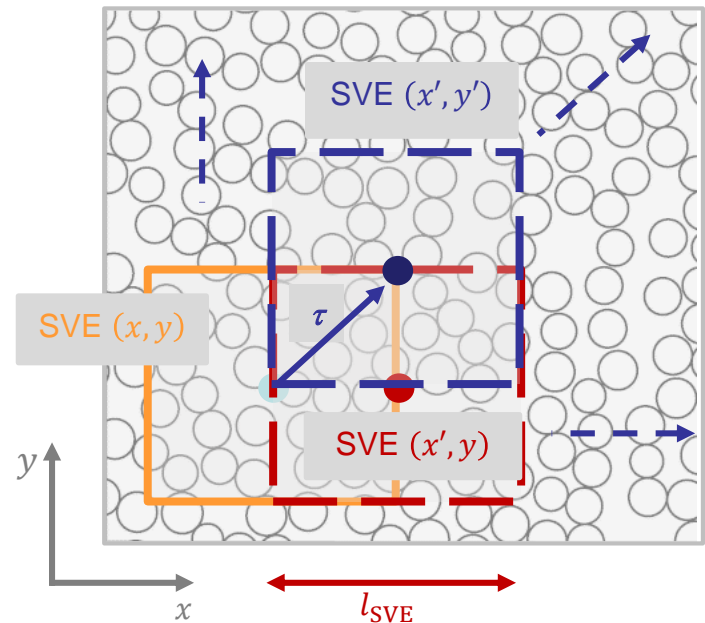
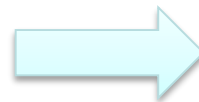
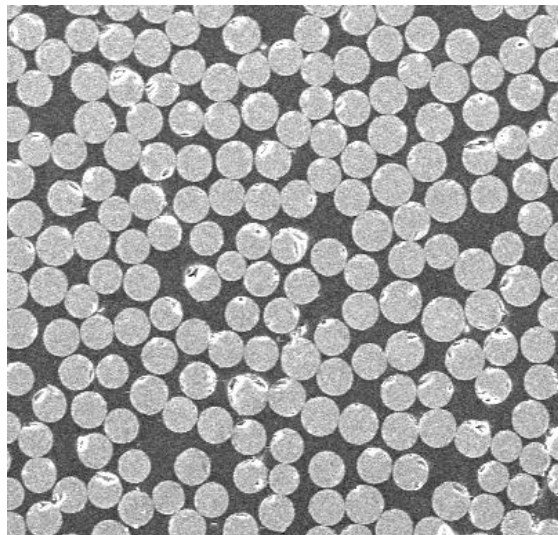
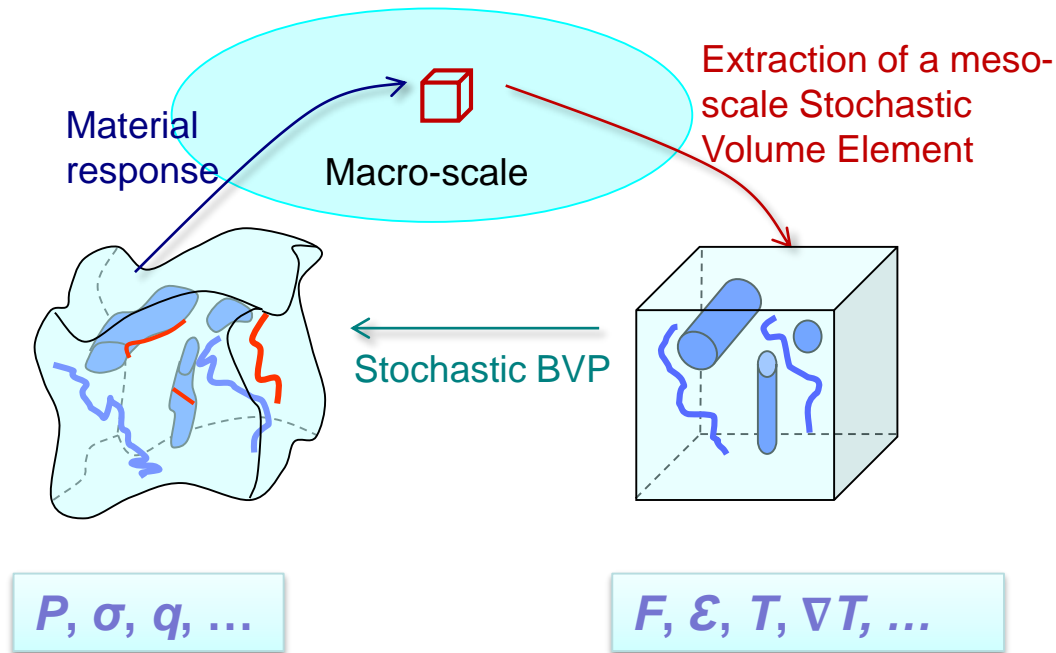


# Multiscale stochastic simulations using a MFH model constructed from full-field SVE realizations



# Objectives

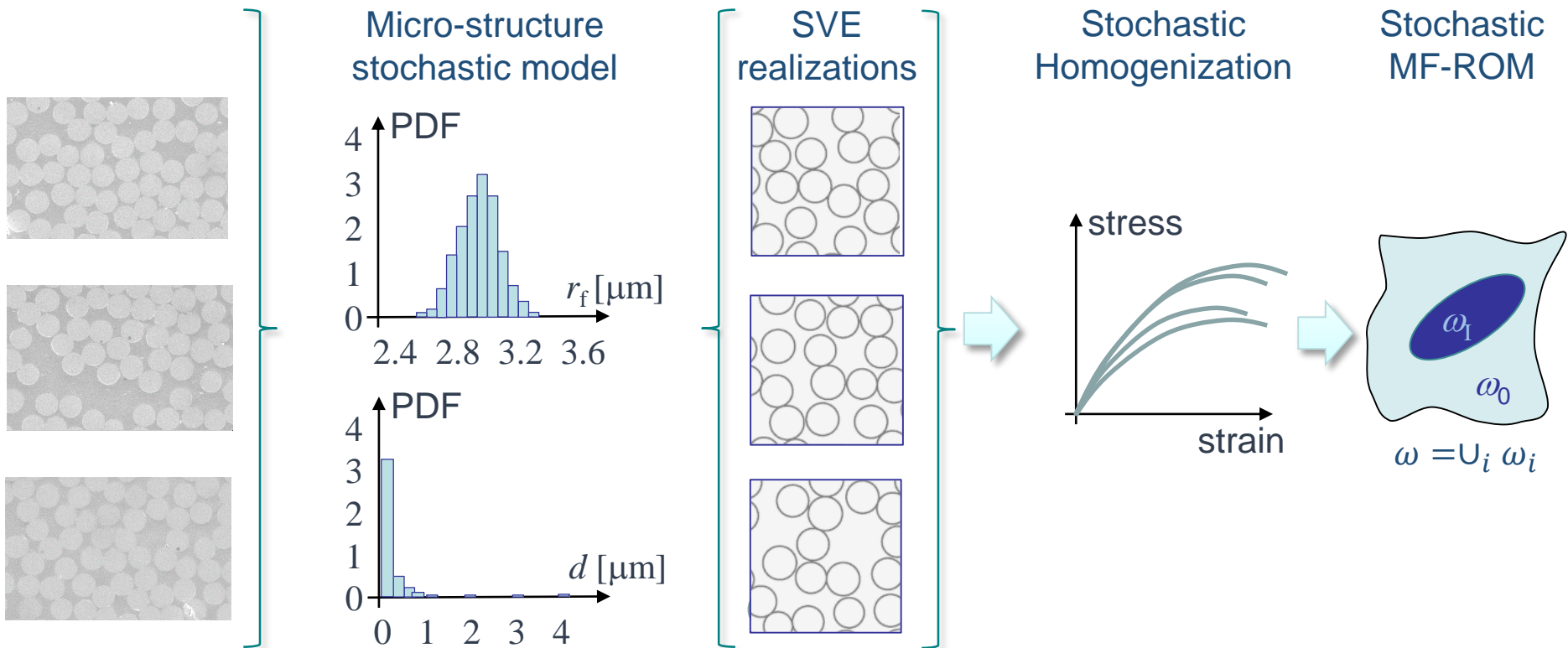
- Main objective
  - To develop an integrated stochastic multiscale approach to predict failure of composites



- Main deliverable
  - Development of a modelling methodology able to account for micro-scale uncertainties

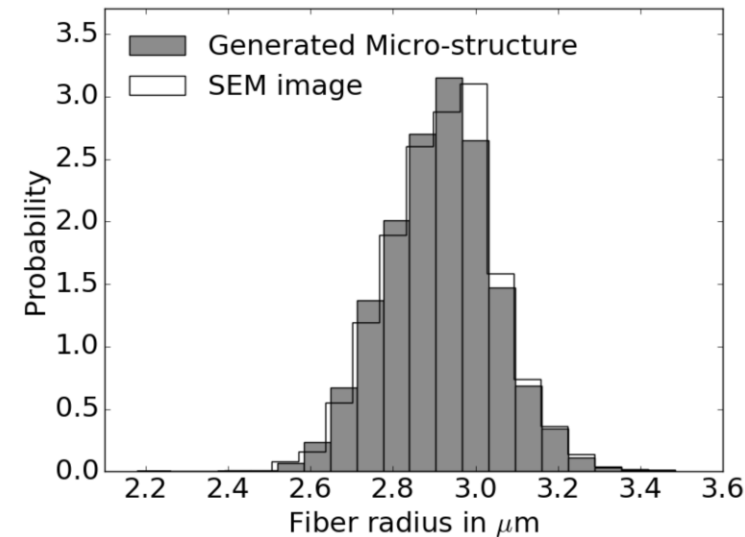
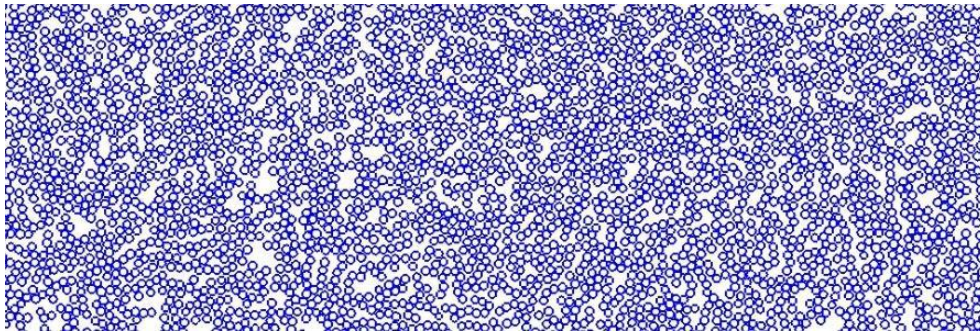
# Methodology

- Proposed methodology for material system:
  - To develop a stochastic Mean Field Homogenization method able to predict the probabilistic distribution of material response at an intermediate scale from micro-structural constituents characterization

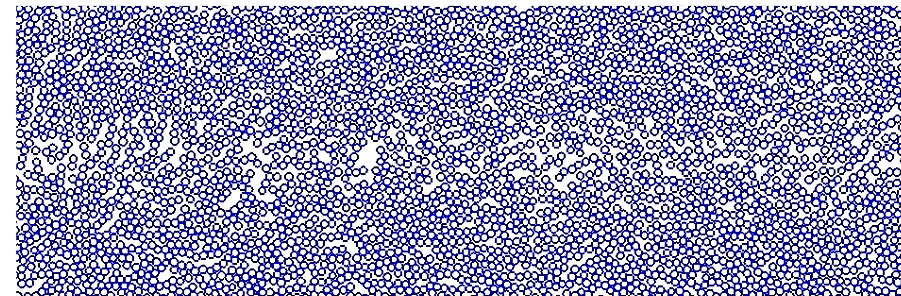
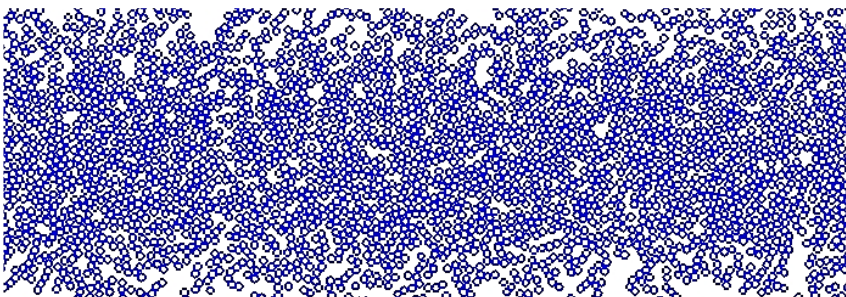


# Micro-structure stochastic model

- Numerical micro-structures are generated by a fiber additive process
  - Arbitrary size
  - Arbitrary number



- Possibility to generate non-homogenous distributions



# Stochastic Mean-Field Homogenization

- Mean-Field-Homogenization (MFH)

- Linear composites

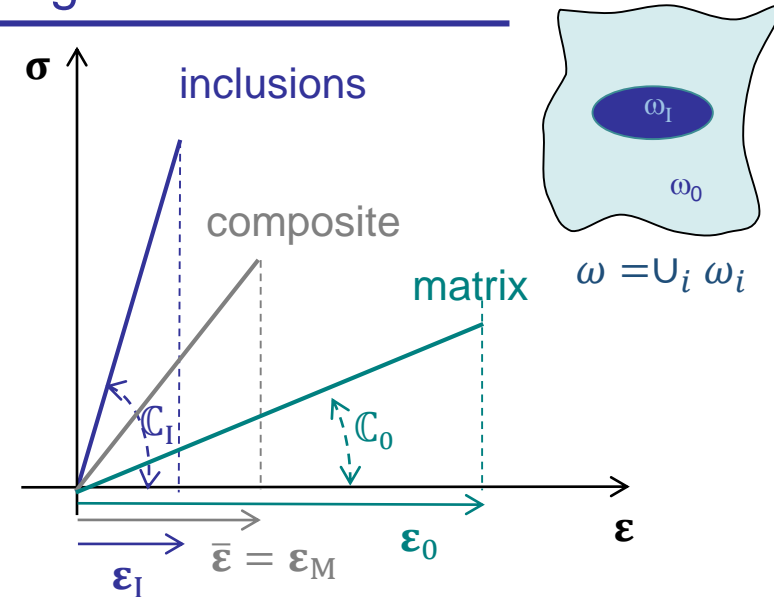
$$\left\{ \begin{array}{l} \boldsymbol{\sigma}_M = \bar{\boldsymbol{\sigma}} = \nu_0 \boldsymbol{\sigma}_0 + \nu_I \boldsymbol{\sigma}_I \\ \boldsymbol{\varepsilon}_M = \bar{\boldsymbol{\varepsilon}} = \nu_0 \boldsymbol{\varepsilon}_0 + \nu_I \boldsymbol{\varepsilon}_I \\ \boldsymbol{\varepsilon}_I = \mathbb{B}^\varepsilon(I, \mathbb{C}_0, \mathbb{C}_I) : \boldsymbol{\varepsilon}_0 \end{array} \right.$$

→  $\hat{\mathbb{C}}_M = \hat{\mathbb{C}}_M(I, \mathbb{C}_0, \mathbb{C}_I, \nu_I)$

- We use Mori-Tanaka assumption for  $\mathbb{B}^\varepsilon(I, \mathbb{C}_0, \mathbb{C}_I)$

- Stochastic MFH

- How to define random vectors  $\mathcal{V}_{MT}$  of  $I, \mathbb{C}_0, \mathbb{C}_I, \nu_I$  ?



# Stochastic Mean-Field Homogenization

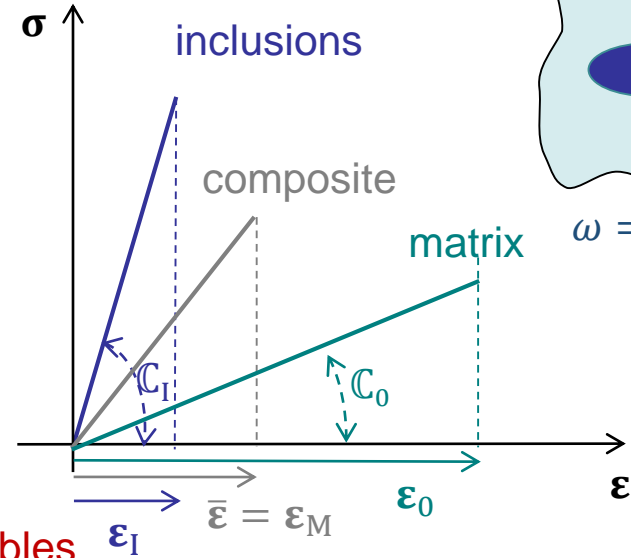
- Mean-Field-Homogenization (MFH)

- Linear composites

$$\left\{ \begin{array}{l} \sigma_M = \bar{\sigma} = v_0 \sigma_0 + v_1 \sigma_1 \\ \varepsilon_M = \bar{\varepsilon} = v_0 \varepsilon_0 + v_1 \varepsilon_1 \\ \varepsilon_1 = \mathbb{B}^\varepsilon(I, C_0, C_I) : \varepsilon_0 \end{array} \right.$$

$$\hat{C}_M = \hat{C}_M(I, C_0, C_I, v_I)$$

Defined as random variables



- Consider an equivalent system

- For each SVE realization  $i$ :

→  $C_M$  and  $v_I$  known

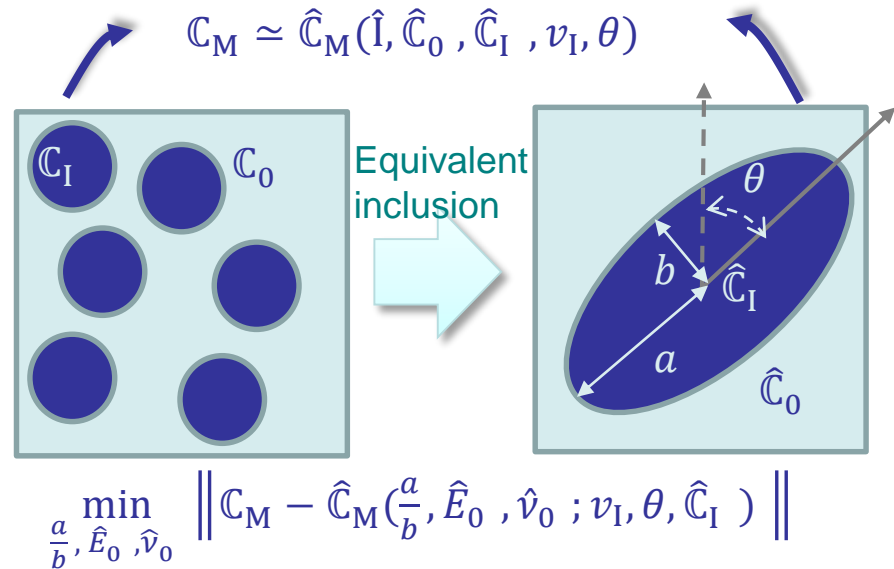
- Anisotropy from  $C_M^i$

→  $\theta$  is evaluated

- Fiber behavior uniform

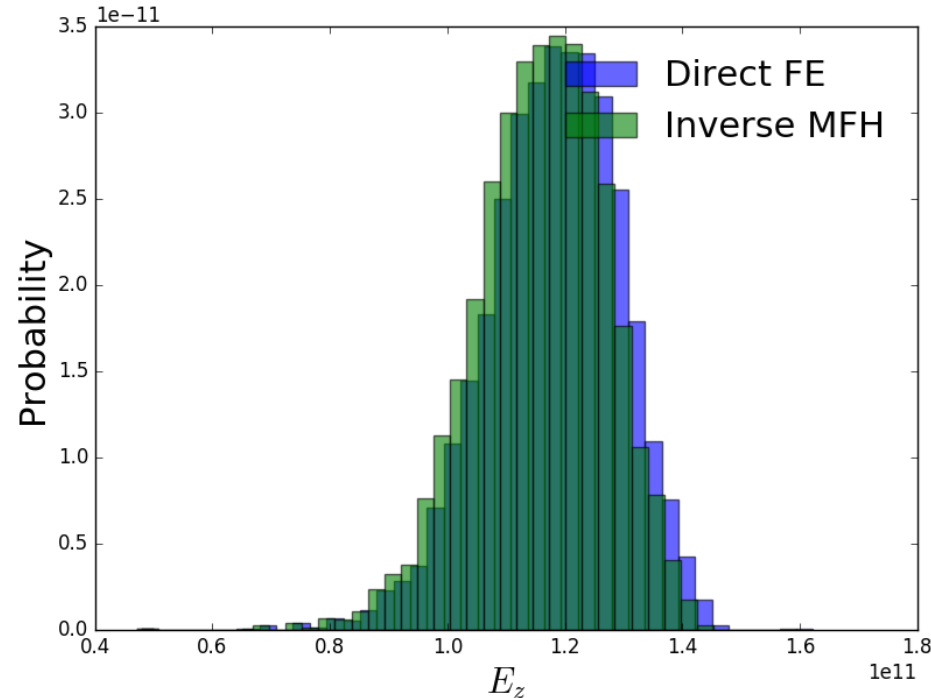
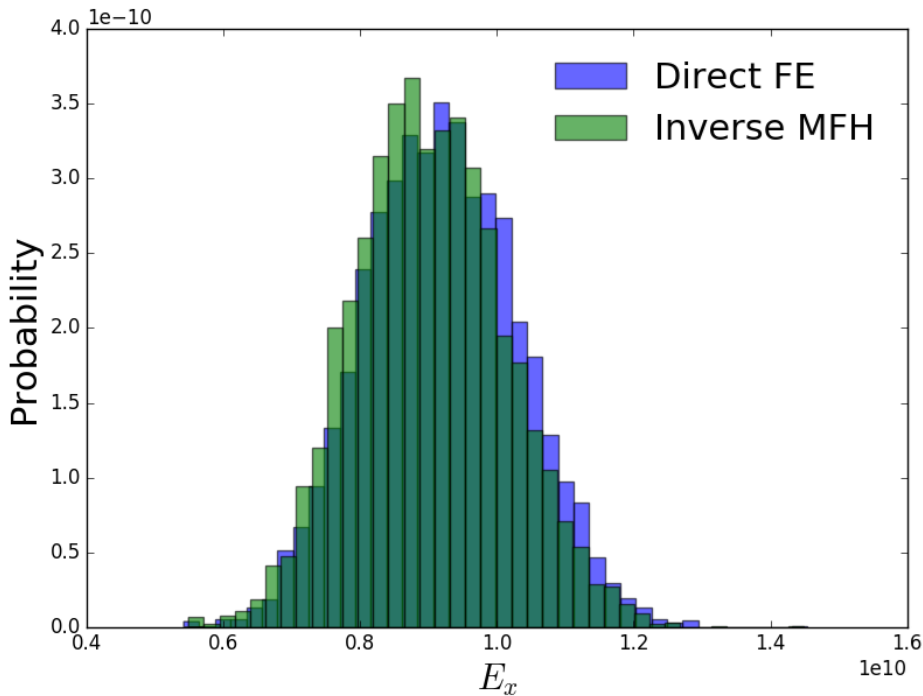
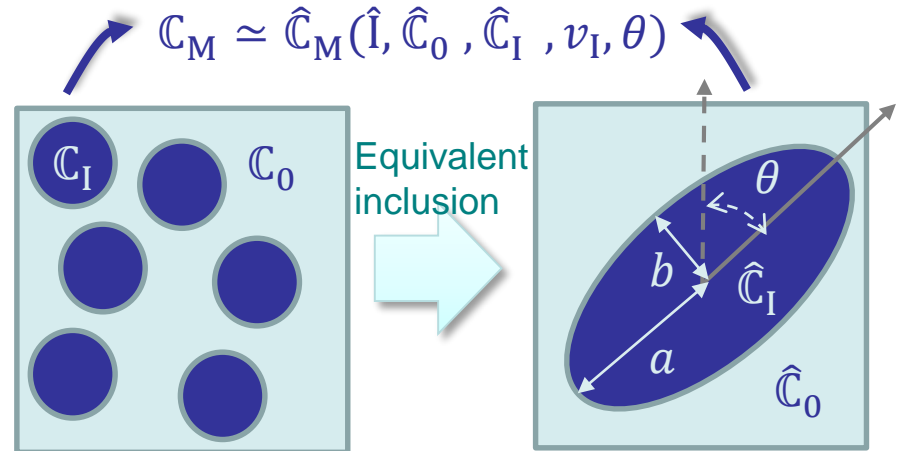
→  $\hat{C}_I$  for one SVE

- Remaining optimization problem:



# Stochastic Mean-Field Homogenization

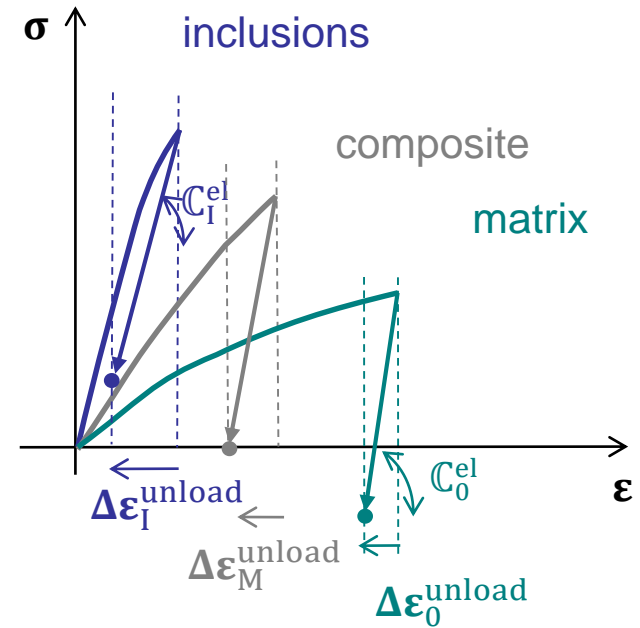
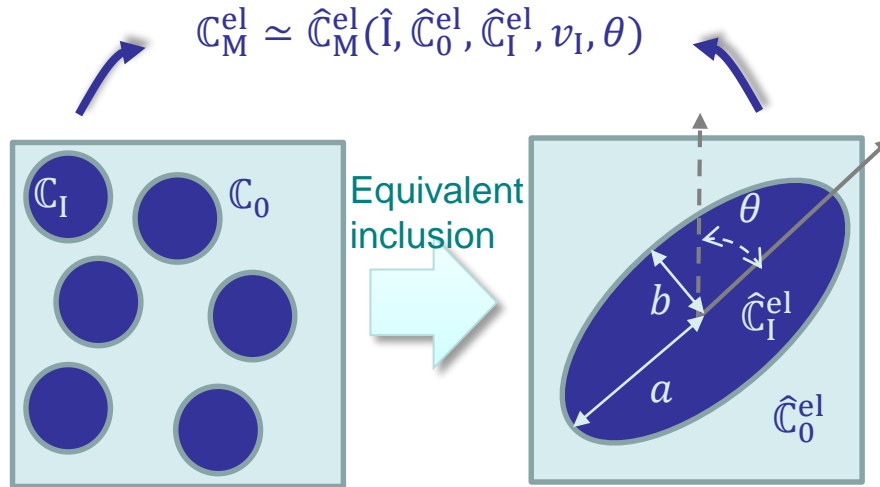
- Inverse stochastic identification
  - Comparison of homogenized properties from SVE realizations and stochastic MFH



L. Wu, V.-D. Nguyen, L. Adam, L. Noels, An inverse micro-mechanical analysis toward the stochastic homogenization of nonlinear random composites (2019)

# Non-linear stochastic Mean-Field Homogenization

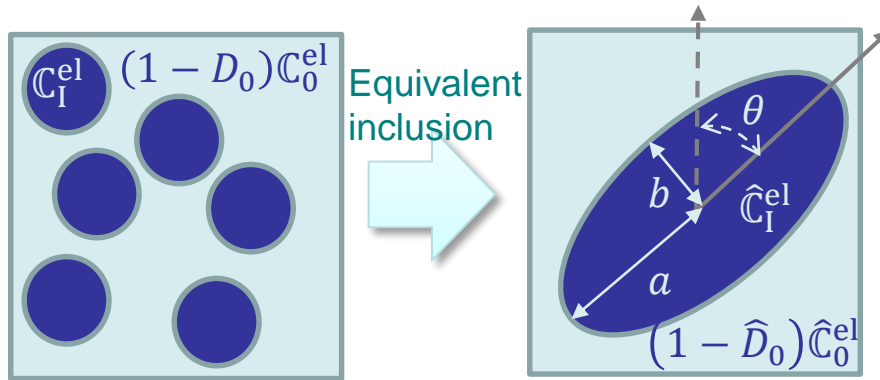
- Damage-enhanced inverse identification
  - First step from elastic response
    - Before damage occurs





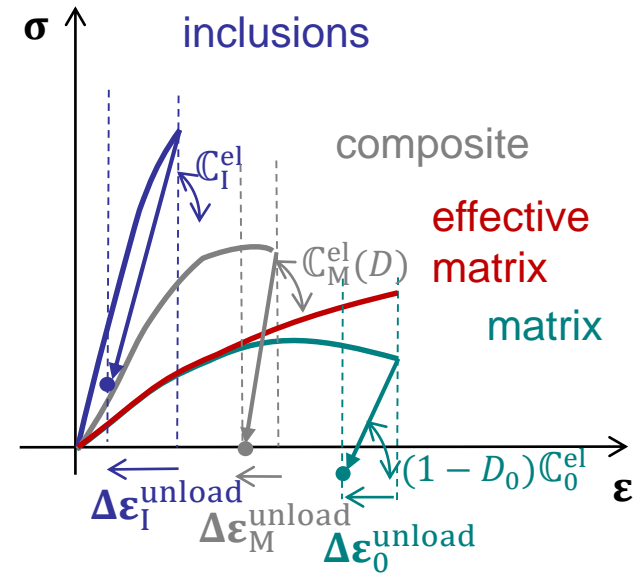
# Non-linear stochastic Mean-Field Homogenization

- Damage-enhanced inverse identification
  - Second step: elastic unloading



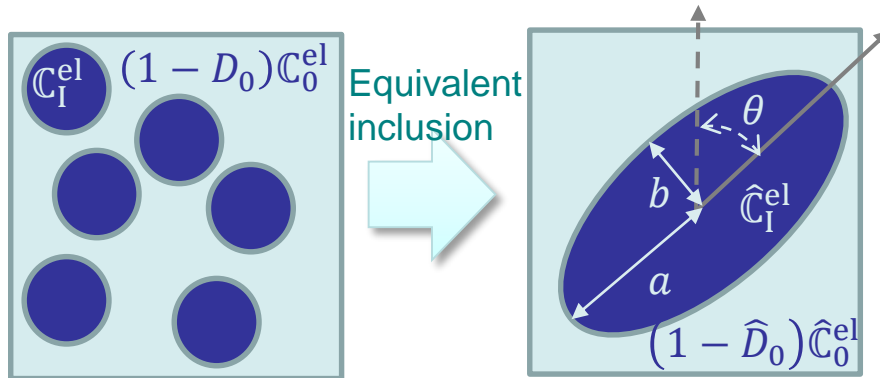
- Identify damage evolution  $\hat{D}_0$

$$\mathbb{C}_M^{el}(D) \simeq \hat{\mathbb{C}}_M^{el}(\hat{I}, (1 - \hat{D}_0)\hat{\mathbb{C}}_0^{el}, \hat{\mathbb{C}}_I^{el}, \nu_I, \theta)$$



# Non-linear stochastic Mean-Field Homogenization

- Damage-enhanced inverse identification
  - Second step: elastic unloading



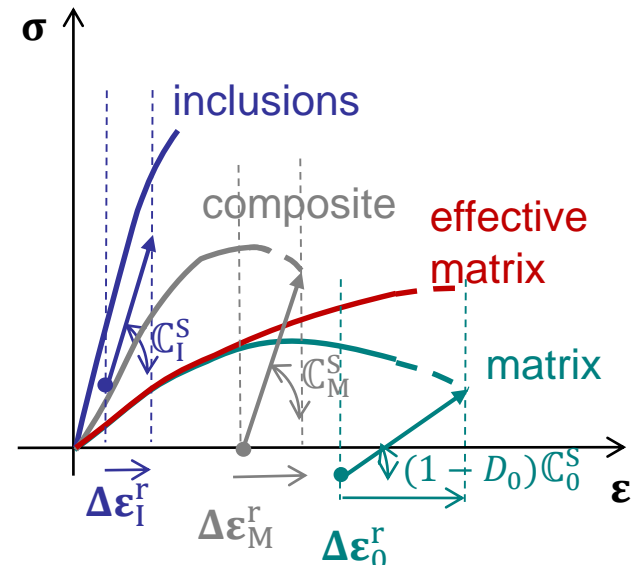
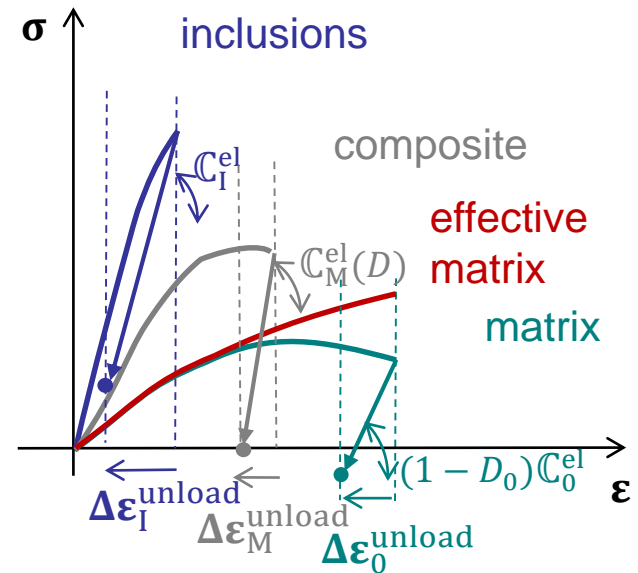
- Identify damage evolution  $\hat{D}_0$

$$\mathbb{C}_M^{el}(D) \simeq \hat{\mathbb{C}}_M^{el}(\hat{I}, (1 - \hat{D}_0)\hat{\mathbb{C}}_0^{el}, \hat{\mathbb{C}}_I^{el}, \nu_I, \theta)$$

- Third step from the LCC

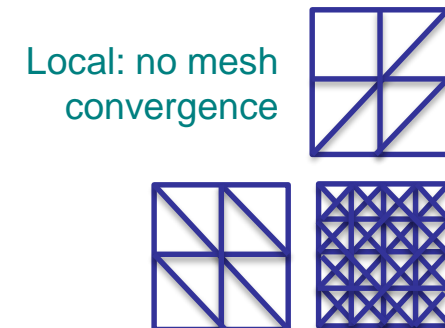
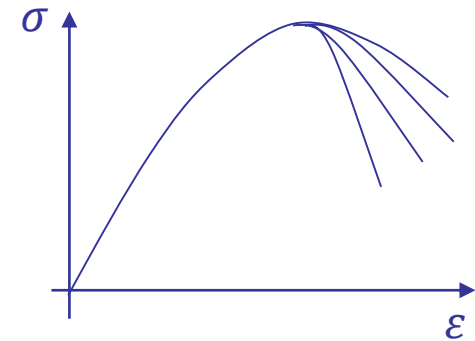
- $\Delta\sigma_M = \mathbb{C}_M^S(I, (1 - D_0)\mathbb{C}_0^S, \mathbb{C}_I^S, \nu_I) : \Delta\varepsilon_M^r$
- Extract the equivalent hardening  $\hat{R}(\hat{p}_0)$  & damage evolution  $\hat{D}_0(\hat{p}_0)$  from incremental secant tensor:

$$(1 - D_0)\mathbb{C}_0^S \simeq (1 - \hat{D}_0(\hat{p}_0))\hat{\mathbb{C}}_0^S(\hat{R}(\hat{p}_0); \hat{\mathbb{C}}_0^{el})$$



# Micro-structural simulation of fiber-reinforced highly crosslinked epoxy

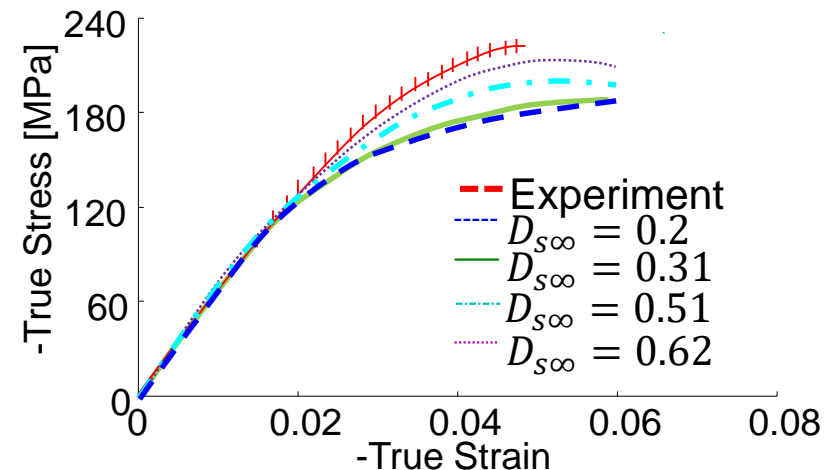
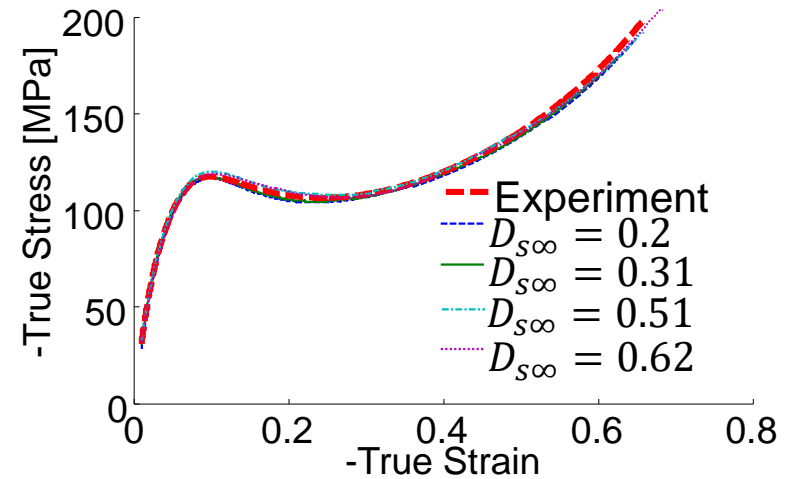
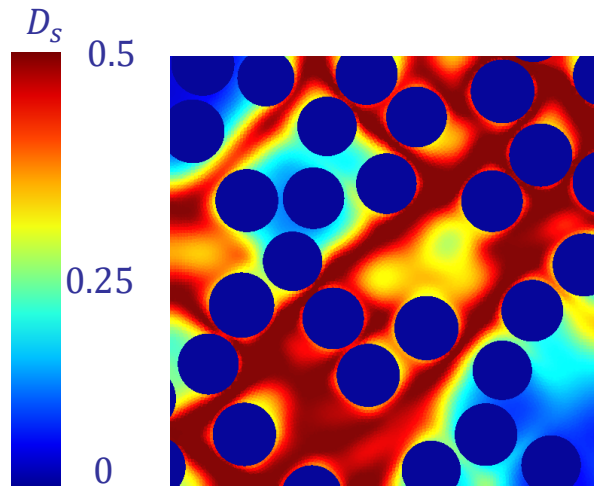
- UD Composites with RTM6 epoxy matrix
  - Identified matrix material behaviour
    - Hyperelastic viscoelastic-viscoplastic constitutive model enhanced by a multi-mechanism nonlocal damage model
      - To be used in micro-structural analysis
        - » Behaviour in composite is different
        - » Introduce a length-scale effect
- Resin model implementation:
  - Requires non-local form [Bažant 1988]
    - Introduction of characteristic length  $l_c$
    - Weighted average:  $\tilde{Z}(\mathbf{x}) = \int_{V_c} W(\mathbf{y}; \mathbf{x}, l_c) Z(\mathbf{y}) d\mathbf{y}$
  - Implicit form [Peerlings et al. 1998]
    - New degrees of freedom:  $\tilde{Z}$
    - New Helmholtz-type equations:  $\tilde{Z} - l_c^2 \Delta \tilde{Z} = Z$



# Micro-structural simulation of fiber-reinforced highly crosslinked epoxy

- UD Composites with RTM6 epoxy matrix
  - Identified matrix material behaviour
    - Hyperelastic viscoelastic-viscoplastic constitutive model enhanced by a multi-mechanism nonlocal damage model
- Resin model: saturated softening
  - Damage evolution

$$\left\{ \begin{array}{l} \dot{D}_s = H_s (\chi_s - \chi_{s0})^{\zeta_s} (D_{s\infty} - D_s) \dot{\chi}_s \\ \chi_s = \max_{\tau} (\chi_{s0}, \tilde{\gamma}_s(\tau)) \\ \tilde{\gamma}_s - l_s^2 \Delta \tilde{\gamma}_s = \gamma \end{array} \right.$$



V.-D. Nguyen, L. Wu, L. Noels, A micro-mechanical model of reinforced polymer failure with length scale effects and predictive capabilities. Validation on carbon fiber reinforced high-crosslinked RTM6 epoxy resin (2019)

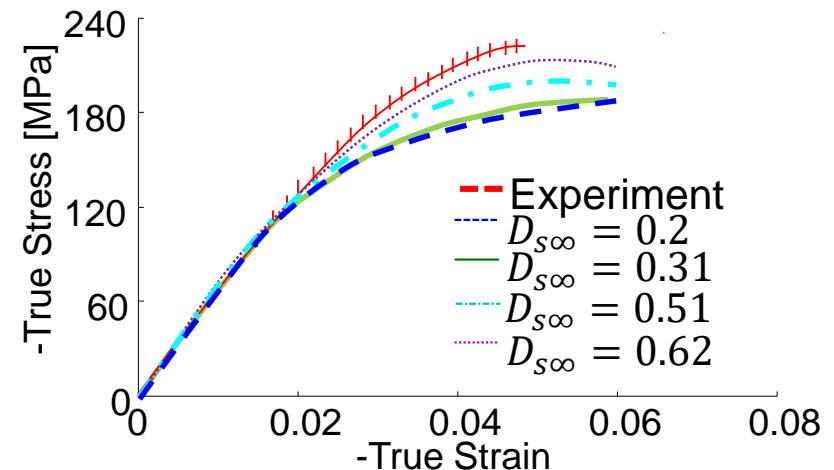
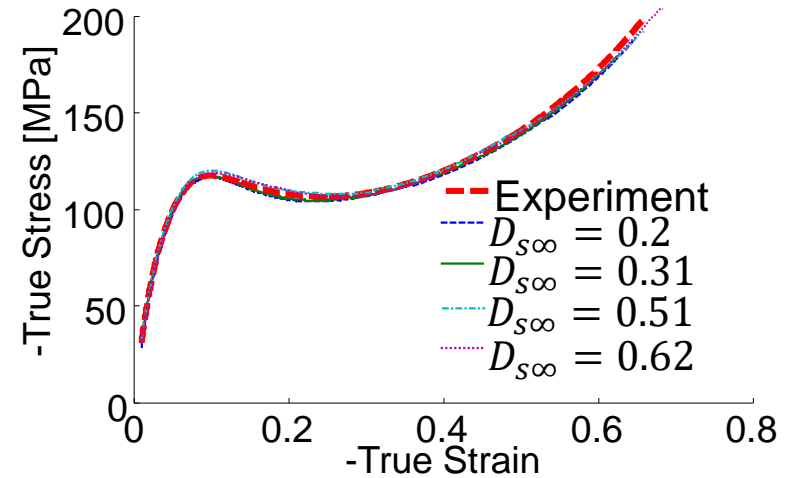
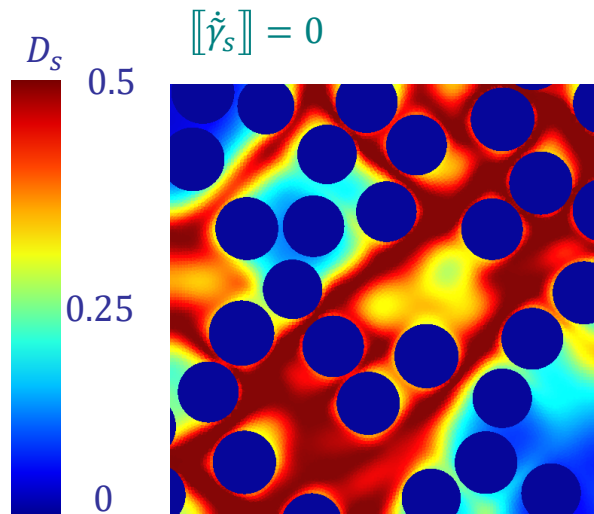
# Micro-structural simulation of fiber-reinforced highly crosslinked epoxy

- UD Composites with RTM6 epoxy matrix

- Identified matrix material behaviour
  - Hyperelastic viscoelastic-viscoplastic constitutive model enhanced by a multi-mechanism nonlocal damage model
- Calibration
  - Several hardening/softening combinations
  - Requires composite material tests
    - Length scale effect

$$l_s = 3\mu\text{m} \left( 1 - \frac{D_s}{D_{s\infty}} \right)$$

- Non-local BC at fibre interface



V.-D. Nguyen, L. Wu, L. Noels, A micro-mechanical model of reinforced polymer failure with length scale effects and predictive capabilities. Validation on carbon fiber reinforced high-crosslinked RTM6 epoxy resin (2019)

# Micro-structural simulation of fiber-reinforced highly crosslinked epoxy

## Resin model: failure softening

– Failure surface

$$\left\{ \begin{array}{l} \phi_f = \gamma - a \exp\left(-b \frac{\text{tr}(\hat{\tau})}{3\hat{\tau}^{eq}}\right) - c \\ \phi_f - r \leq 0; \dot{r} \geq 0; \text{ and } \dot{r}(\phi_f - r) = 0 \\ \dot{\gamma}_f = \dot{r} \end{array} \right.$$

– Damage evolution

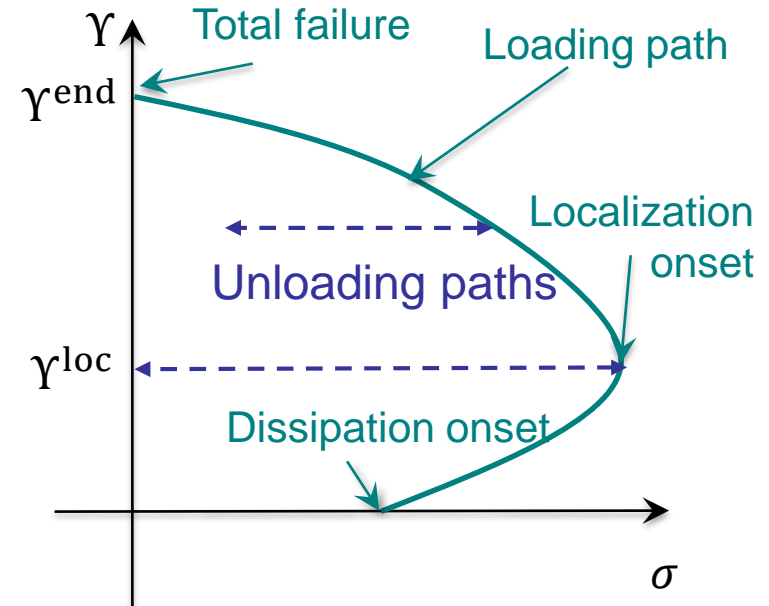
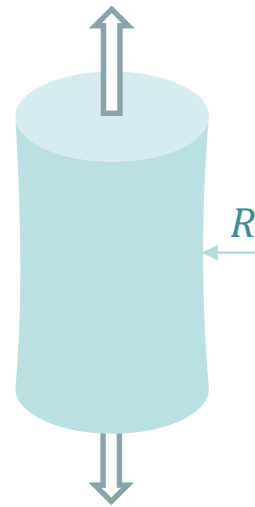
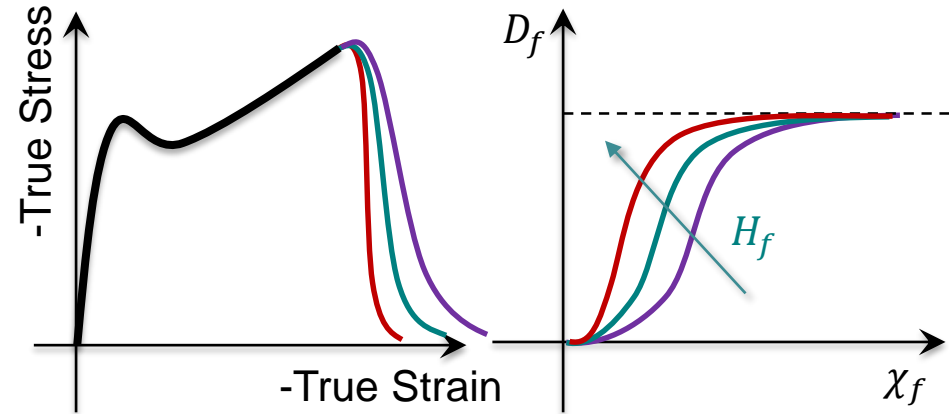
$$\left\{ \begin{array}{l} \dot{D}_f = H_f(\chi_f)^{\zeta_f} (1 - D_f)^{-\zeta_a} \dot{\chi}_f \\ \chi_f = \max_{\tau} (\tilde{\gamma}_f(\tau)) \\ \tilde{\gamma}_f - l_f^2 \Delta \tilde{\gamma}_f = \gamma_f \\ l_f = 3 \mu\text{m} \quad \nabla_0 \tilde{\gamma}_f \cdot \mathbf{N} = 0 \end{array} \right.$$

– Affect ductility

– Calibration

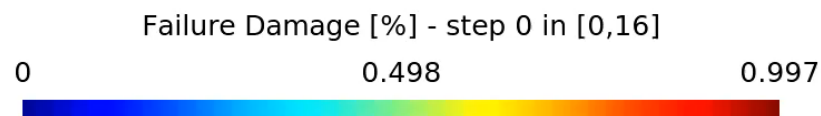
- Recover the epoxy  $G_c$
- From localization simulation

$$G_c = \frac{\mathcal{D}_{V_0}^{\text{end}} - \mathcal{D}_{V_0}^{\text{loc}}}{A_0}$$



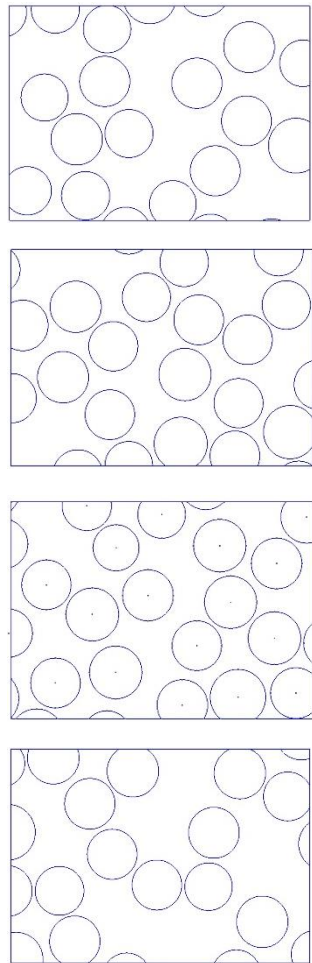
V.-D. Nguyen, L. Wu, L. Noels, A micro-mechanical model of reinforced polymer failure with length scale effects and predictive capabilities. Validation on carbon fiber reinforced high-crosslinked RTM6 epoxy resin (2019)

- Non-linear SVE simulations

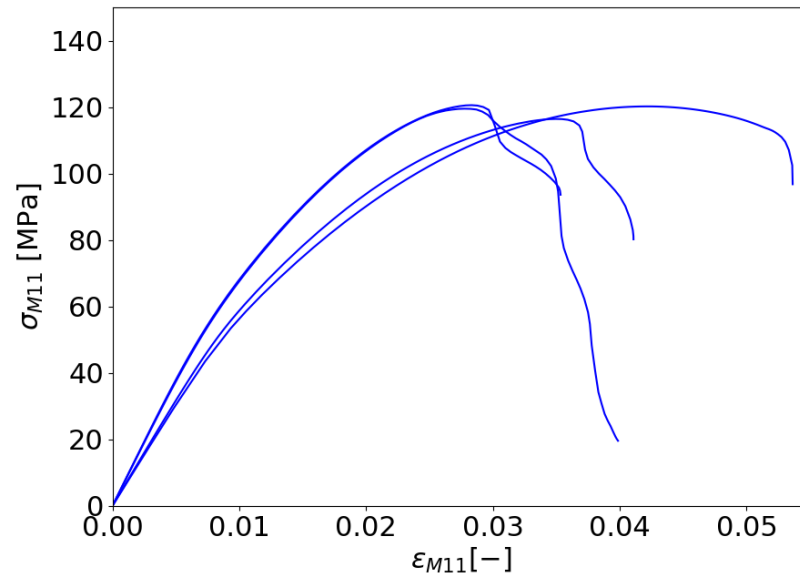


Y  
| Z X

- Non-linear SVE simulations



- Elastic Tensor  $C_M^i$
- Strain-stress evolution
- Energy Release Rate ( $G_c$ )
- Fibre volume Fraction
- Maximum Attained Stress
- Strain at Maximum Stress



$G_c$  [N/mm]

0.0408

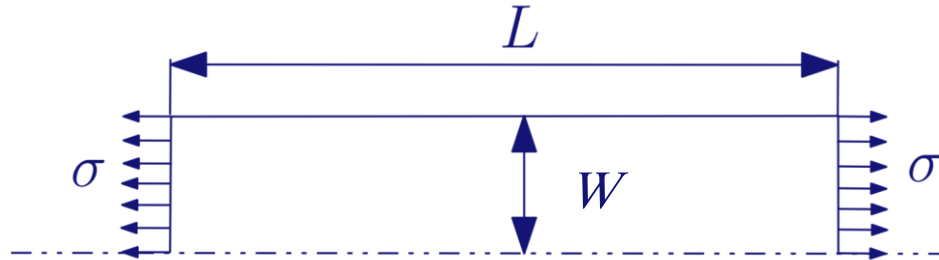
0.0780

0.0654

0.1019



- Non-linear SVE simulations
  - Energy Release Rate ( $G_c$ ) from MFH simulations
    - $L \gg l$ ;  $W \ll l$  to obtain non-dimensional  $G_c$

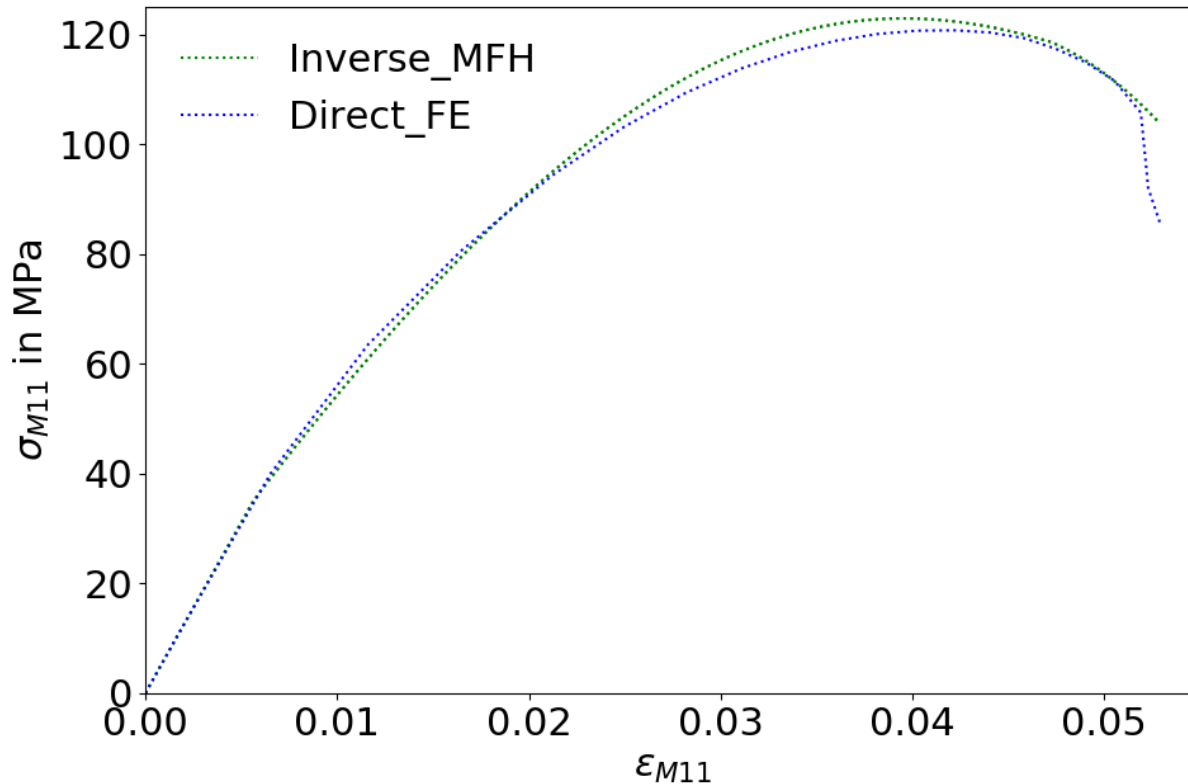


- Damage evolution modelled with a two-part function:
  - Linear increase of damage up to onset point
    - Rate of the damage evolution depends on the damage and the accumulated plastic strain at onset point  $\rightarrow$  All parameters obtained directly after inverse process
      - $\dot{\tilde{D}}_0 = \tilde{D}_{0_{onset}} / \tilde{p}_{0_{onset}} (\Delta \tilde{p}_0)$
  - After onset point,  $G_c$  becomes the only objective value. Objective function based on matching full-field  $G_c \rightarrow$  Optimization process for parameter identification ( $\alpha, \beta$ )
    - $\dot{\tilde{D}}_0 = \alpha (\tilde{p}_0 + \Delta \tilde{p}_0 - \tilde{p}_{0_{onset}})^\beta \Delta \tilde{p}_0$

# Non-linear stochastic Mean-Field Homogenization

- MFH model results:

- The non-local length affect final  $G_c \rightarrow$  Its value results from micro-scale simulations
  - Used non-local length of  $5.5 \mu\text{m}$



Method	$G_c$ [N/mm]
Direct FE	0.0945
Inverse MFH	0.0932



- Good representation of SVE behavior up to failure point

- On-Going work
  - Study of post-onset function parameter optimization process
  - Optimization of initial parameters to reduce iterations
  - Possibility of introducing a crack to obtain the desired  $G_c$
- Next step
  - Perform multiple SVE simulations
    - Stochastic data for macro-scale simulations
  - Perform efficient macro scale simulations of a ply failure taking into account geometrical variabilities of the microstructure

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Thank you for your attention!

Special thanks to:



Wallonie  
STOMMMAC project

**fnrs**  
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