





Development and processability of AISI S2 tool steel by Laser Powder Bed Fusion

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Introduction

Tool steels are used in applications such as cutting, forming, shearing, stamping.



Zumofen et al., 2020



Laser Powder Bed Fusion is becoming attractive for the possibility of producing complex shape implementing internal cooling channels.



Tool steels by LPBF are difficult to process due to the complex chemical compositions (C, Cr, Mo, V, W, Co...) leading to defects within the final part (cracks, lack of fusion, key-hole porosities...).

This study aims to provide the basis for understanding the **PROCESSABILITY** and **MICROSTRUCTURE** of tool steel AISI S2.

Conventional manufacturing of tool steels are:

casting, electroslag remelting or powder metallurgy.







Materials and Methods

Powders obtained by Gas Atomization







Aconity MIDI Laser Powder Bed Fusion machine



20 cubes 10 x 10 x 10 mm³



Parameters

- **Power** (P, 100 250 W)
- <u>Scan speed</u> (v_s, 400 2000 mm/s)
- Layer Thickness (t, 30 µm)
- Hatch Spacing (h, 80 µm)
- Laser beam size (80 µm)
- Scan Strategy (Bidirectional 90°)
- **Preheating NO**

Volumetric







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Microstructure and in-situ Thermal Treatments



Optical micrograph after ecthing with Nital 3%

Continuous epitaxial growth





Microstructure and in-situ Thermal Treatments









Conclusions

Processability

- Medium v_s (750 1000 mm/s) and P (200 W) (Ed ~ 97 J/mm³) allow to achieve fully dense and defect-free parts;
- > High v_s (1000 2000 mm/s) and low E_d (35 69 J/mm³) cause lack of fusion and balling, regardless of the P applied;
- Low v_s (400 1000 mm/s) and mostly high E_d (69 208 J/mm³) cause large spatters and key-hole porosities, regardless of the P applied;

<u>Microstructure</u>

- > Epitaxial grains growth is observed within the microstructure;
- Fresh martensite, bainite and tempered martensite are observed. In-situ thermal treatments during the process are responsible for the change in the microstructure;













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