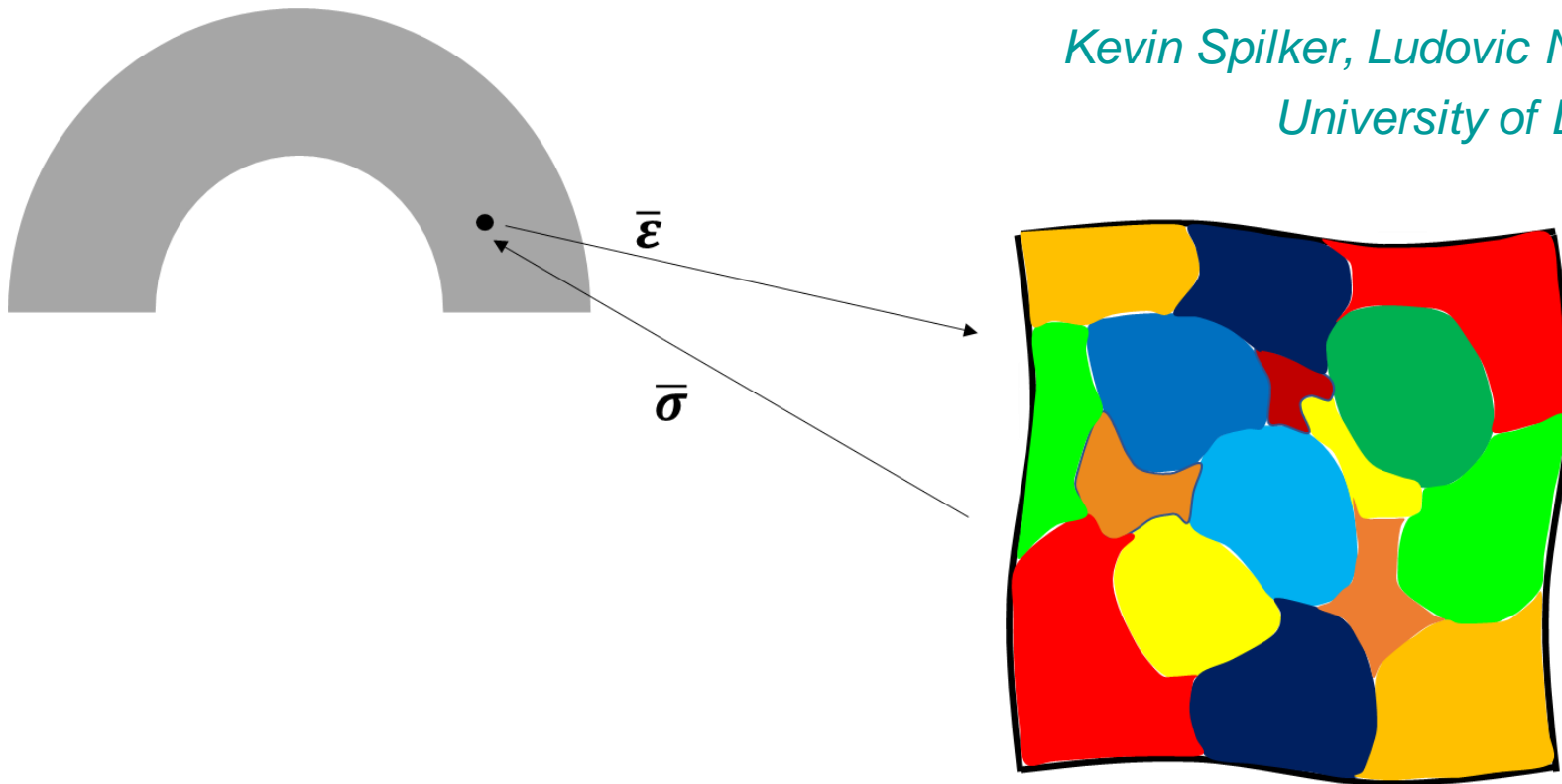


Multiscale Modeling of Composites – Piecewise-Uniform Model Order Reduction

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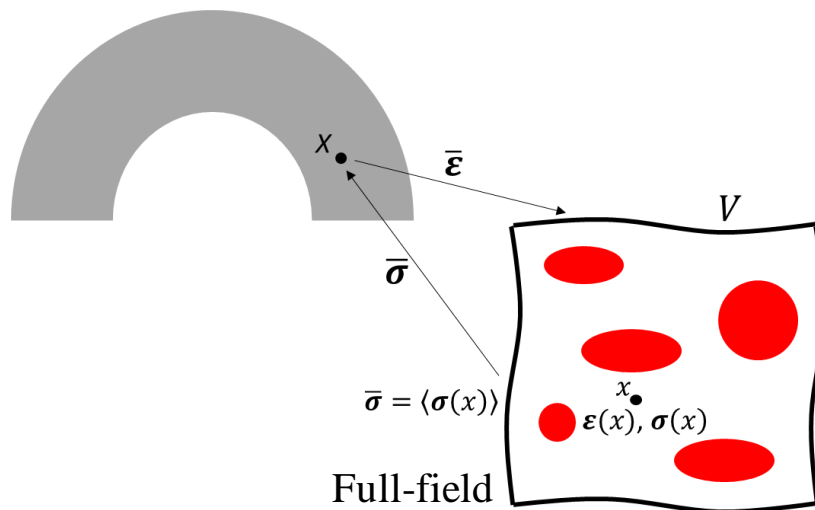


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Homogenization of the mechanics of composite materials

- Strategy: finding a representative volume element (RVE)
- RVE represents a macroscopic point X , contains the micro-structure
- Aim: achieving macroscopic/effective RVE response $\bar{\epsilon} \rightarrow \bar{\sigma}$ (BVP)
- Use of bounds, mean-field homogenization \rightarrow only need for phase properties and volume fractions
- Two-scale modeling: Linking microscopic and macroscopic scales: $x \leftrightarrow X$

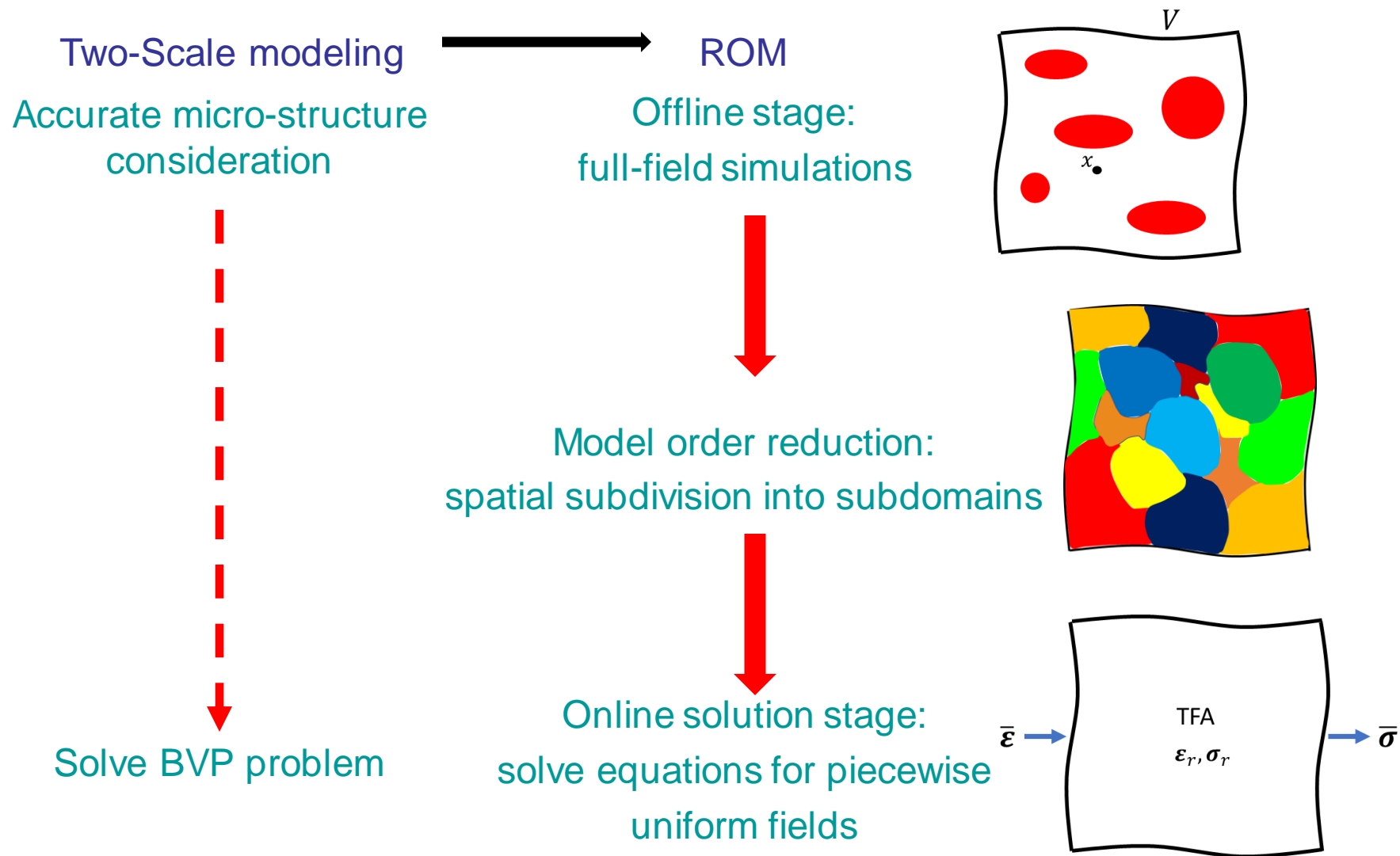


Taking into account micro-structure and microscopic effects

macro-response by averaging over the RVE
 \rightarrow computational effort is immense

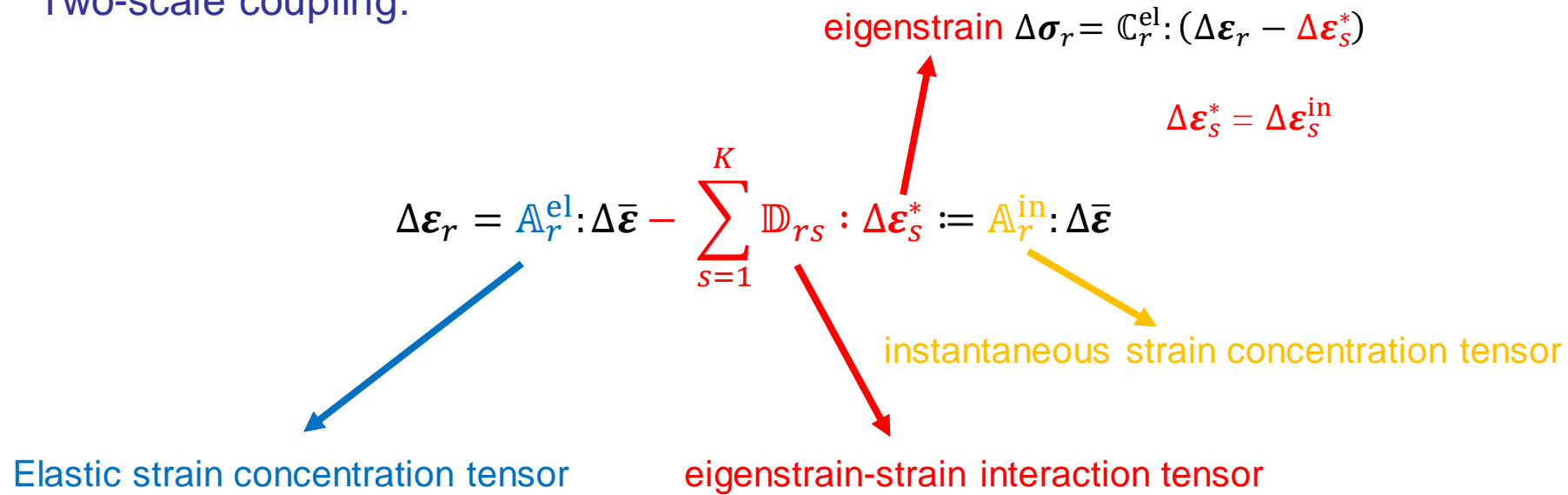
Two-scale reduced order modeling: Uniform fields

Solution strategy



Transformation Field Analysis (TFA)

- Two-scale coupling:



$$\Delta \bar{\boldsymbol{\varepsilon}} = \sum_{r=1}^K v_r \Delta \boldsymbol{\varepsilon}_r$$

- Determination of \mathbb{A}_r^{el} for each subdomain and \mathbb{D}_{rs} between all subdomains in the offline stage

Offline stage

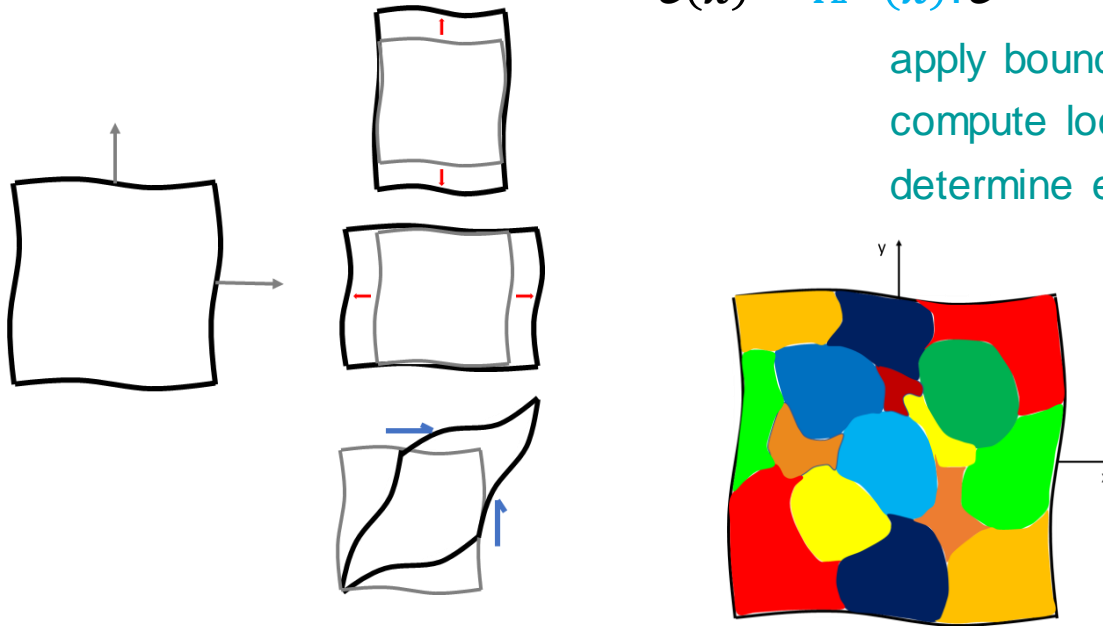
- Microscopic elastic strain concentration tensors

$$\boldsymbol{\varepsilon}(\boldsymbol{x}) = \mathbb{A}^{\text{el}}(\boldsymbol{x}) : \bar{\boldsymbol{\varepsilon}}$$

apply boundary conditions $\bar{\boldsymbol{\varepsilon}}$

compute local strains $\boldsymbol{\varepsilon}(\boldsymbol{x})$

determine entries of $\mathbb{A}^{\text{el}}(\boldsymbol{x})$



Spatial decomposition based on $\mathbb{A}^{\text{el}}(\boldsymbol{x})$

$\rightarrow \mathbb{A}_r^{\text{el}}$

- Eigenstrain – strain interaction tensors

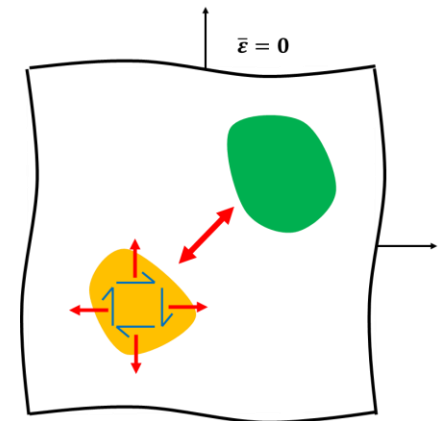
- With $\bar{\boldsymbol{\varepsilon}} = \mathbf{0}$ and eigenstrain in one subdomain

$$\boldsymbol{\varepsilon}_r = \mathbb{D}_{rs} : \boldsymbol{\varepsilon}_s^*$$

apply eigenstrain $\boldsymbol{\varepsilon}_s^*$

compute $\boldsymbol{\varepsilon}_r$

determine entries of \mathbb{D}_{rs}

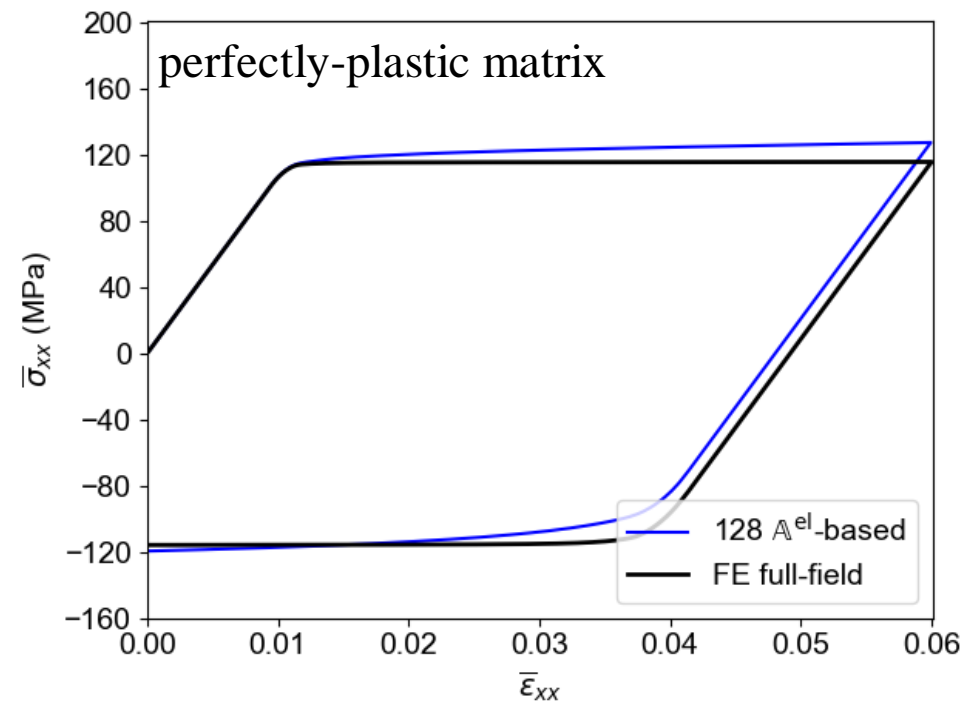


Results

- TFA issue: not well-captured inelastic fields using elasticity-based subdomains
- Under-represented inelastic fields \longleftrightarrow under-represented interaction effects
 - Incorrect strain distributions over the subvolumes
 - Especially: overestimated strain accumulation in stiff material phase

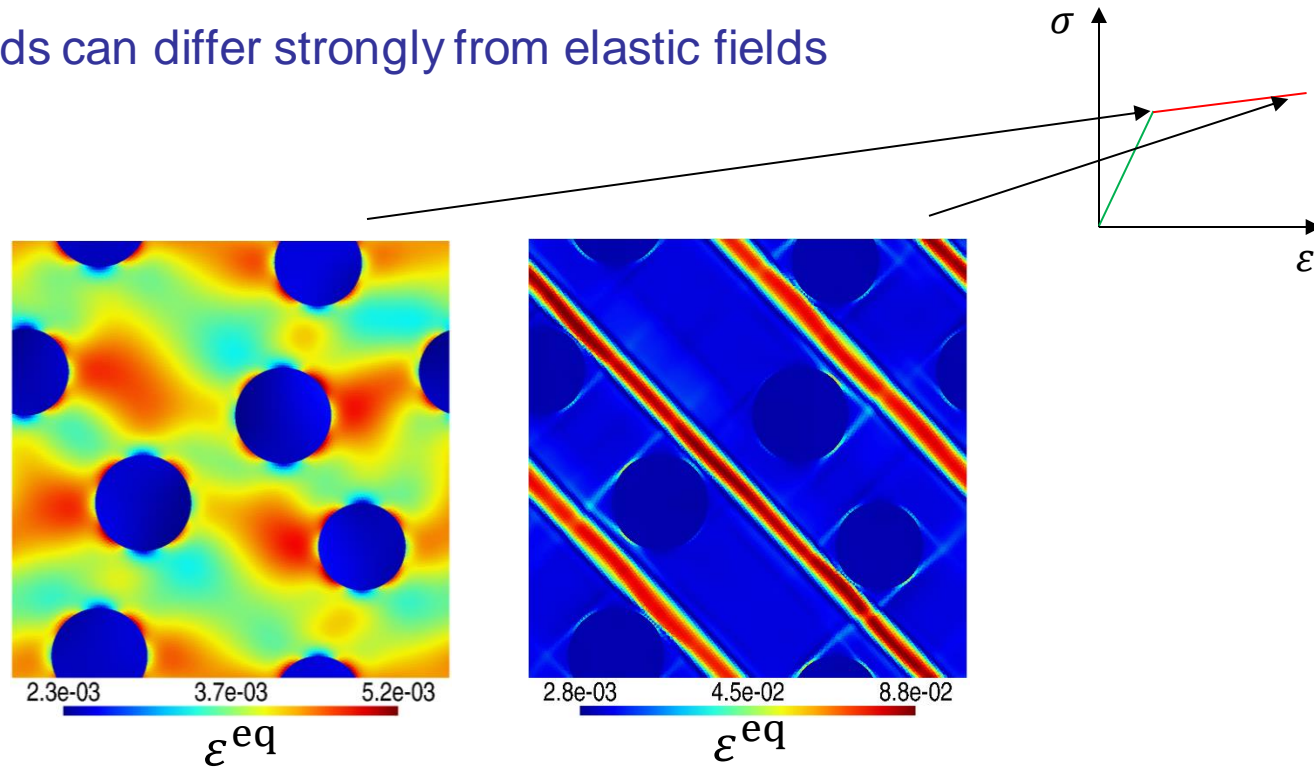
→ Over-stiff composite overall response

Particularly in cases of
highly-localized plasticity or
highly heterogeneous plastic fields



TFA: Integration of inelasticity

- Inelastic fields can differ strongly from elastic fields



→ Change of plan: cluster decomposition based on inelastic micro-fields to respect inelastic micromechanical patterns

→ include more information of the real physics into the two-scale modeling approach

- Evident: well-chosen boundary conditions in order to capture different possible shear bands which can occur in the online stage
- Selection based on low-fidelity FE simulations:
compare plastic fields under various complex loading conditions and under several proportional loadings
- Chosen four monotonic strain boundary conditions

$$(1) \bar{\varepsilon} = \frac{1}{2} E (e_1 \otimes e_1 - e_2 \otimes e_2) \quad \text{bi-axial isochoric}$$

$$(2) \bar{\varepsilon} = \frac{1}{2} E (e_1 \otimes e_2 + e_2 \otimes e_1) \quad \text{pure shear}$$

$$(3) \bar{\varepsilon} = (1) + \frac{1}{2} (2) \quad \text{mixed isochoric}$$

$$(4) \bar{\varepsilon} = \frac{1}{2} (1) + (2) \quad \text{mixed isochoric}$$

→ Ability to detect different kinds of evolving shear bands

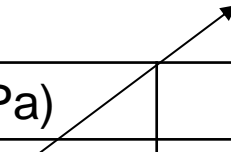
A numerical application

- Material system: elastic inclusions in a J2-plastic matrix

$$\sigma^Y = \sigma^{Y0} + R \quad \text{yield stress}$$

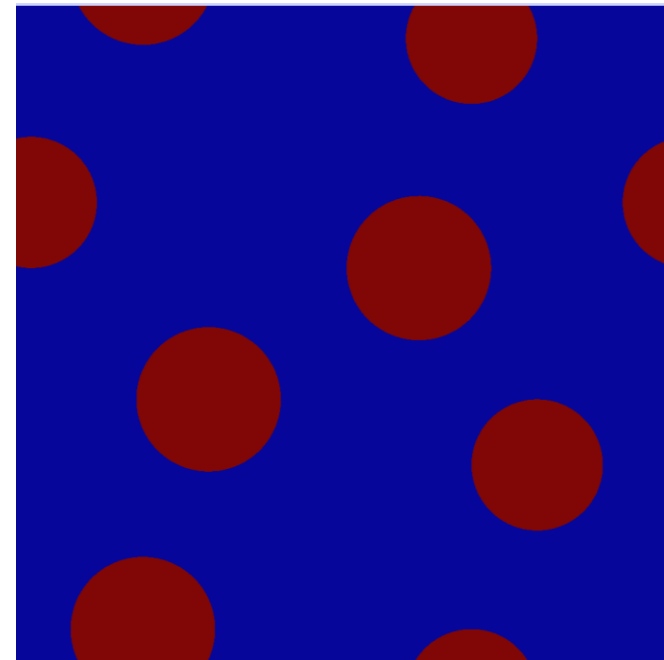
$$R = Hp^m \quad \text{hardening stress}$$

Perfect plasticity



	G (GPa)	K (GPa)	v (-)	H (MPa)	m (-)
Matrix	3	10	0.36	~ 0	1
Inclusions	6	20	0.36	-	-

- 2D periodic RVE micro-structure:
random distribution of circular inclusions
- Inclusion volume fraction: 21 %



- perfectly-plastic matrix behavior

$$(1) \bar{\varepsilon} = \frac{1}{2} E (e_1 \otimes e_1 - e_2 \otimes e_2)$$

bi-axial isochoric

$$(2) \bar{\varepsilon} = \frac{1}{2} E (e_1 \otimes e_2 + e_2 \otimes e_1)$$

pure shear

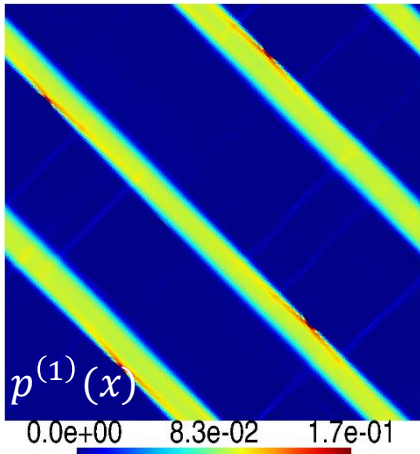
$$(3) \bar{\varepsilon} = (1) + \frac{1}{2} (2)$$

mixed isochoric

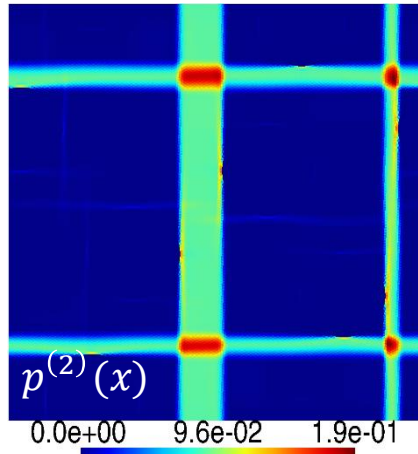
$$(4) \bar{\varepsilon} = \frac{1}{2} (1) + (2)$$

mixed isochoric

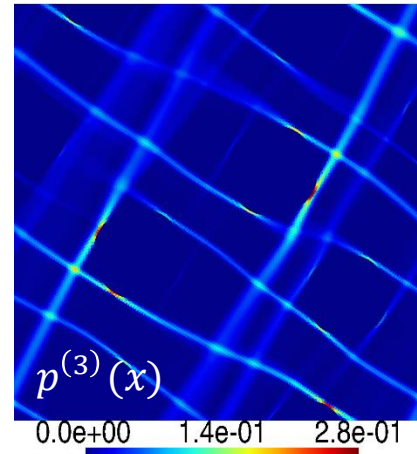
(1)



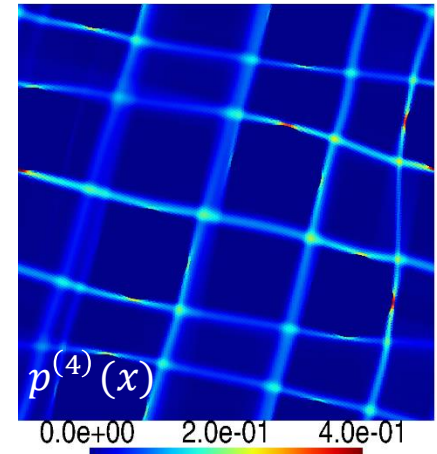
(2)



(3)



(4)



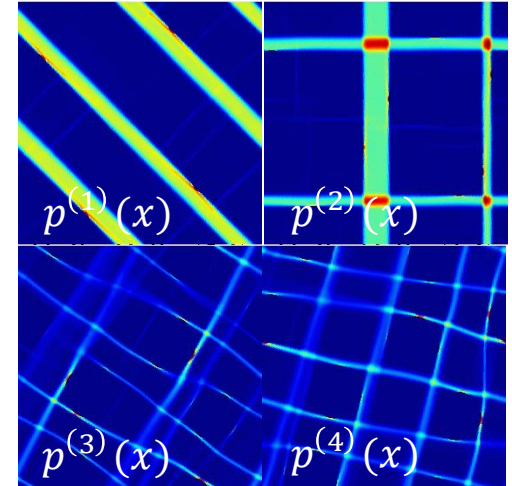
- Weighting factors: $w^{(1)} = 10, w^{(2)} = 10, w^{(3)} = 1, w^{(4)} = 1$

- Normalized, weighted plastic fields

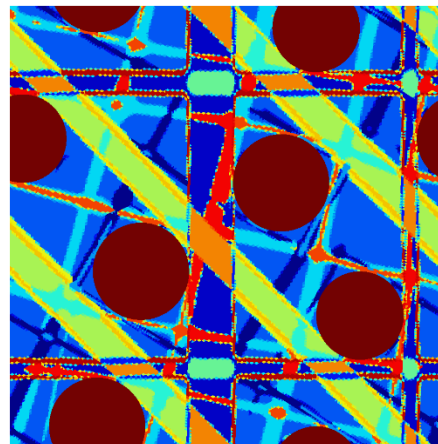
$$q^{(l)}(x) = w^{(l)} \frac{p^{(l)}(x)}{\bar{p}^{(l)}} \quad l \in 1, \dots, 4$$

$$\rightarrow \mathbf{q}(x) = (q^{(1)}(x), q^{(2)}(x), q^{(3)}(x), q^{(4)}(x))^T$$

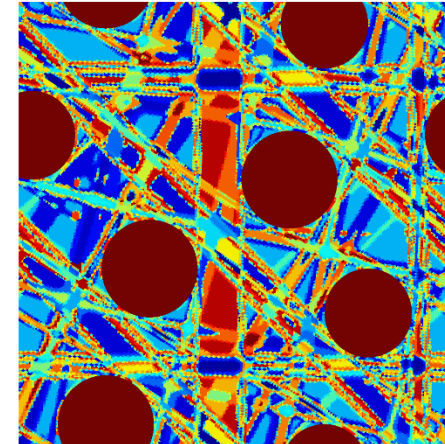
- K-means clustering based on similarity of local field $\mathbf{q}(x)$



→
resulting subdomains



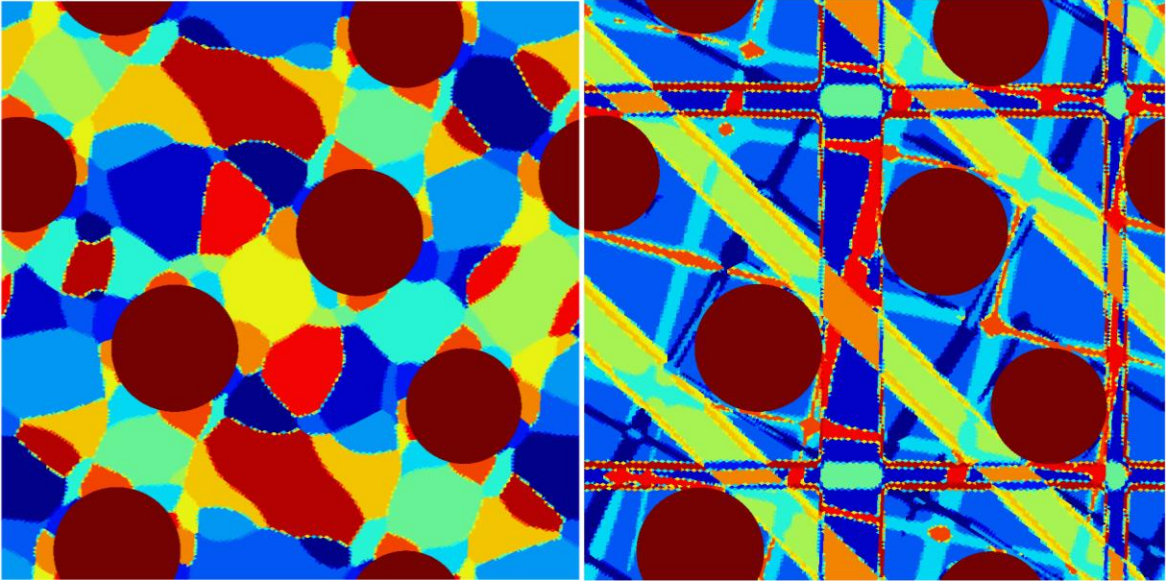
16



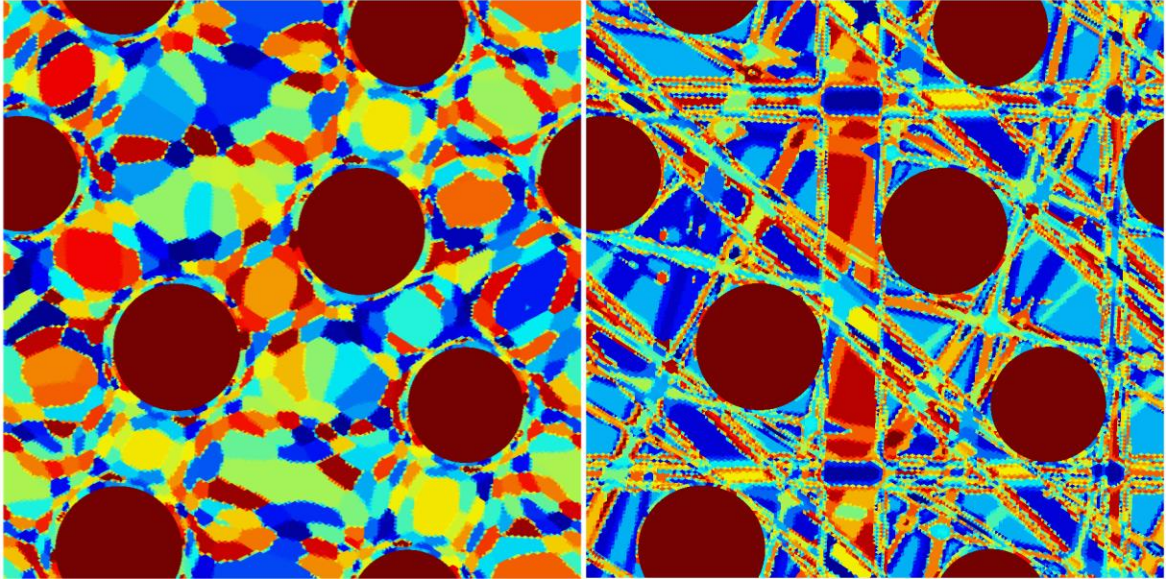
128

Offline stage: Application

16



128

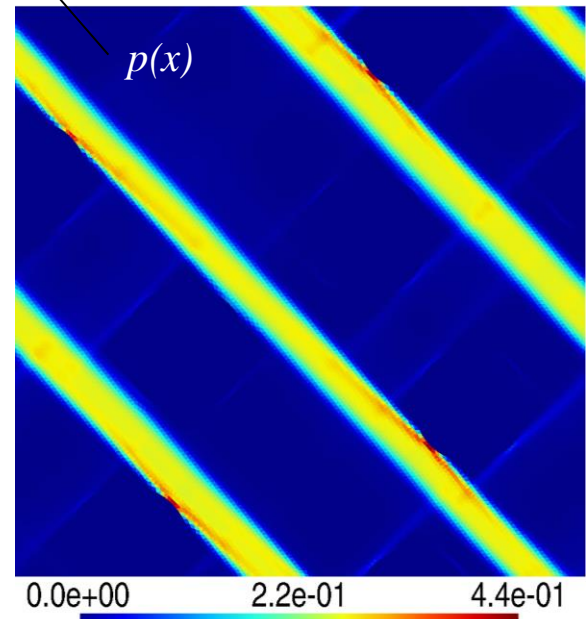
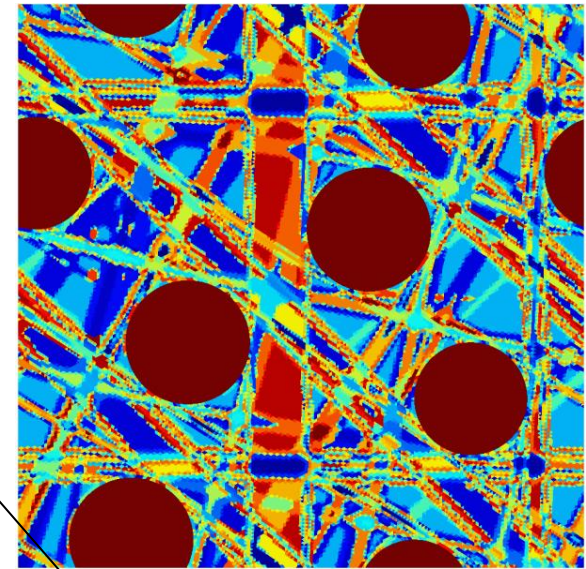
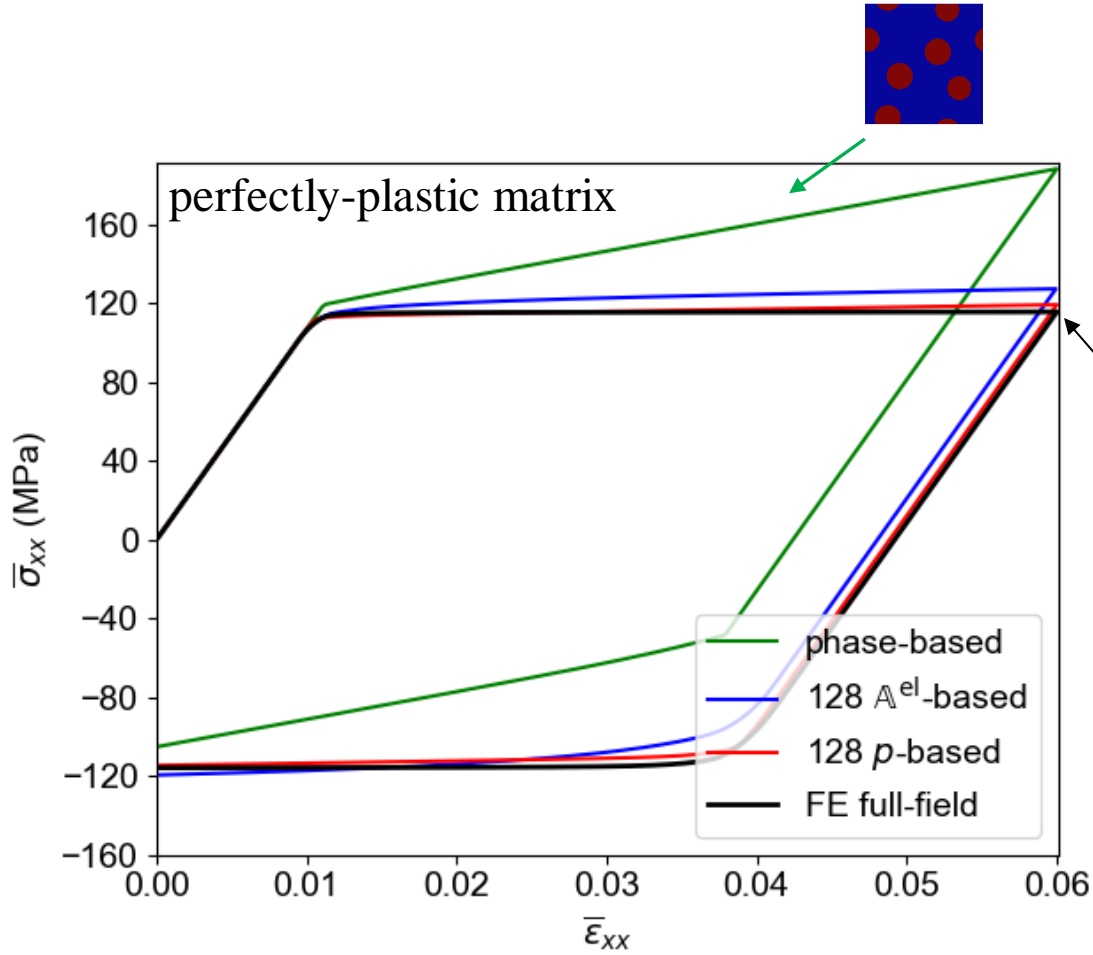


Elasticity

Plasticity

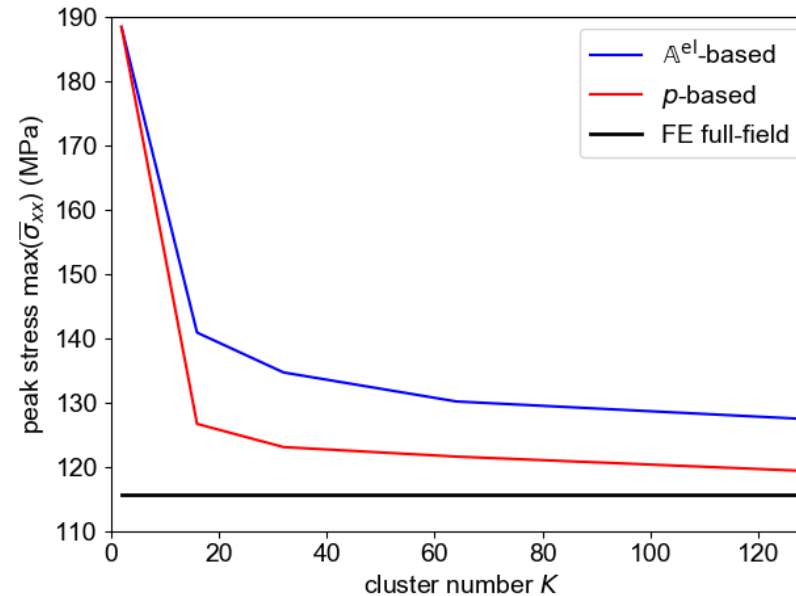


Results: Uniaxial tension



Results

- Plasticity-based clustering allows more accurate modeling than elasticity-based clustering



- Clearly faster convergence to full-field result
- Drawbacks:
 - Inelastic simulations with fine mesh required in offline stage
 - Integration of many more loading cases in offline stage → interfering patterns



Wallonie

- Transformation Field Analysis:

George J. Dvorak (1992), Transformation Field Analysis of Inelastic Composite Materials, *Proceedings: Mathematical and Physical Sciences*, 437: 311-327

Pierre Suquet (1997), *Continuum Micromechanics*, Springer