relation between the FeO/MgO weight ratio in the melt and melt SiO₂ content. The dashed line indicates the boundary between the calc-alkaline and tholeiitic fields after Miyashiro (1974). Experimental melts (*f*O₂: QFM-1 to QFM+1) from Toplis and Carroll (1995) are shown for comparison

Comparison between low-pressure liquid lines of descent and natural lavas

Here, we use our experimental data to discuss the low-pressure differentiation of natural lavas in LIPs. We choose the Emeishan LIP as an example due to the considerable amount of data on those natural lavas and because our starting materials were selected based on them. However, we note that Emeishan LIP magmas may have differentiated at higher pressure (Tao et al. 2015) and that the system may also have been slightly hydrous (Liu et al. 2017).

Figure 12 compares our experimental liquid lines of descent (LLDs) with a compilation of Emeishan high-Ti and low-Ti lavas. Overall, our high-Ti LLDs only partly overlap with observed high-Ti lavas, whereas our low-Ti LLDs match the low-Ti lavas relatively well in terms of TiO₂, Al₂O₃, and SiO₂. This mismatch with the natural lavas may be related to the differences in differentiation conditions (pressure, water content, *f*O₂), source heterogeneities and perhaps mineral accumulations (i.e., pyroxene, plagioclase and Fe–Ti oxide) in the natural rocks; therefore, the lavas may not represent true liquid compositions. The CaO contents and CaO/Al₂O₃ ratios of natural lavas are often lower than those of our experimental melts, likely reflecting the pressure effect on the stability of clinopyroxene: high-pressure conditions favor the crystallization of clinopyroxene, decreasing the melt CaO content (Villiger et al. 2004, 2007). Alternatively, these low-CaO lavas could also be explained by mixing between the primitive and evolved. Furthermore, natural lavas often have higher FeO contents than our

experimental melts (< 7 wt% MgO), although this is probably not caused by the early crystallization of plagioclase (see above discussion above), because high pressure or slightly hydrous conditions, that may be realistic for the Emeishan LIP (Tao et al. 2015; Liu et al. 2017; Ganino et al. 2008), would suppress the plagioclase crystallization and delay or inhibit the enrichment of the melt FeO. Thus, the observed difference between the FeO contents of the natural lavas with our experimental melts could result from three processes. (1) The natural lavas may have evolved from a primitive melt with a higher FeO content than our selected compositions; indeed, the compiled database exhibits a broad range of primitive FeO contents (Fig. 1c). (2) The lavas may have evolved from a multi-component source. For example, the partial melting of a pyroxenitic source can produce parental melts more enriched in FeO than those produced by the melting of a peridotitic source (Lambart et al. 2013). This is supported by the multivariate trace element and radiogenic isotopic composition of primitive picrites and olivines in the Emeishan LIP (e.g., Kamenetsky et al. 2012); (3) Finally, the accumulation of dense magnetite phases that crystallized at high *f*O₂ conditions (Howarth et al. 2013) may have increased the lavas' FeO contents, consistent with previous estimates of high *f*O₂ conditions for the Emeishan lavas (Cao and Wang 2022; Bai et al. 2019).

Predicting the onset of silicate liquid immiscibility

Liquid immiscibility is known to occur during the late-stage fractionation of basalts and has been identified in many layered intrusions related to LIPs (Veksler and Charlier 2015), such as the Skaergaard intrusion (Holness et al. 2011; Humphreys 2011; Jakobsen et al. 2005; McBirney and Nakamura 1974), the Sept Iles intrusion (Charlier et al. 2011; Namur et al. 2012a; Keevil et al. 2020), the Bushveld Complex (Fischer et al. 2016; VanTongeren and Mathez 2012; Yuan et al. 2017) and the Panzhihua intrusion (Zhou et al. 2013; Wang et al. 2018; 2021). It is commonly accepted that unmixing begins at temperatures around 1040 °C (Philpotts 2008; Charlier and Grove 2012; Hou et al. 2018; Honour et al. 2019a, 2019b), although Hou and Veksler (2015) also produced super-liquidus immiscibility at temperature up to 1200 °C. Our and Charlier and Grove's (2012) experiments suggest that immiscibility does not require extreme iron enrichments (> 20 wt%; Veksler 2009) and can even start after the crystallization of magnetite, which depletes FeO in the residual melt. Increased TiO₂, P₂O₅ and alkali contents have also been shown to promote immiscibility by expanding the two liquids field (Charlier and Grove 2012). Increasing *f*O₂ and *a*H₂O may also widen the immiscibility field (Hou et al. 2018; Naslund 1983).

Considering the energy of the entire system, when immiscibility occurs, the total Gibbs free energy of mixing (*G*^{mix})

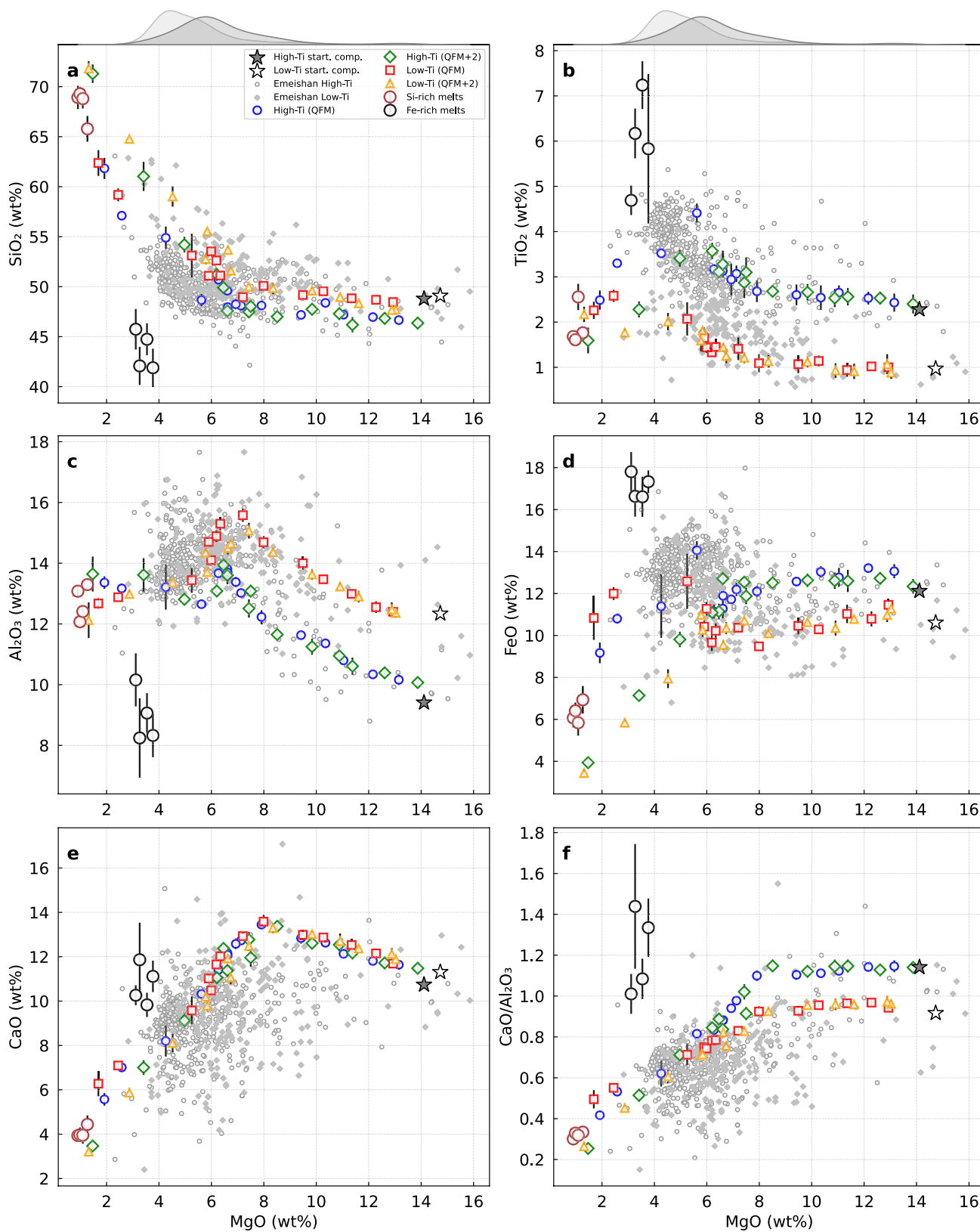


Fig. 12 Harker diagrams showing the evolution of experimental melts obtained in this study compared to natural lavas from the Emeishan LIP (the compiled dataset is listed in the caption to Fig. 1). The histo-

grams atop the uppermost panels show the distribution of MgO contents in natural Emeishan high-Ti and low-Ti lavas

747 for a single liquid prior to unmixing is higher than the sum
 748 of G^{mix} for the immiscible pair, with G^{mix} being identical
 749 for the coexisting immiscible pair (Ghiorso and Carmichael
 750 1980; Ghiorso et al. 1983):

751
$$\Delta G^{\text{mix}} = G^{\text{ideal}} + G^{\text{excess}} \quad (1)$$

 752

753
$$\Delta G^{\text{mix}}_{\text{single liquid}} > \Delta G^{\text{mix}}_{\text{A}} + \Delta G^{\text{mix}}_{\text{B}} \quad (2)$$

 754

755 with A and B representing the immiscible pairs. The first
 756 derivative of the Gibbs free energy of mixing ($\partial G^{\text{mix}}/\partial n_k$,
 757 where n_k is the moles of the component k) with respect
 758 to any component in the silicate liquid will thus change
 759 from negative to positive when unmixing occurs (Fig. 13).
 760 Ghiorso and Carmichael (1980) defined 17 independent
 761 components (see their Table 1) to quantitatively calculate
 762 G^{mix} based on the interaction energies between those compo-
 763 nents in the silicate liquids. Thus $\partial G^{\text{mix}}/\partial n_k$ can be equated
 764 to $RT \ln a_k$ as (see Eq. 18 in Ghiorso and Carmichael 1980):

765
$$\left(\frac{\partial G^{\text{mix}}}{\partial n_k} \right)_{T,P,n \neq n_k} = RT \ln X_k + RT \ln \gamma_k = RT \ln a_k, \quad (3)$$

 766

767 where R is the gas constant ($8.314 \text{ J mol}^{-1} \text{ K}^{-1}$), T is the
 768 experimental temperature in Kelvin and X_k , γ_k and a_k are
 769 the mole fraction, activity coefficient and activity of compo-
 770 nent k . Ghiorso and Carmichael (1980) gave the example
 771 of calculating the first derivative of Gibbs free energy of
 772 mixing with respect to the SiO_2 component ($\partial G^{\text{mix}}/\partial n_{\text{SiO}_2}$;
 773 see their Fig. 2), which can potentially be used to investigate
 774 the development of immiscibility by solving the evolution of
 775 a_{SiO_2} in the melt upon cooling. When immiscibility devel-
 776 ops, $\partial G^{\text{mix}}/\partial n_{\text{SiO}_2}$ of the liquid, i.e., $RT \ln a_{\text{SiO}_2}$ can be used
 777 to define the binodal surface of the immiscible pair (i.e.,
 778 the width of the binodal surface). We tested Ghiorso and
 779 Carmichael's (1980) model, but it did not successfully pre-
 780 dict immiscibility in our experiments or other experiments
 781 where immiscibility was observed. This is probably due to
 782 our still far-from-perfect knowledge of thermodynamics in

783 multi-component silicate melts, precluding the accurate cal-
 784 culation of the activities of their various components.

785 Here, by analogy to the thermodynamic expression of
 786 Ghiorso and Carmichael (1980), we propose a new empiri-
 787 cal parameterization to predict immiscibility in silicate
 788 melts. Previous studies have found that the sum of the
 789 mole percentages of TiO_2 , FeO , MgO , CaO and P_2O_5 may
 790 effectively distinguish Si-rich and Fe-rich melts (Charlier
 791 and Grove 2012; Hou et al 2018; Kamenetsky et al 2013).
 792 We thus use this term instead of the mole fraction of the
 793 SiO_2 component (as in Ghiorso and Carmichael 1980)
 794 to describe the compositional contrast between the two
 795 immiscible melts. We performed a polynomial curve fit
 796 on all the available experiments with immiscible melts
 797 under condition relevant to natural systems. Our best fit-
 798 ting result is:

799
$$RT \ln X_{\text{SiO}_2} = \frac{4.6 \times 10^{-4} \times ((-14736.5 - \phi)^2 - 12323.9 \times (\phi^2))}{2.155}, \quad (4)$$

 800

801 where X_{SiO_2} is the mole percentage of SiO_2 in the liq-
 802 uid and, ϕ is the sum of the mole percentage of oxides
 803 dominantly partitioning into the Fe-rich melt, i.e.
 804 $X_{\text{TiO}_2} + X_{\text{FeO}} + X_{\text{MgO}} + X_{\text{CaO}} + X_{\text{P}_2\text{O}_5}$. The coefficient
 805 of determination (r^2) for this fit is 0.95. The fitted curve
 806 covers the calibration range: $\phi = 6.3\text{--}67.8$ (mol%);
 807 $T = 938\text{--}1053$ °C; P from 1 atm to 100 MPa, f_{O_2} from QFM
 808 -6 to QFM $+3.3$, $0\text{--}2.6$ wt% H_2O in the Si-rich melt and
 809 $0\text{--}1.6$ wt% in the Fe-rich melt.

810 The fitted curve and the experimental data used to
 811 calibrate it are shown in Fig. 14. When only miscible
 812 melts are considered, the melt gradually evolves towards
 813 $\text{TiO}_2 + \text{FeO} + \text{MgO} + \text{CaO} + \text{P}_2\text{O}_5$ depletion in the single-
 814 liquid field, reflecting the effect of the fractionation of
 815 mafic phases on melt differentiation. We further tested our
 816 model on natural immiscible melts hosted in melt inclu-
 817 sions from Siberian Traps (Kamenetsky et al. 2013) and
 818 the Bushveld Complex (Fischer et al. 2016); they mostly
 819 plot well within the two-liquids field delimited by our fit-
 820 ted binodal surface, supporting the use of the model on
 821 natural samples. Some of the data of Bushveld data plot
 822 above the binodal surface, possibly due to contamination
 823 by the host phase resulting from the small sizes of these
 824 melt inclusions, which limit the width of the electron
 825 beam. Nonetheless, they still intersect the Fe-rich melt
 826 endmember at high $\text{TiO}_2 + \text{FeO} + \text{MgO} + \text{CaO} + \text{P}_2\text{O}_5$. In
 827 contrast, the crystallization experiments of Juster et al.
 828 (1989) and Toplis and Carroll (1995), which did not
 829 develop liquid immiscibility, follow the fractionation trend
 830 but correctly plot above the binodal surface on the Si-rich
 831 melt endmember.

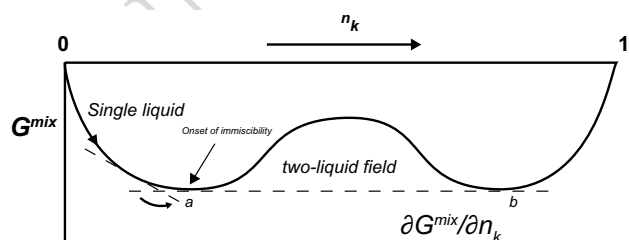


Fig. 13 Sketch of the Gibbs free energy of mixing (G^{mix}) and the change of its first derivative ($\partial G^{\text{mix}}/\partial n_k$) with respect to component k in the system, modified after Ghiorso and Carmichael (1980). a and b indicate where $\partial G^{\text{mix}}/\partial n_k$ changes to 0, i.e., the onset of unmixing and the development of immiscible melt compositions

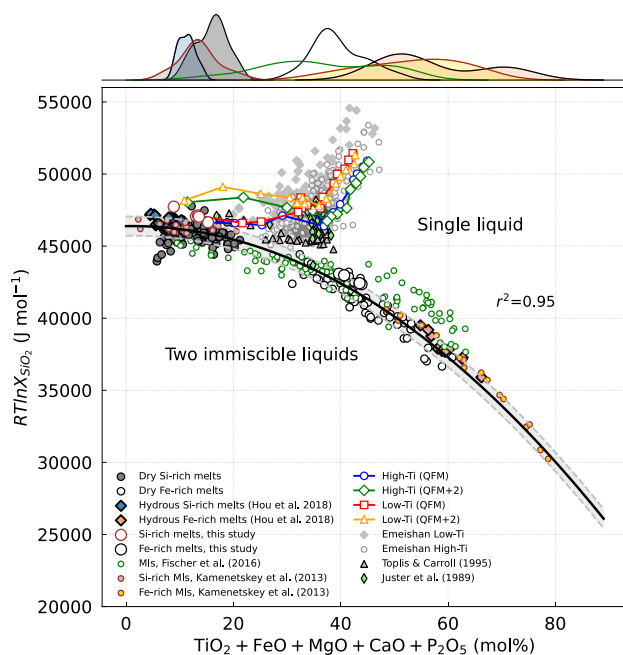


Fig. 14 Parameterization of the binodal surface for immiscible melts in LIPs. The kernel density plot at the top indicates the distribution of experimental and natural immiscible melts. The solid black line represents the best fit of our model and the gray envelope indicates the 1σ error of our estimation. Experimental data used for fitting include dry immiscibility experiments (this study; Charlier and Grove 2012; Dixon and Rutherford 1979; Philpotts and Doyle 1983; Hess et al. 1975; Longhi 1990; Ryerson and Hess 1978, 1980; Hou et al. 2017; Rutherford et al. 1974; Honour et al. 2019a) and hydrated experiments from Hou et al. (2018). Natural melt inclusion (MI) data are from the Siberian Traps, Russia (Kamenetsky et al. 2013) and the Bushveld Complex, South Africa (Fischer et al. 2016); Experimental melts that did not develop immiscibility (Toplis and Carroll 1995; Juster et al. 1989) are shown for comparison

832 Immiscibility and implications for Fe–Ti ore deposits 833 in LIPs

834 Immiscibility between Si-rich and Fe-rich melts is often pro-
835 posed as a mechanism for the origin of Fe–Ti oxide deposits
836 in layered intrusions (e.g., Zhou et al. 2005; 2013; Char-
837 lier et al. 2011; Fischer et al. 2016). However, this remains
838 controversial and these Fe–Ti oxide deposits could also be
839 produced by the simple mechanical accumulation of Fe–Ti
840 oxide crystals (e.g., Tegner et al. 2006; Song et al. 2013).

841 The Panzhihua layered intrusion within the Emeishan
842 LIP hosts one of the largest Fe–Ti oxide ore deposit and
843 is genetically associated with the Emeishan high-Ti lavas
844 (Zhou et al. 2005; Pang et al. 2008). To assess whether natu-
845 ral Emeishan lavas could have developed immiscibility, we
846 apply our model (Eq. 4) to the high-Ti and low-Ti lavas in the
847 database compiled in Fig. 14. Both the high-Ti and low-Ti
848 lavas follow trends similar to our experimental melts before
849 they attained immiscibility. The high-Ti lavas eventually

850 intersect our binodal, but this is not the case for the low-
851 Ti lavas. We note that the Emeishan lavas may have been
852 slightly hydrous (Ganino et al.; 2008; Liu et al. 2017), which
853 may shift them closer to the binodal, because H₂O depresses
854 the liquidus temperature (Parman and Grove 2004; Méd-
855 ard and Grove 2008). These results suggest that the high-Ti
856 lavas may have developed immiscibility and contributed to
857 the formation of the Fe–Ti oxide ore deposits, as previously
858 proposed by Zhou et al. (2008) and consistent with obser-
859 vations of immiscibility textures in the Emeishan LIP (Liu
860 et al. 2016; Dong et al. 2021; Wang et al. 2018, 2021; Xiong
861 et al. 2021).

862 Moreover, our experimental results indicate that FeO
863 depletion caused by Fe–Ti oxide crystallization does not
864 hinder the development of immiscibility. This suggests that
865 the formation of Fe–Ti ore could result from a combina-
866 tion of both fractionation and immiscibility. If a magma is
867 buffered at high fO_2 , either intrinsically or due to interac-
868 tion with a sedimentary host (e.g., Ganino et al. 2008) and
869 crystallizes Fe–Ti oxides early, the residual melt may still
870 evolve to develop immiscibility. It is thus probable that early
871 crystallized Fe–Ti oxides would preferentially settle to the
872 bottom of the intrusion/magma chamber, causing the melt to
873 evolve by fractional crystallization and eventually develop
874 liquid immiscibility. The unmixing process would further
875 enrich the accumulated layer of Fe–Ti oxides by downward
876 percolation of the dense immiscible Fe-rich melt or produce
877 Fe–Ti enriched layers at a more evolved stage in upper levels
878 of the intrusion/chamber.

879 Conclusions

880 We performed experiments to determine the controls of
881 primitive magma compositions and fO_2 on the liquid lines
882 of descent, phase stability, and compositions of high-Ti and
883 low-Ti LIP basalts. Our results indicate that starting com-
884 position affects phase chemistry, whereas fO_2 exerts a major
885 control on the stability and chemistry of Fe–Ti oxides. Both
886 high-Ti and low-Ti magmas can reach the two-liquid field
887 at late stages of differentiation. High fO_2 conditions trigger
888 the early crystallization of Fe–Ti oxides but do not hinder
889 the development of immiscibility. We propose a new empiri-
890 cal parameterization for the binodal surface of the two liq-
891 uid field in compositional space. Our model successfully
892 replicates unmixing in both experiments and natural melt
893 inclusions. Massive Fe–Ti oxide ores in layered intrusions
894 associated with LIPs may thus form by the segregation of
895 immiscible Fe-rich melts and/or the accumulation of early
896 Fe–Ti oxides during fractionation.

897 **Supplementary Information** The online version contains supplemen-
898 tary material available at <https://doi.org/10.1007/s00410-022-01990-x>.

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909 **Availability of data and materials** Additional data are available as supplementary electronic files.

911 **Declarations**

912 **Conflict of interest** Not applicable.

913 **References**

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