Multi-responsive γ-Methylene-γ-Butyrolactone / N-Vinyl Caprolactam Copolymers Involving pH-depend Reversible Lactonization

Zhuoqun Wang,^a Antoine Debuigne ^{*a}

^a Center for Education and Research on Macromolecules (CERM), University of Liege (ULiege), CESAM-RU, Sart-Tilman, Building B6a, B-4000 Liege, Belgium

* Corresponding authors: E-mail: <u>adebuigne@uliege.be</u>

Abstract

Poly(N-vinyl caprolactam) is one of the most popular thermo-responsive polymers used in the biomedical field. For application purposes, its transition temperature can be adjusted by copolymerizing N-vinyl caprolactam with comonomers of various hydrophilicity. Insertion of carboxylic acid containing vinyl monomers in P/WCL also imparts some pHresponsiveness to the latter although, at low pH, H-bonding between the protonated carboxylic groups and the amide functions of WCL induces aggregation phenomena. The present work describes a unique multi-responsive P/WCL system based on γ -methylene- γ -butyrolactone as comonomer involving a peculiar pH-depend reversible lactonization reaction. A series of narrowly distributed statistical NVCL/2M2BL-based copolymers with different molar masses and compositions were prepared via organometallic-mediated radical polymerization. Hydrolysis of the pendant lactones led to the corresponding hydroxy acid-containing copolymers allowing to tune the cloud point temperature of the polymer solution over a wide range of temperature. In contrast to the previously reported acid-bearing PMCL derivatives, acidification of the hydroxy carboxylate containing WCL-based copolymers induced the lactonization reaction leading to the regeneration of the starting $MCL/_{2}M_{2}BL$ -based copolymers, which ultimately prevents aggregation via hydrogen bonding. The synthesis and responsiveness of NVCL/2M2BL-based block copolymers were also considered and discussed. Overall, this unusual multi-responsive methylene lactone-based system constitutes a promising platform for tuning the solution properties of P/WCL, and possibly of other smart polymer sequences, opening new applicative prospects.

Introduction

Polymer foams are versatile materials that allow combining the advantageous properties Driven by the need of controlled drug delivery systems, innovative biosensors and smart actuation devices, there has been an intensive research for the development of stimuliresponsive polymers which can undergo abrupt physiochemical changes in response to slight changes in their external environmental variables such as temperature,¹ pH,² electric³ or magnetic field,⁴ light,⁵ and ionic strength.⁶ Thermo-responsive polymers characterized by a lower critical solution temperature (LCST) are certainly among the most studied stimuli responsive systems.^{7–11} In particular, those characterized by a LCST close to the physiological temperature,¹²⁻²¹ such as poly(N-vinylcaprolactam) (PMVCL),¹⁶⁻¹⁹ have attracted a lot of attention especially for biomedical applications. PMCL is a very popular thermosensitive sequence which exhibits a Type I phase change and a classical Flory-Huggins thermo-responsive phase diagram with a continuous phase transition from about 30 °C to 50 °C, depending on molar mass and concentration.^{16,22,23} Compared to another common LCST polymer, namely poly (N-isopropylacrylamide) (PMPAAm), the nitrogen atoms of PMVCL are advantageously connected to carbons of the polymer backbone, prohibiting the release of toxic primary amines upon hydrolysis.^{16,19,24,25} Another major difference with P/IPAAm is the absence of H-bond donating group in the molecular structure of P/WCL that is not able to form intra- or interchain hydrogen bonds which limits the hysteresis phenomenon.²³ Given its biocompatibility and coil to globule transition temperature close to the body temperature, P/WCL has been largely considered for biomedical applications,^{26–30} including drug delivery systems and smart surfaces in the forms of micelles,³¹ micro^{27,29,32}/nano gel^{33,34} or multilaver hydrogels.^{35,36}

A classical approach to tune the LCST of P/WCL-based polymers but also to achieve multi-responsive behaviors consists in the incorporation of comonomers in the backbone.^{29,37–39} For example, copolymerizing *N*/CL with ionizable monomers imparted dual pH- and thermo-sensitivity to the final copolymers.^{40–46} Typical ionizable comonomers which have been associated to *N*/CL in order to fabricate dual responsive microgels of interest for drug delivery applications are acrylic acid (AA),^{42,46} methacrylic acid (MAA) ^{27,43,44} and itaconic acid (IA).^{27,45} At physiological pH, the ionized carboxylate groups predominated, increasing the hydrophilicity of the copolymer as well as its LCST. At pH values lower than pKa, however, carboxylic acid functions were mainly in their protonated state and served as strong hydrogen bond donors for the lactam groups of *N*/CL units leading to the insolubility of copolymers in water at room temperature (Scheme 1a).⁴⁷ This **c**haracteristic impedes the design of a PNVCL-based coil to globule transition system in which LCST can be simply tuned by pH adjustment.

Recently, some methylene heterocyclic compounds have received great interest as monomers notably for the preparation of functional polymers, including carboxylic acid-containing ones. For example, γ -methylene- γ -butyrolactone (γ M γ BL) was copolymerized in a controlled manner with vinyl acetate followed by hydrolysis of both the acetate and lactone functions leading to acid-functionalized poly(vinyl alcohol) with improved water solubility at neutral pH.⁴⁸ As another illustration, saponification of poly(α -methylene- γ -butyrolactone) (poly(α M γ BL)) generated one hydroxyl group and one carboxylate function per repeating unit.⁴⁹ This highly hydrophilic polymer sequence was also used to manufacture superabsorbent materials.^{49–51} Interestingly, relactonization reaction was observed upon acidification of the hydroxy carboxylate-functional polymer at 25°C (Scheme 1b).⁴⁹



Scheme 1. Illustration of the pH-responsive acrylic acid/NVCL copolymers with intra/inter H-bond formation (a), the reversible lactonization of poly($\alpha M \gamma BL$) (b) and the targeted dual pH- and thermoresponsive $\gamma M \gamma BL/NVCL$ copolymers.

The present work reports an innovative multi-responsive NVCL-based system involving a methylene-*p*-butyrolactone compound as comonomer and emphasizing a peculiar pHdepend ring opening-closure of the pendant lactones (Scheme 1c). Compared to the (meth)acrylic acid-containing P/WCL copolymers, aggregation via intra/intermolecular Hbonding should be prevented for such MBL/NVCL copolymers thanks to the lactonization reaction in acidic medium. For this purpose, γ -methylene- γ -butyrolactone ($\gamma M \gamma BL$) was selected as comonomer of NVCL since both compounds consists in less activated monomers (LAMs) and should thus exhibit close reactivity favorable to their copolymerization. A series of narrowly distributed P(*NVCL-co-yMyBL*) copolymers with different molar masses and compositions were prepared via organometallic-mediated radical polymerization (OMRP), an efficient reversible deactivation radical polymerization (RDRP) method suitable for LAMs⁵²⁻⁵⁶ including NCVL.^{11,57,58} The solution behavior of these copolymers upon pH- and temperature changes was then investigated emphasizing their specific stimuli responsiveness and the key role of the reversible lactonization reaction. The incorporation of MyBL within MVCL-based block copolymers and its effect on the copolymer self-assembly in water was also examined.

Experimental section

Materials

Cobalt(II) acetylacetonate (Co(acac)₂) (97%, Acros), 2,2,6,6-tetramethylpiperidine 1-oxyl (TEMPO) (98%, Aldrich), butylamine (99%, Fluorochem) and 2,2'-azobis(4-methoxy-2,4-dimethyl valeronitrile) (V-70, $t_{1/2} = 10$ h at 30 °C) (>98%, Wako) were used as received. *N*-vinyl caprolactam (98%, Aldrich) was purified by flash basic alumina column with n-hexane at 40 °C, and then further purified by recrystallization in n-hexane at 6 °C. 2,2'-azobisisobutyronitrile (AIBN, $t_{1/2} = 10$ h at 65 °C) (98%, Aldrich) was recrystallized in methanol at 6 °C. Sodium hydroxide (NaOH, ≥97%, Acros), silica gel for column chromatography (60 Å, ROCC S.A.), tetrahydrofuran (THF, ≥99.9%, VWR), methanol (MeOH, ≥99.8%, VWR), n-hexane (>99%, VWR), diethyl ether (Et₂O, 100%, VWR) and ethyl acetate (≥99.9%, VWR) were used as received. N,N-dimethylformamide (DMF, >99%, VWR) was dried over molecular sieves prior to use. Dichloromethane (CH₂Cl₂) was degassed and dried over 4 Å molecular sieves. 1,4-dioxane (99.8%, Aldrich) was purified by distillation under reduced pressure. VAc (>99%, Aldrich) was dried under calcium hydride, purified by distillation under reduced pressure, and degassed by freeze-drying cycle under vacuum. The alkyl-cobalt(III) adduct initiator (PVAc<-Co(acac)₂), [Co(acac)₂-

(CH(OAc)-CH2)_{<4}R₀], R₀ being the primary radical generated by 2,2'-azo-bis(4-methoxy-2,4-dimethyl valeronitrile) (V-70, Wako), was prepared as described previously⁵⁹ and stored as a CH₂Cl₂ solution at -20 °C under argon. γ -methylene- γ -butyrolactone (γ M γ BL) was prepared as described previously and stored under argon.⁴⁸ Dialyses were carried out with a Spectra/Por dialysis membrane (pretreated RC tubing 1 kDa).

Characterization

The size exclusion chromatography (SEC) analyses were carried out in DMF containing 0.025 M LiBr at 55 °C with a Waters chromatograph equipped with three columns [PSSGRAM 1000 Å (×2), 30 Å], a dual λ absorbance detector (Waters 2487), and a refractive index detector (Waters 2414). The system was operated at a flow rate of 1 mL/min, and polystyrene calibration was used. The absolute molar masses of the P/WCL, $P(NVCL-co-\gamma M\gamma BL)$ and $P(NVCL)-b-P(NVCL-co-\gamma M\gamma BL)$ block copolymer were determined by SEC equipped with a multiangle laser light scattering (MALLs) detector in DMF/LiBr (0.025 M). The Wyatt MALLs detector (120 mW solid-state laser, k ¹/₄ 658 nm, DawnHeleos S/N342-H) measures the excess Rayleigh ratio Rh (related to the scattered intensity) at different angles for each slice of the chromatogram. The specific refractive index increment (dn/dc) of polymer was measured by using a Wyatt Optilab refractive index detector (k ¹/₄ 658 nm). Data were processed with the Astra V software (Wyatt Technology). ¹H nuclear magnetic resonance (¹H NMR) (400 MHz) and power gate proton-decoupled ¹³C NMR (¹³C{¹H} NMR) (101 MHz) spectra were recorded at 298 K with a Bruker AVANCE III HD spectrometer ($B_0 = 9.04$ T) (400 MHz) and treated with MestreNova software. Fourier transform infrared (FT-IR) spectroscopy spectra were recorded on a Thermo Fisher Scientific Nicolet IS5 equipped with an ATR ID5 module using a diamond crystal (650-4000 cm⁻¹). Dynamic light scattering (DLS) and zeta potential analyses were performed on a Delsa NanoC instrument. The transmittance of the aqueous solutions of polymers in MiliQ-water (5 mg/mL) was recorded on a Jasco V630 UV-vis spectrophotometer at 260 nm. Aqueous solution of polymers were placed in a thermostated quartz cuvette. The heating/cooling rate was 2 °C/min between 10 °C and 80 °C. The cloud point temperature (T_{CP}) was defined at the half drop transmittance from the initial transmittance. pH measurements were performed via pH test paper.

Conventional radical copolymerization of NVCL and \gamma M \gamma BL. AIBN (0.079 g, 0.48 mmol), *N*VCL (2.00 g, 14.37 mmol) and $\gamma M \gamma BL$ (0.16 g, 1.60 mmol) were placed in a 25 mL Schlenk tube with 1.94 mL of anhydrous 1,4-dioxane. After degassing by bubbling argon for 20 min, the polymerization medium ($f \circ_{\gamma M \gamma BL} = 0.1$) was heated at 65 °C. After 19 h, reaction mixture was very viscous and was diluted with THF. The monomer conversion and the molecular parameters (M_n , D) of the polymer were then determined by ¹H NMR in CDCl₃ and SEC in DMF containing 0.025 M LiBr, respectively. The final copolymer was

purified via repeated precipitations (three times) in hexane and dried under vacuum overnight at 60 °C before characterization by ¹H NMR and ¹³C NMR in CDCl₃. SEC-MALLS measurements were carried out in DMF/LiBr (0.025 M) in order to determine the absolute molar mass of P(*N*VCL-*co*- γ M γ BL) ($f^{\circ}_{\gamma M\gamma BL} = 0.1$) (d*n*/d*c*=0.1769 mL/g).

A similar experiment was carried out with a different initial comonomer feed ratio, namely $f_{M,BL}^{\circ} = 0.2$, keeping constant all other parameters (65 °C, [comonomers]₀/[AIBN]₀ = 100/3).

The copolymer composition was calculated based on the relative intensity of ¹H NMR signals corresponding to each comonomer using the following equation, where signals **c** and **e** correspond to 2H in α -position of the nitrogen atom and 2H in α -position of the carbonyl group of NVCL units, respectively, whereas signals **g** and **h** correspond together to 4H of the lactone of the MBL units.

$$F_{\gamma M \gamma BL} = \frac{\frac{\int e, h, g - \int c}{4}}{\frac{\int e, h, g - \int c}{4} + \frac{\int c}{2}}$$

Organometallic-mediated radical copolymerization of MVCL and MyBL. NVCL (3.60 g, 25.8 mmol) and yMyBL (0.28 g, 2.88 mmol) were placed in a 25 mL Schlenk tube, degassed by three freeze-pump-thaw cycles followed by bubbling argon for 10 min at 40 °C. A solution of alkyl-cobalt(III) initiator (PVAc_{<4}-Co(acac)₂) in CH₂Cl₂ (1.82 mL of a 0.063 M stock solution, 0.12 mmol) was placed in a second Schlenk tube followed by the evaporation of CH₂Cl₂ under reduced pressure at room temperature. The NVCL/yMyBL mixture was then transferred via cannula into the Schlenk tube containing the alkyl cobalt ([NVCL]0/[yMyBL]0/[PVAc<4polymerization initiator followed by at 40 °C $Co(acac)_{2} = 225/25/1$, $f_{MBL} = 0.1$). During polymerization, samples were regularly taken and added with TEMPO in order to guench the polymerization. The monomer conversion was determined by ¹H NMR in CDCl₃ (1 mg of TEMPO was added per mL of CDCl₃). The molecular parameters (M_n , D) of the polymer were determined by SEC in DMF/LiBr (0.025) M) (samples were dissolved in DMF containing 10 mg of TEMPO/mL). Polystyrene was used as calibration. After 2.5 h, the polymerization mixture was guenched by the addition of 2 mL of TEMPO in THF (0.10 g/mL). The final copolymer was purified by precipitation in hexane three times followed by drying at 60 °C overnight before characterization via ¹H NMR and ¹³C NMR in CDCl₃.

Similar experiments were carried out with different initial comonomer feed ratios, namely $f_{\gamma M \gamma BL}^{\circ} = 0.2$, $f_{\gamma M \gamma BL}^{\circ} = 0.3$, keeping constant all other parameters (65 °C, bulk,

[comonomers]₀/[PVAc₄-Co(acac)₂]₀ = 250/1). The monomer/initiator ratios were also varied [comonomers]₀/[PVAc₄-Co(acac)₂]₀ = 450/50/1 and 675/75/1, keeping constant all other parameters (65 °C, bulk, $f_{\gamma M \gamma BL}^{\circ} = 0.1$).

Organometallic-mediated radical polymerization of NVCL. *N*VCL (1.00 g, 7.18 mmol) was placed in a 25 mL Schlenk tube under argon followed by bubbling argon for 10 min at 40 °C. This Schlenk tube was then placed in an ice bath to cool the temperature followed by addition of a solution of alkyl-cobalt(III) initiator (PVAc_{<4}-Co(acac)₂) in CH₂Cl₂ (0.45 mL of a 0.063 M stock solution, 0.029 mmol) and evaporation of CH₂Cl₂ under reduced pressure at room temperature. Subsequently, the reaction mixture was carried out at 40 °C. After 6 h, the polymerization mixture was quenched by the addition of 2 mL of TEMPO in THF (0.10 g/mL) and samples were withdrawn for determining the conversion by ¹H NMR in CDCl₃ and the molecular parameters (M_n , D) by SEC in DMF/LiBr (0.025 M). The final copolymer was purified by precipitation in hexane three times followed by drying at 60 °C overnight before characterization via ¹H NMR and ¹³C NMR in CDCl₃.

Synthesis of PNVCL-b-(NVCL-co-yMyBL) block copolymer by OMRP. NVCL (2.50 g. 18.0 mmol) was placed in a 25 mL Schlenk tube under argon followed by bubbling argon for 10 min at 40 °C. This Schlenk tube was then placed in an ice bath to cool the temperature. A solution of alkyl-cobalt(III) initiator (PVAc<4-Co(acac)2) in CH2Cl2 (0.58 mL of a 0.063 M stock solution, 0.036 mmol) was added followed by the evaporation of CH₂Cl₂ under reduced pressure at room temperature. The reaction mixture was carried out at 40 °C. After 1 h, a sample was picked out of the tube and added with traces of TEMPO (~10 mg/mL) for determining the monomer conversion by ¹H NMR in CDCl₃ and the molecular parameters of the P/WCL precursor by SEC MALLs in DMF/LiBr (0.025 M). Then, 2M2BL (0.35 g, 3.59 mmol) degassed by three freeze-pump-thaw cycles was injected to the reaction mixture under argon and the polymerization medium was heated at 40 °C. After 6 h, the polymerization mixture was guenched by the addition of 1 mL of TEMPO in THF (0.10 g/mL) and samples were withdrawn for determining the conversion by ¹H NMR in CDCl₃ and the molecular parameters (M_n , D) by SEC MALLs in DMF/LiBr (0.025 M) (dn/dc=0.0938 mL/g). The final copolymer was purified by precipitation in hexane three times followed by drying at 60 °C overnight before characterization via ¹H NMR and ¹³C NMR in CDCl₃.

Base hydrolysis of P(NVCL-co-\gammaM\gammaBL). P(NVCL-co- γ M γ BL) (0.26 g, 9500g/mol, $F_{\gamma M\gamma BL}$ =0.12) was dissolved in MeOH (2.6 mL). NaOH (0.26 g, 6.5 mmol) in deionized water (2.6 mL) was added to the reaction mixture followed by stirring at room temperature for 48 h. The solution was then dialyzed (membrane 1kDa) against distilled water for 48 h followed by lyophilization. The copolymer, recovered as a white power, was characterized by FT-IR spectroscopy, ¹H NMR and ¹³C NMR in CDCl₃.

Similar treatments were applied to P(*N*VCL-*co*- γ M γ BL) (13000 g/mol, $F_{\gamma M \gamma BL}$ =0.07), P(*N*VCL-*co*- γ M γ BL) (11500g/mol, $F_{\gamma M \gamma BL}$ =0.10) and P*N*VCL-*b*-P(*N*VCL-*co*- γ M γ BL) (31500 g/mol, $F_{\gamma M \gamma BL}$ =0.10).

Results and discussion

Conventional radical copolymerization of $\gamma M \gamma BL$ and NVCL.

The $\gamma M \gamma BL/NVCI$ copolymerization was first performed by conventional radical polymerization in anhydrous 1,4-dioxane at 65 °C using AIBN as initiator (Table 1). Two molar fractions of $\gamma M \gamma BL$ ($f^{\circ}_{\gamma M \gamma BL} = 0.1$ and 0.2) were considered in order to produce P(*NVCI-co-\gamma M \gamma BL*) with different compositions. High conversion of both monomers was achieved after 19 h of reaction (> 90 %) leading to copolymers with molar masses around 20000 g/mol according to SEC measurements using a PS calibration or a multi angle light scattering detector (SEC-MALLs) (M_{n SEC cal PS} = 20000 g/mol, M_{n SEC MALLS} = 18000 g/mol, entry 1 Table 1).





Entry	f° _M ℃L	f° ₇ м ₇ в∟ ·	Conv. (%) ^a		M n SEC	ÞÞ	E.v.o. C	E
			NVCL	γ Μ γBL	(g/mol) ^b	D	I NVCL	<i>I[−] γ</i> M <i>γ</i> BL
1	0.9	0.1	99	91	20000	2.0	0.95	0.05
2	0.8	0.2	99	93	19000	2.2	0.92	0.08

Conditions: 65 °C, in anhydrous 1,4-dioxane, 19 h, [comonomers]₀/[AIBN]₀=400/12. ^a Determined by ¹H NMR in CDCL₃. ^b Determined by SEC in DMF/LiBr using PS calibration. ^c Determined by ¹H NMR in CDCL₃ after purification.

The composition of the copolymers was determined via ¹H NMR by comparison of the intensity of signal **c** at 3.24 ppm corresponding to the -C*H*₂-NC=O- of *N*VCL and signals between 2.2 and 2.8 ppm containing peaks **h** and **g** of γ M γ BL units and peak **e** of *N*VCL (Figure 1, see equation in experimental section). The content of γ M γ BL in the copolymer increased from 5 mol% to 8 mol% when the molar fraction of γ M γ BL in the feed increased

from 0.1 to 0.2. The successful incorporation of $\gamma M \gamma BL$ in the copolymer was confirmed by FT-IR analysis showing a peak at 1770 cm⁻¹ assigned to C=O of the pendant lactone groups (Figure S1).



Figure 1. ¹H NMR of P(*N*VCL-*co*- γ M γ BL) ($M_n = 20000 \text{ g/mol}$, $F_{\gamma M\gamma BL} = 0.05$) prepared by conventional radical polymerization (Table 1, entry 1). * designates the signal of the methyl groups of the AIBN initiating fragment.

Reversible deactivation radical copolymerization of $\gamma M \gamma BL$ and NVCL.

The synthesis of P(*N*/CL-*co*- γ M γ BL) statistical copolymers with predictable molar mass, low dispersity and precise composition was then investigated by reversible deactivation radical copolymerization of *N*/CL and γ M γ BL. For this purpose, the organometallic mediated radical polymerization (OMRP) was selected as RDRP method due to its ability to mediate the radical polymerization of several non-conjugated and less activated monomers (LAMs)^{52,60,61} but also their copolymerizations. In practice, the OMRP of *N*/CL and γ M γ BL was conducted in bulk at 40 °C using a previously reported alkyl cobalt(III) species consisting in oligomers of poly(vinyl acetate) end-functionalized by a bis-acetylacetonate cobalt complex (PVAc_4-Co(acac)₂)⁵⁹ as initiator and controlling agent (Scheme 2 and Table 2).



Scheme 2. Organometallic mediated radical copolymerization of NVCL and yMyBL.

To gain insight into this copolymerization, both the monomer/alkyl cobalt molar ratio and the feed composition were changed. First, the comonomers/R-Co(acac)₂ molar ratio was varied from 250 to 750 while keeping an initial NVCL/2M2BL feed composition of 0.9/0.1 (Table 2, entries 1-3). The comonomer conversion and the molecular parameters of the copolymers were determined throughout the polymerization by ¹H NMR and SEC, respectively. The structure and composition of the copolymers were determined by ¹H NMR and confirmed by COSY and HSQC analyses (Figure S2). Similar polymerization kinetics were observed for these experiments and total conversion exceeded 60 % after few hours in all cases (Figure 2b). As confirmed by the regular shift of the SEC chromatograms towards lower elution volume (Figure S3), the Mn of the P(NVCL-coyMyBL) copolymer increased linearly with the conversion (Figure 2a) confirming the controlled character of the copolymerization. The dispersity of the copolymers remained rather low along the polymerization ($D \sim 1.1-1.35$) and only slightly increased above 50% conversion when higher molar mass was targeted probably due to the occurrence of some irreversible termination reactions (Table 2 entry 3). As expected, for similar monomer conversion, the molar mass of the copolymer increased with the comonomers/R-Co(acac)₂ molar ratio (e.g. 13000 g/mol and 33000 g/mol for [comonomers]₀/[R-Co(acac)₂]₀ of 250 and 750, respectively, compare entries 1 and 3 in Table 2). Copolymers

with similar composition were formed ($F_{\mathcal{M}\mathcal{B}L} = 0.05 \sim 0.07$) in these experiments involving the same comonomer feed composition ($f_{\mathcal{M}\mathcal{B}L}^{\circ} = 0.1$).



Figure 2. (a) Dependence of M_n (full symbols) and \mathcal{D} (hollow symbols) on the total comonomers conversion and (b) time dependence of $\ln(M_0/M)$ for the OMRP of NVCL and $\gamma M \gamma BL$ with different [monomers] $_0/[$ R-Co(acac) $_2$] $_0$ ratio: \blacksquare [monomers] $_0/[$ R-Co(acac) $_2$] $_0$ =250/1, \bullet [monomers] $_0/[$ R-Co(acac) $_2$] $_0$ = 500/1, \blacktriangle [monomers] $_0/[$ R-Co(acac) $_2$] $_0$ =750/1. ($f^\circ_{\gamma M \gamma BL} = 0.1$, Table 2 entry 1-3)

Entry	[///CL]0/[7/M7/BL]0/	f° ₂M₂BL	Time	Conv. (%) ^a		Conv.	M _{n SEC}	Þ	E c
	[R-Co(acac) ₂] ₀		(min)	NVCL	γMγBL	total (%)	(g/mol) ^b	D.	₽ _% M _% BL °
			20	4	0	4	1600	1.09	
			40	6	4	6	2100	1.10	
			60	9	5	9	2800	1.10	
1	225/25/1	0.1	80	13	11	13	3400	1.10	
			100	16	14	16	4200	1.08	
			180	33	35	33	8200	1.11	
			300	61	48	60	13000	1.21	0.07
2			20	4	12	5	2400	1.10	
			60	14	16	14	7200	1.08	
	450/50/1	0.1	100	19	17	19	10600	1.08	
			120	26	25	26	13000	1.14	
			360	70	49	68	23000	1.28	0.05
3			30	7	10	7	5100	1.09	
			60	9	11	9	8200	1.08	
	675/75/1	0.1	90	12	16	12	10700	1.07	
			120	16	19	16	14000	1.08	
			540	74	31	70	33000	1.35	0.07
4			30	7	1	6	2000	1.11	
			60	12	3	10	3000	1.09	
	200/50/1	0.2	90	17	4	14	4100	1.09	
			128	25	10	22	5500	1.08	
			300	63	46	60	11500	1.18	0.10
5			30	2	1	2	2000	1.07	
			66	7	3	6	3200	1.10	
	175/75/1	0.3	98	14	4	11	4100	1.10	
			120	18	4	14	4900	1.10	
			300	55	25	46	9500	1.15	0.12
6	250/1	0	300	60	/	60	28000	1.18	0

Table 2. Organometallic mediated radical copolymerization of NVCL and MyBL.

Conditions: 40 °C, bulk. ^a Determined by ¹H NMR in CDCL₃. ^b Determined by SEC in DMF/LiBr using PS calibration. ^c Determined by ¹H NMR in CDCL₃ after purification.

Next, the $\gamma M \gamma BL/NVCL$ feed ratio was changed ($f^{\circ}_{\gamma M \gamma BL}$ between 0.1 ~ 0.3) while keeping a [comonomers]₀/[R-Co(acac)₂]₀ ratio equal to 250 (Table 2, entries 1,4-5). The copolymerization slowed down when the $f^{\circ}_{\gamma M \gamma BL}$ was increased, i.e. the global comonomer conversion reached 46 % after 5 hours when $f^{\circ}_{\gamma M \gamma BL}$ was increased to 0.3 (entry 5 in Table 2, Figure S4b). Nevertheless, all these copolymerizations took place in a controlled manner (Figures S4a and S5). Ultimately, P(*NVCL-co-\gamma M \gamma BL*) copolymers of low dispersity (1.15~1.21) and molar mass between 9500 g/mol and 13000 g/mol were collected. The copolymer composition was successfully tuned by adjusting the $\gamma M \gamma BL$

content in the copolymerization feed, namely $F_{\gamma M \gamma BL}$ increased from 0.07 to 0.12 as $f_{\gamma M \gamma BL}^{\circ}$ increased from 0.1 to 0.3.

Hydrolysis of the $\gamma M \gamma BL/NVCL$ statistical copolymers.

The copolymerization of NVCL with hydrophobic or hydrophilic monomers is a widespread approach to modulate its LCST and meet the requirements of specific applications.¹⁸ In this respect, $P(NVCL-co-\gamma M\gamma BL)$ is of particular interest since the ring strain of butyrolactone and the electrophilicity of its carboxyl group enable a wide variety of postpolymerization modifications that are likely to change the hydrophilicity of the copolymer, and so its thermo-responsive behavior. In particular, we considered the hydrolysis of the lactones of P(NVCL-co-yMyBL) in order to generate hydroxy acid functions along the P/WCL backbone (Figure 3a). For this purpose, the P(MCL-co- $\gamma M\gamma BL$) was treated with NaOH in a MeOH/H₂O (1/1:v/v) mixture for 48h at room temperature in order to hydrolyze the pendant lactone and produce pH responsive copolymers containing hydroxy carboxylate functions, namely P(MCL-co-HPEA). The successful saponification of the butyrolactone was notably confirmed by FT-IR (Figure 3b) as assessed by the disappearance of the characteristic signal of the ester functions (C=O ester) at 1769 cm⁻¹ and the appearance of a signal at 1568 cm⁻¹ corresponding to carboxylate groups. The formation of P(NVCL-co-HPEA) was further confirmed by ¹H NMR, COSY and HSQC analyses (Figure S6).



Figure 3. (a) General conditions for the conversion of $P(NVCL-co-\gamma M\gamma BL)$ into $P(NVCL-co-\gamma M\gamma BL)$ and (b) FT-IR spectra of $P(NVCL-co-\gamma M\gamma BL)$ (sample from Table 2, entry 4) before (black) and after hydrolysis (blue) of the lactone functions.

Thermo-responsiveness of the $\gamma M \gamma BL/WCL$ and HPEA/WCL statistical copolymers.

The effect of the incorporation of $\gamma M\gamma BL$ in P/WCL and of the hydrolysis of the pendant lactones on the solution behavior of the copolymers was then studied. The cloud point temperature (*T*_{CP}) of the copolymers aqueous solutions (5 mg/mL) were measured by turbidimetry. A heating rate of 2 °C/min was used and *T*_{CP} was defined as the temperature at which half of the transmittance of the solution was lost. First, we examined the thermal response of a series of P(*N*/CL-*co*- $\gamma M\gamma BL$)s with a $\gamma M\gamma BL$ molar fraction around 0.07 and a molar mass ranging from 13000 to 33000 g/mol (Table 2, entries 1-3). A P/N/CL of 10000 g/mol characterized by a *T*_{CP} of 36.9 °C served as reference. According to the

overlay of the transmittance vs temperature plots, the incorporation of hydrophobic $\gamma M\gamma BL$ in the P/WCL decreased the T_{CP} by few degrees to 34.2 °C (Figure 4a). The transition temperature of the P(/WCL-*co*- $\gamma M\gamma BL$) with similar composition (Figure 4a) also slightly decreased from 34.2 °C to 33.1 °C as the molar mass of the copolymer increased from 13000 to 33000 g/mol in line with the typical Flory–Huggins behavior of P/WCL-based copolymers.¹⁸ In addition, increasing the $\gamma M\gamma BL$ content of the copolymer from 7 to 12 mol% while keeping similar molar masses (9500-13000 g/mol, Table 2 entries 1,4,5) decreased the T_{CP} from 34.2°C to 31.1 °C (Figure 4b). In all cases, hysteresis of only few degrees was observed.

According to the previous section, the hydrophobic $\gamma M\gamma BL$ units of P(*N*VCL-*co*- $\gamma M\gamma BL$)s were converted into more hydrophilic hydroxy carboxylate functions upon treatment with NaOH. The *T*_{CPS} of the resulting P(*N*VCL-*co*-HPEA)s, measured at pH 6 to ensure proper deprotonation of the carboxylic groups, were superior to those of their P(*N*VCL-*co*- $\gamma M\gamma BL$)s precursors (Figure 4c). For example, the transition temperature of the P(*N*VCL-*co*-HPEA) containing only 7 mol% of hydroxy carboxylate functions was 24.1 °C higher compared its $\gamma M\gamma BL$ counterpart (Table 2, entry 1). Moreover, no precipitation of the P(*N*VCL-*co*-HPEA) was observed between 10 °C and 80 °C when the HPEA content reached 10 and 12 mol%. Overall, the introduction $\gamma M\gamma BL$ within P*N*VCL and the postpolymerization modification of the lactones constitute a promising platform for tuning the *T*_{CP} of the copolymer solution over a wide temperature range.

Dynamic light scattering (DLS) analyses also revealed different solution behaviors for P(NVCL-co- $\gamma M \gamma BL$) and P(NVCL-co-HPEA) ($F_{\gamma M \gamma BL} = F_{HPEA} = 0.07$) above their T_{CP} (Figure 5). A transition from free chains to particles was observed when the aqueous solution of the $\gamma M \gamma BL$ -containing copolymer exceeded the T_{CP}, namely 34.2 °C. In this case, the diameter of the particles increased form 500 nm to several microns as temperature increased from 35 °C to 55 °C due to dehydration and aggregation of P(MVCL-co- $\gamma M\gamma BL$) that is deprived of any stabilizing group. For P(*NVCL-co*-HPEA), however, only submicron particles (200-400 nm) were observed above the T_{CP}, namely 61.0 °C. In contrast to $P(NVCL-co-\gamma M\gamma BL)$, the carboxylate functions of P(NVCL-co-HPEA) provided electrostatic stabilization to the particles and prevented their further aggregation. It is noteworthy that the particle size first increased up to about 400 nm at 75 °C before decreasing to 220 nm at higher temperature. As an explanation, particles formed at 75 °C were still swollen by water and underwent further dehydration of their core when the temperature was increased leading to the reduction of their size to about 200 nm. For both copolymers, no more objects were detected by DLS after cooling the solutions to room temperature proving the perfect reversibility of the process.



Figure 4. Turbidimetry analysis of aqueous solutions of (a) $P(NVCL-co-\gamma M\gamma BL)$ s with different molar masses (Table 2, entries 1-3), (b) $P(NVCL-co-\gamma M\gamma BL)$ s with different compositions (Table 2, entries 1, 4, 5), (c) P(NVCL-co-HPEA)s with different compositions (*F*_{HPEA} =0.07~0.12) (samples derived from precursors in Table 2, entries 1, 4, 5) at a concentration of 5 mg/mL.



Figure 5. DLS measurements of object diameter (red line) and turbidimetry (black lines) in water at 5 mg/mL as a function of temperature for (a) P(NVCL-*co*- γ M γ BL) (Mn = 13000 g/mol, $F_{\gamma M\gamma BL} = 0.07$) and (b) the corresponding hydrolyzed P(NVCL-*co*-HPEA) ($F_{HPEA} = 0.07$) at pH 6.

pH-responsiveness of HPEA/NVCL statistical copolymers

The pH-responsive behavior of carboxylic acid-containing P(*N*VCL-co-HPEA) was also explored. As mentioned above and illustrated in Figure 6a, at 25 °C and pH 6, the P(NVCL-co-HPEA) containing 12 mol% of HPEA units was mainly in its deprotonated state and completely soluble in water. After acidification with HCl to pH 2, the solution became instantaneously turbid. This phenomenon was reported for other acidfunctionalized P/WCL such as P(/WCL-co-AA),⁴⁷ P(/WCL-co-MAA)^{43,62} and P(/WCL-co-IA)⁴⁵ and rationalized by the formation of intra and intermolecular H-bonding between the protonated carboxylic acid of HPEA and the amide functions of the NVCL units. In sharp contrast to the previously reported acid-functionalized PMCL, however, the P(MCL-co-HPEA) completely dissolved in water after 12 h without extra stimuli. This specific behavior of P(WCL-co-HPEA) compared to other acid-containing PWCL derivatives was attributed to the re-lactonization reaction and the regeneration of the pristine P(NVCL-covMvBL) copolymer. This assumption was confirmed by FT-IR analyses. As shown in Figure 6b, right after acidification of P(NVCL-co-HPEA), the signal at 1568 cm⁻¹ corresponding the carboxylate moieties was replaced by a peak characteristic of the protonated carboxylic group at 1720 cm⁻¹. After 12h at room temperature and pH 2, this COOH signal spontaneously disappeared and the lactone signal reappeared at 1769 cm⁻ ¹ confirming the regeneration of the $\gamma M \gamma BL$ units. A very similar behavior was observed for the copolymer containing a lower HPEA content, namely 7 mol%.

Finally, the thermo-responsiveness of the reconstructed P(*N*VCL-*co*- γ M γ BL) at pH 2 was investigated and compared to the P(*N*VCL-*co*-HPEA) before acidification (pH 6) and to the initial P(*N*VCL-*co*- γ M γ BL). After acid treatment of P(*N*VCL-*co*-HPEA) containing 7 mol% of HPEA and 12 h, the *T*_{CP} of the copolymer markedly decreased from 61.0 °C to 38.6 °C, so close to the *T*_{CP} of the starting P(*N*VCL-*co*- γ M γ BL) (Figure 7a). For the P(*N*VCL-*co*-HPEA) with higher content of HPEA (12 mol%), the acidification essentially restored the LCST behavior of the starting P(*N*VCL-*co*- γ M γ BL) copolymer which further supports the relactonization reaction in acidic medium (Figure 7b). Note however that the transition temperatures of the initial and the reconstructed P(*N*VCL-*co*- γ M γ BL)s differ by few degrees. The slight discrepancy might be due to residual traces of alcohol and acid functions in the final copolymer rendering the latter slightly more hydrophilic than the starting P(*N*VCL-*co*- γ M γ BL)s. It is also worth remembering that salts generated during the acid/base treatments might also affect the LCST behavior of the copolymer. In the end, the P(*N*VCL-*co*- γ M γ BL) copolymer exhibits a peculiar dual pH- and thermo-responsive behavior involving a unique reversible delactonization reaction.



Figure 6. (a) Illustration of the solution behavior of P(*N*VCL-*co*-HPEA) ($F_{\gamma M\gamma BL}$ =0.12, Table 2 entry 5) at different pH. (b) FT-IR spectra of the initial P(*N*VCL-*co*- $\gamma M\gamma BL$) copolymer and P(*N*VCL-*co*-HPEA) after different acid-base treatments.



Figure 7. Turbidimetry analysis of aqueous solutions (5mg/mL) of the P(*N*VCL-*co*- γ M γ BL) precursors (a) $F_{\gamma M\gamma BL} = 0.07$ and (b) $F_{\gamma M\gamma BL} = 0.12$ (at pH 7, and their corresponding P(*N*VCL-*co*-HPEA)s (pH 6) and P(*N*VCL-*co*- γ M γ BL)-II copolymers resulting from the relactonization reaction (pH 2).

Synthesis and stimuli-responsiveness of $\gamma M \gamma BL$ - and HPEA-based block copolymers.

Stimuli-responsive polymers often deliver their full potential when incorporated in more complex macromolecular architectures leading to specific solution behaviors such as multistep assembly.^{11,57} In this perspective, we considered the synthesis of a P/VCL-*b*-P(/VCL-*co*- γ M γ BL) block copolymer and its P/VCL-*b*-P(/VCL-*co*-HPEA) derivative before studying their thermal response in water. In theory, the incorporation of more hydrophobic γ M γ BL or more hydrophilic HPEA units in one of the sequence of the copolymer could induce distinct transition temperatures for each block and possibly a peculiar assembly of the copolymer in solution.

Taking advantage of the controlled character of the OMRP of MVCL (Table 2, entry 6) and of its copolymerization with $\gamma M \gamma BL$, we synthesized well-defined P/WCL-b-P(/WCL-co- $\gamma M \gamma BL$) by sequential organometallic-mediated radical polymerization of MVCL and copolymerization of $MCL/\gamma M\gamma BL$ in one-pot (Figure 8a). First, the OMRP of MCL was performed in bulk at 40 °C using a [NVCL]/[R-Co(acac)₂] molar ratio of 500. After 1h, the MVCL conversion reached 13% leading to a PMVCL-Co(acac)₂ precursor with $M_{n \text{ MALLs}}$ of 12600 g/mol and a dispersity of 1.04. Without prior purification, an $\gamma M \gamma BL$ was added to this medium in order to reach a [MCL]/[$\gamma M\gamma BL$] feed ratio equal to 82/18 and the copolymerization was conducted at 40 °C for 6h. As assessed by the clear shift of the SEC chromatograms towards lower elution time (Figure 8b), the chain extension polymerization was successful with an almost complete consumption of the P/WCL precursor. As a result, a well-defined P/WCL-*b*-P(/WCL-*co*- γ M γ BL) block copolymer (M_{h} MALLs = 31500 g/mol, D = 1.19) was synthesized. The composition of the block copolymer was determined based on the $M_{n \text{ MALLs}}$ of the PMVCL precursor as well as the $M_{n \text{ MALLs}}$ and ¹H NMR of the final copolymer (Figure S7), namely P(WCL)₉₁-b-P(WCL₁₁₉-co- $\gamma M \gamma B L_{23}$). Following the hydrolysis procedure described in the previous section, the MyBL-containing block copolymer was successfully converted into PMVCL91-b- $P(MVCL_{119}-co-HPEA_{23}).$



Figure 8. General strategy (a) and overlay of SEC traces (b) for the synthesis of the P/N/CL-*b*-P(///CL-*co*- γ M γ BL) block copolymer by OMRP. First step: [///CL]₀/[R-Co(acac)₂]₀=500/1, second step: [///CL]₀/[γ M γ BL]₀/[P/N/CL-Co(acac)₂]₀=410/90/1).

Next, the thermal response of P/WCL₉₁-*b*-P(/WCL₁₁₉-*co*- γ M γ BL₂₃) and the corresponding P/WCL₉₁-*b*-P(/WCL₁₁₉-*co*-HPEA₂₃) was investigated by turbidimetry and DLS (Figure 9). Figure 9a shows the evolution of the transmittance upon heating the aqueous solution of the P/WCL₉₁-*b*-P(/WCL₁₁₉-*co*- γ M γ BL₂₃) copolymer. In this case, the LCSTs of the P/WCL and the P(/WCL₁₁₉-*co*- γ M γ BL₂₃) segments were too close to be distinguished. Only one transition temperature appeared around 34 °C and DLS emphasized a transition from free chains to particles at this temperature (Figure 9a). Similarly to its statistical counterparts, the size of the P/WCL₉₁-*b*-P(/WCL₁₁₉-*co*- γ M γ BL₂₃) particles increased beyond the micron when the solution is further heated inducing sedimentation of the copolymer.

The thermal response of P/WCL91-b-P(/WCL119-co-HPEA23) was then studied in water at pH 6 (Figure 9b). According to the previous section, outside of a block copolymer type structure, PMCL should exhibit a phase transition around 37 °C whereas the P(NVCL119co-HPEA₂₃) in its ionized state should remain soluble at least until 80 °C. In the light of these considerations, when increasing the solution temperature above the critical transition temperature of the PMCL block, one might expect the self-assembly of the copolymer from free chains to micelles composed of a dehydrated PNVCL core stabilized by a charged P(MCL₁₁₉-co-HPEA₂₃) shell. In practice, no transition appeared around 37-40°C but a progressive decrease of transmittance was observed at 70.5 °C. This transition corresponded to the formation of particles with a diameter ranging from 440 nm at 75 °C to 520 nm at 85 °C. In this case, the T_{CP} of the PM/CL block might be shifted to higher temperature (from 34 °C to 70 °C) due to its combination with a much hydrophilic P(*N*VCL₁₁₉-*co*-HPEA₂₃) block. Nevertheless, a classical self-assembly of the block copolymer into micelles by selective dehydration of the P/WCL block at 70.5 °C is quite unlikely considering the rather large size of the objects. Instead, the PNVCL91-b-P(*IVCL*₁₁₉-*co*-HPEA₂₃) most probably follows a complex particle formation pathway upon heating with a preferential location of the P(NVCL-co-HPEA) segment at the outer region of these nanoobjects. In this respect, zeta potential analyses confirmed the presence of negative charges at the surface of the particles (from -40 to -55 mv, Table S1) which contribute to their stabilization preventing further aggregation. Again, DLS confirmed the complete dissolution of the objects upon cooling and so the reversibility of the transition phenomena.

Overall, this block copolymerization proves that $P(NVCL-co-\gamma M\gamma BL)$ and P(NVCL-co-HPEA) segments can be incorporated in more complex polymer architectures. In the future, the latter could be associated to other hydrophilic, hydrophobic, thermoresponsive or pH-sensitive sequences paving the way to unique stimuli-responsive materials.



Figure 9. Turbidimetry (black lines) and object diameter measured by DLS (red lines) in water at 5 mg/mL as a function of temperature for (a) $PNVCL_{91}$ -*b*- $P(NVCL_{119}$ -*co*- $\gamma M\gamma BL_{23}$) and (b) the corresponding hydrolyzed $PNVCL_{91}$ -*b*- $P(NVCL_{119}$ -*co*-HPEA₂₃) at pH 6. The dotted line correspond to the cooling curve.

Conclusions

In this work, we reported the successful statistical radical copolymerization of NVCL with a methylene lactone monomer, namely $\mathcal{M}\mathcal{P}BL$, which constitutes a unique and promising platform for the design of multi-responsive MVCL-based materials. A series of low dispersity NVCL/MYBL-based copolymers with different molar masses and compositions were produced via organometallic-mediated radical polymerization, a reversible deactivation radical polymerization known for its efficiency to mediate the radical polymerization of less activated monomers (LAMs), a category comprising the comonomers under investigation here. The synthesized NVCL/2M2BL-based copolymers displayed a T_{CP} in water which decreases when increasing the copolymer molar mass and □M□BL content. As post-polymerization modification, we described the successful hydrolysis of the pendant lactones leading to the corresponding ring opened hydroxy acid PNVCL-based copolymers. This chemical transformation allowed to tune the cloud point temperature of the copolymer solution over a wide range of temperatures (from 27 °C to >80 °C). The pH responsiveness of the hydroxy acid functional M/CL-based copolymer was examined in more details. At pH 6, so in its deprotonated state, the copolymer composed of 7 mol% of hydroxy carboxylate monomer units showed a T_{CP} of 61 °C and no transition was observed below 80 °C when this content was increased to 10 mol%. Upon reacidification at pH 2, the hydroxy acid-functional P/WCL solution first became opaque due to the protonation of the carboxylate groups and the formation of acid/amide H-bonds but, in sharp contrast with the previously reported acid-functional P/WCL copolymers, this solution cleared up spontaneously over time at room temperature. This phenomenon was demonstrated to be due to the relactonization reaction and the regeneration of the initial MCL/MyBL-based copolymers exhibiting the original LCST behavior. Finally, the stimuli responsiveness of a well-defined P/WCL-b-P(/WCL-co- $\gamma M \gamma BL$) block copolymer and the corresponding PMVCL-b-P(NVCL-co-HPEA) was investigated by turbidimetry. DLS and zeta potential analyses. Overall, this unusual multiresponsive lactone-functionalized PNVCL system involving a pH dependent and reversible delactonization reaction constitutes a promising platform for tuning the solution properties of PMCL, and possibly of other smart polymer sequences, opening up new prospects and innovation possibilities.

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