



Numerical Analysis of Thermal Stress in Laser Cladding Technology of M4 High Speed Steel

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Material High Speed Steel M4

- Fe-Cr-C-X alloys with X: carbide-forming element (i.e. V, Nb, Mo or W)
- Hard carbides \Rightarrow High hardness and wear resistance
- Applications: high speed machining, cutting tools, cylinders for hot rolling mills, molds...



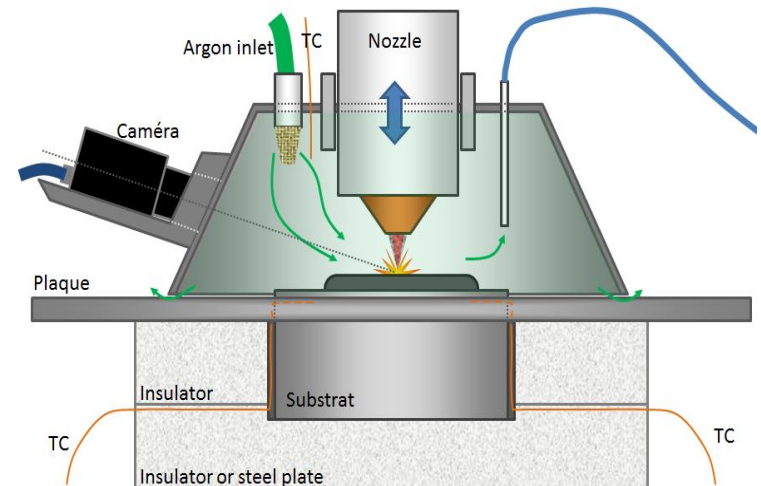
Towards a thermo- mechanical validated model

For High Speed Steel (M4 grade) wt%

C	Cr	Mo	V	W	Ni	Si	Fe
1.35	4.30	4.64	4.10	5.60	0.34	0.9	0.33

Particle size [50 to 150 μm]

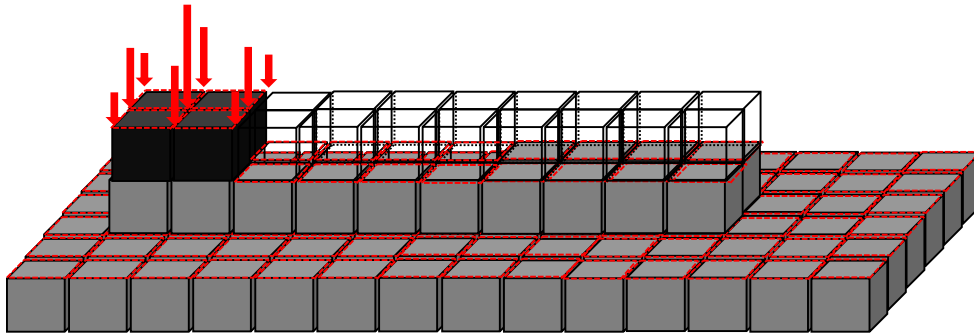
Direct Energy Deposition DED process



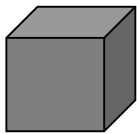
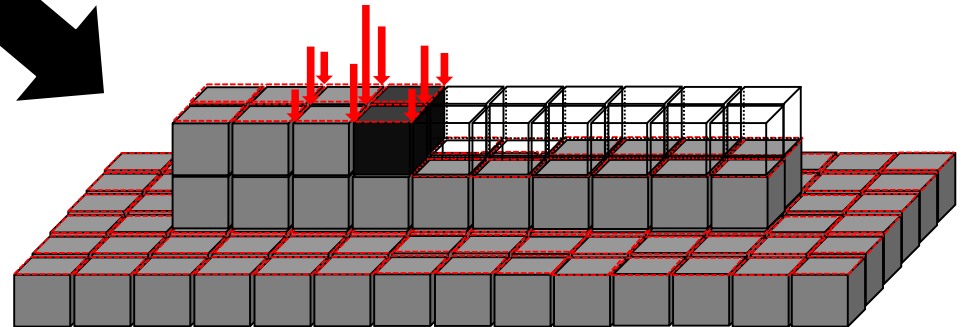
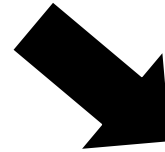
Content

- In house FE code « Lagamine »
- Bulk experiments
2D thermal simulations
- Thin wall experiments
3D thermo-mechanical simulations
- Conclusions - Perspectives

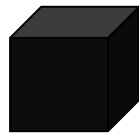
Element birth technique



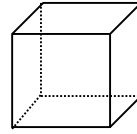
**For a thin wall 3D
Bulk Sample 2D**



Active element



Newly active element



Inactive element



Convection and radiation element

convection-radiation elem. on vertical planes of the clad not drawn

Thermal equations

Heat transfer per conduction

$$\frac{\partial}{\partial x} \left(k \frac{\partial T}{\partial x} \right) + \frac{\partial}{\partial y} \left(k \frac{\partial T}{\partial y} \right) + \frac{\partial}{\partial z} \left(k \frac{\partial T}{\partial z} \right) + Q_{\text{int}} = \rho c_p \frac{\partial T}{\partial t}$$

Conductivity
Volume energy
Density
Heat Capacity

Heat transfer per convection and radiation

$$-K \cdot (\nabla T \cdot n) = -h(T - T_0) - \varepsilon \sigma (T^4 - T_0^4)$$

Convection Coef.
Emissivity
Stefan-Boltzmann Constant

Melting latent Heat

$$C_p^* = \frac{L_f}{T_{em} - T_{sm}} + C_p$$

Enthalpic formulation

$$H = \int \rho \cdot c(T) dT$$

Enthalpy

Mechanical equations

- Hooke's law

$$\underline{\underline{\sigma}} = \frac{E(T, y)}{1 + \nu(T, y)} \left(\underline{\underline{\varepsilon}}^e + \frac{\nu(T, y)}{1 - 2\nu(T, y)} \text{Tr}(\underline{\underline{\varepsilon}}^e) \underline{\underline{\mathbf{I}}} \right)$$

- Plastic criterion: von Mises and associated plasticity

$$f = \frac{3}{2} \underline{\underline{\tilde{\sigma}}} : \underline{\underline{\tilde{\sigma}}} - R^2 \quad \underline{\underline{\dot{\varepsilon}}}^p = \dot{\lambda} \frac{\partial f}{\partial \underline{\underline{\sigma}}}$$

- Hardening law: isotropic (**multilinear curve**)

$$R = \sigma_y(T, y) + E^p(T, y) \varepsilon_{\text{eq}}^p \quad \text{avec} \quad \varepsilon_{\text{eq}}^p = \sqrt{\frac{2}{3} \underline{\underline{\varepsilon}}^p : \underline{\underline{\varepsilon}}^p}$$

Compression tests at 3 temperatures and 3 different strain rates
→ NO need viscous approach

Model identification phase

In put material data

conduction, heat capacity, latent heat
measured on samples extracted
from the clad & the substrate
(DSC, Laser flash, dilatometry)

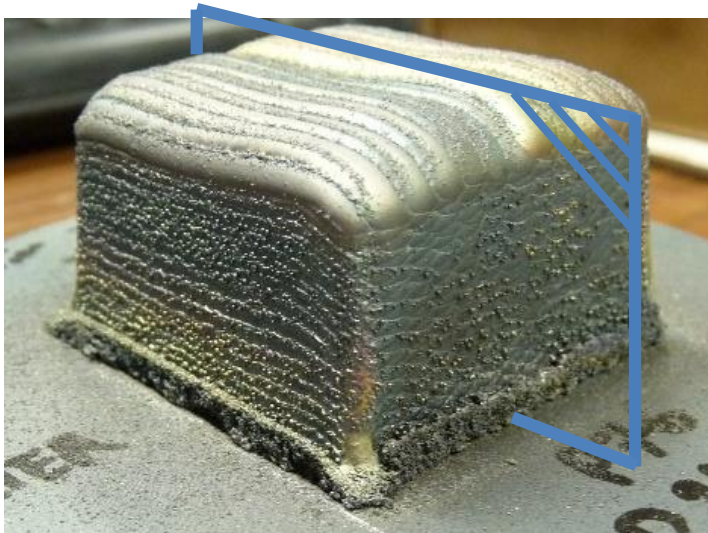
Convection, Radiation, laser absorption
fitted by inverse modelling

Target **BOTH**

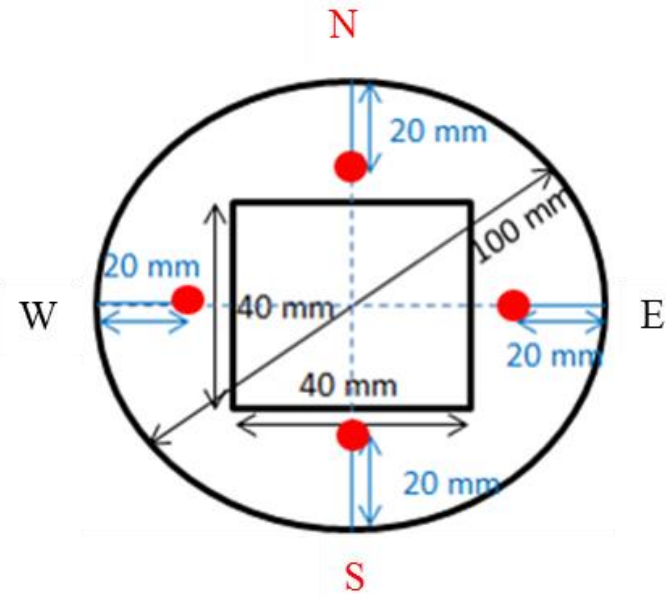
Temperature + Melt pool depth measured

“2D” bulk samples

	Bulk Sample
Laser beam speed (mm/s)	6.67
Laser power (W)	1100
Pre-heating (°C)	300
Mass flow (mg/s)	76
Number of tracks per layer	27
Total number of layers	36



40 x 40 x 27.5 mm (972 tracks)

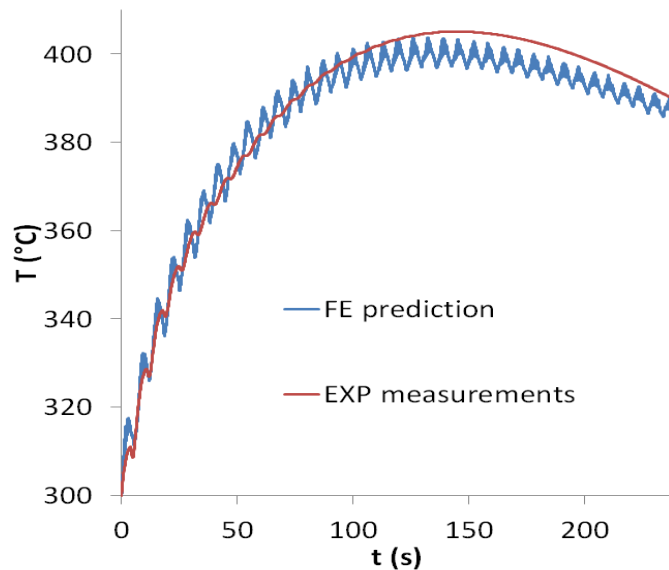


4 Thermocouples

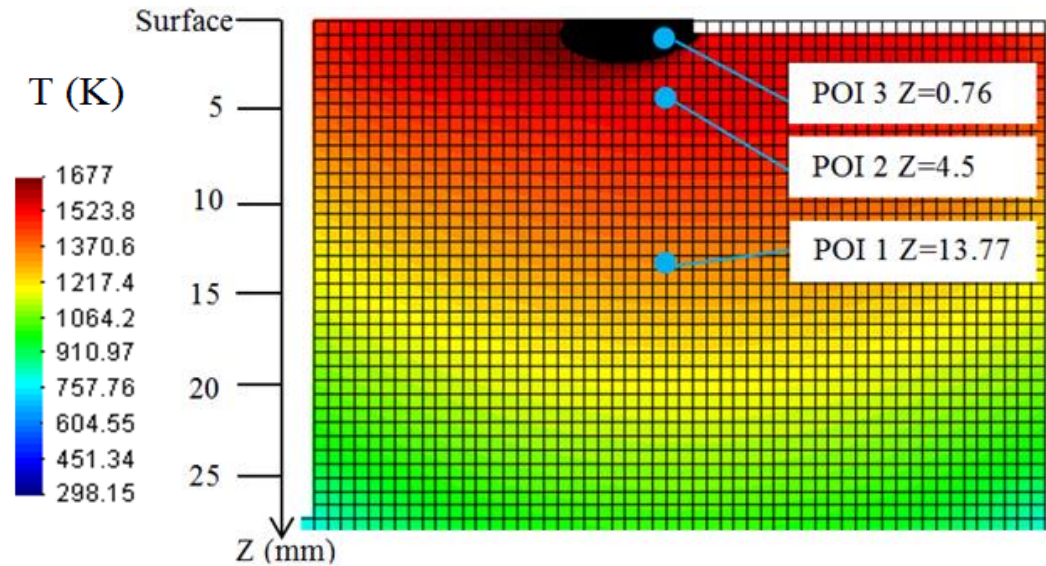
Thermal measurement in the substrate

“2D” bulk samples

T_p° in the substrate

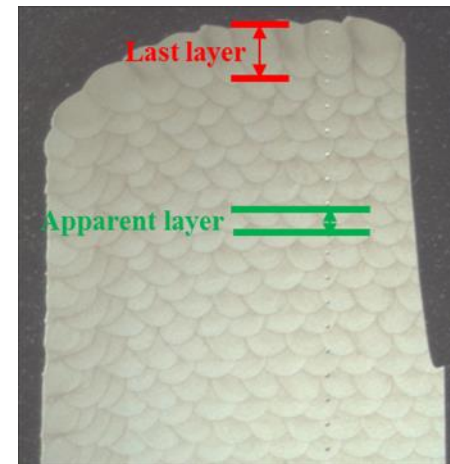


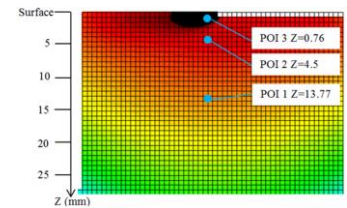
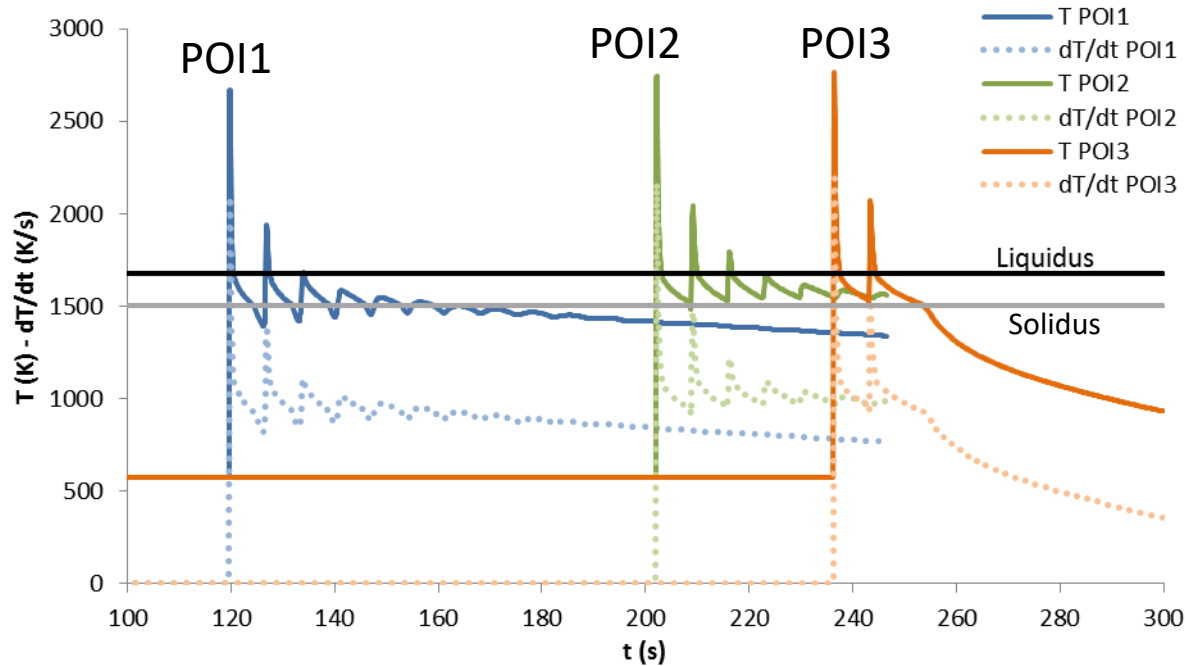
Predicted T_p° in the clad



Melt pool depth

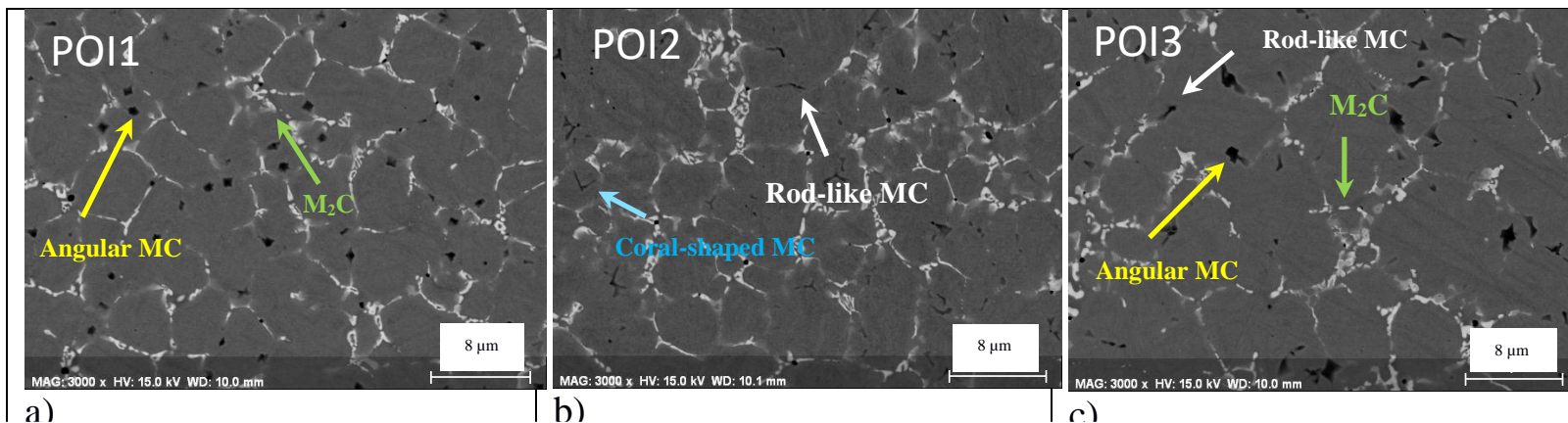
Key data for identifying single set of data by inverse simulations (convection, radiation, absorption coefficient)





- Number of full partial remelting
- T_p° Level between solidus and liquidus
- Superheating temperature

Jardin R.T., et al. (2019)
Materials Letters. 236:42-45

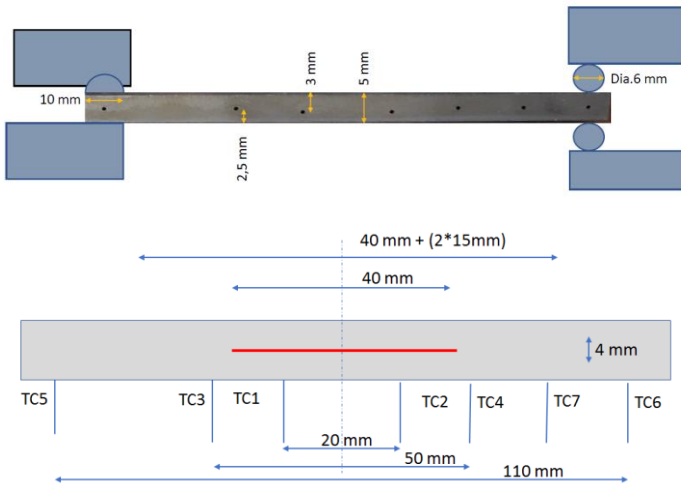


a) star-like MC and lamellar eutectic M_2C intercellular carbides

b) coral-shaped intracellular MC, intercellular eutectic M_2C and refined cells due to multiple melting

c) coarse angular MC and eutectic M_2C within intercellular zones

“3D” thin wall experiments (January)



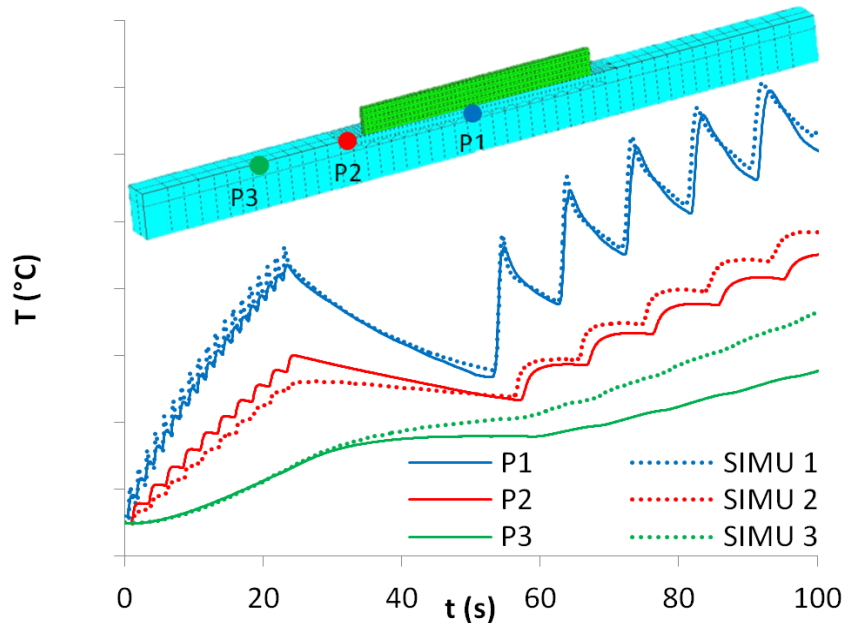
Preheating reached = 150°C

	Substrate pre-heating	Clad deposition
Length of centered laser pass for pre-heating (mm)	40	40
Laser beam speed (mm/s)	41.7	8.3
Laser power (W)	260	(Constant)500
Temperature at thermocouple P1 at preheating end and at cladding start in °C	217	134
Number of laser passes	20	10

With a thinner substrate there too much bending → risk for laser position

With thicker substrate crack situation worst

“3D” thermal analysis - thin walls



Simulations until 5th layer
Convection needs to be function of T
Constant value not OK

Previous measured thermophysical parameters for the clad

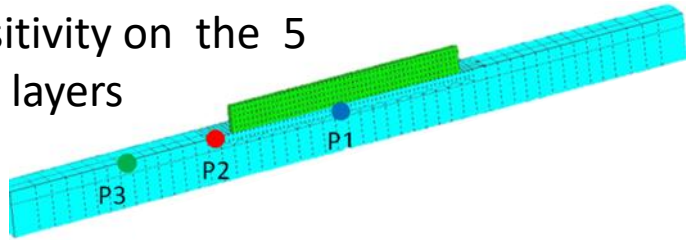
Substrate 42crMo4
different origin than for bulk sample

- Impossible to recover temperature measurements with previous values of conductivity and thermal capacity.
- New measurements indeed showed different results for conductivity and heat capacity

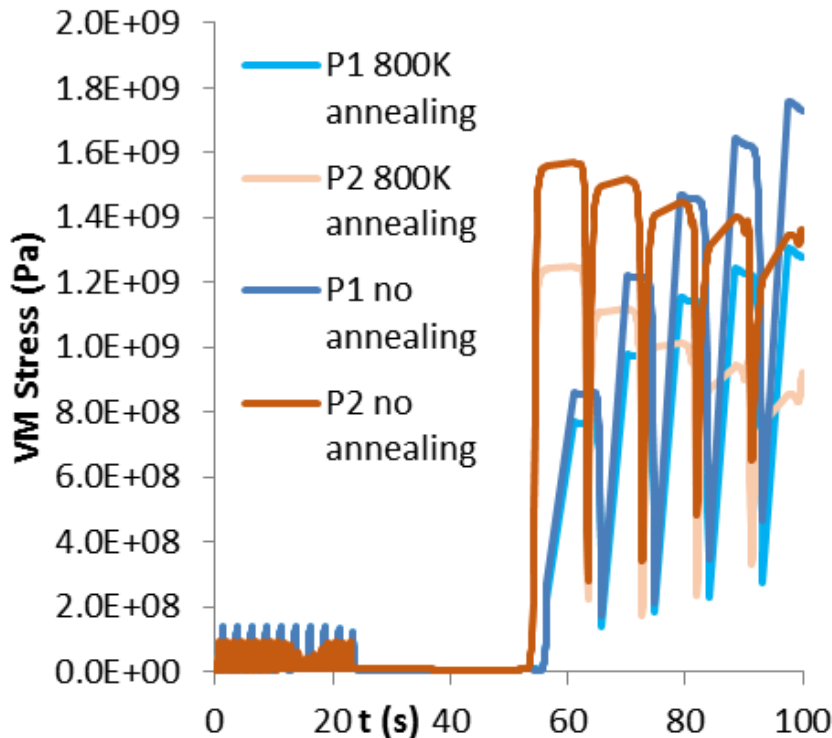
(Previous block for bulk sample in martensite state, current bars in Pearlitic state)

“3D” thermo-mechanical data analysis - thin walls

Sensitivity on the 5 first layers

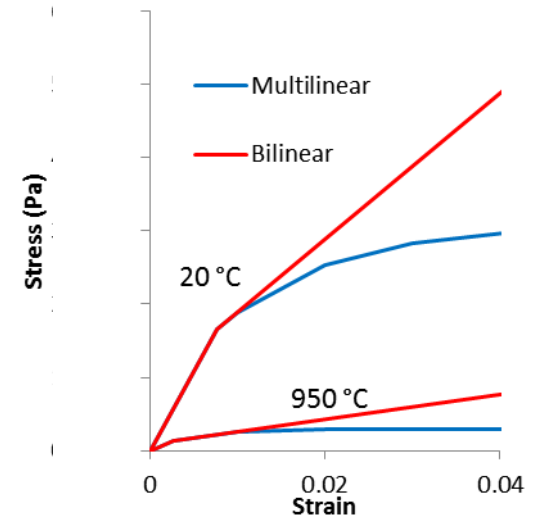


Numerical annealing temperature: plastic strain is forgotten if t_p° decreases below this annealing t_p°

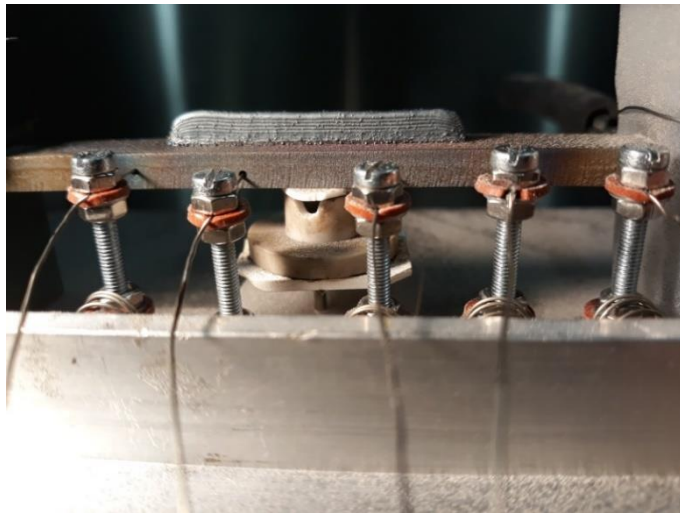


Results for bilinear stress-strain curves

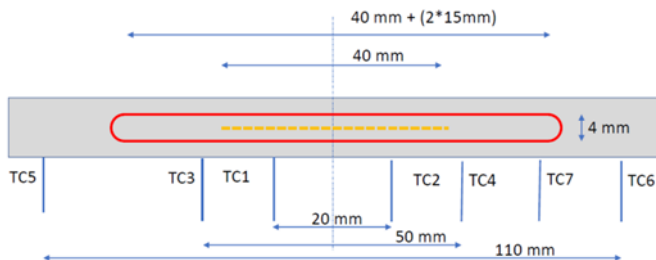
Far Less sensitive for multi linear curves



“3D” thin wall experiments (March)



No more crack
Nearly constant height



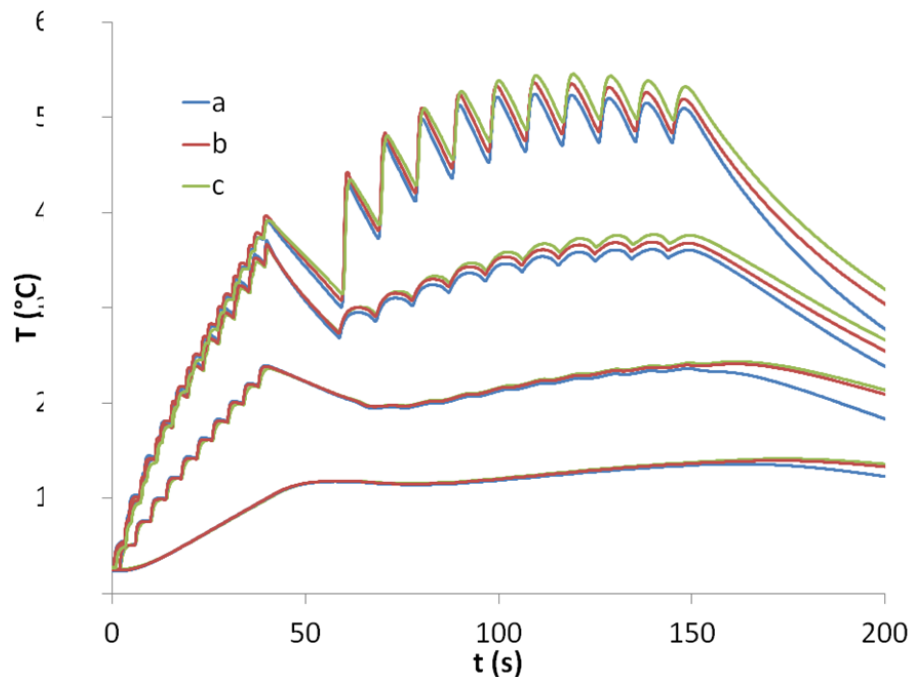
	Substrate pre-heating	Clad deposition
Length of centered laser pass for pre-heating (mm)	70	40
Laser beam speed (mm/s)	41.7	8.3
Laser power (W)	260	600+500=>400
Temperature at thermocouple P1 at preheating end and at cladding start in °C	400	310
Number of laser passes	20	10

Pre heating at 300°C

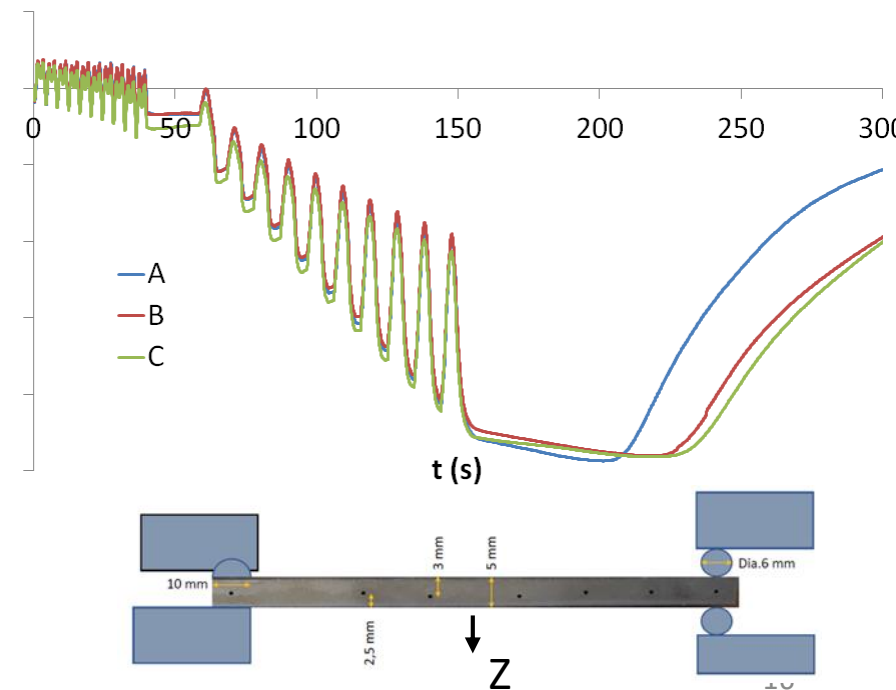
“3D” thin wall experiments

3 Experiments with similar conditions 10 layers without crack

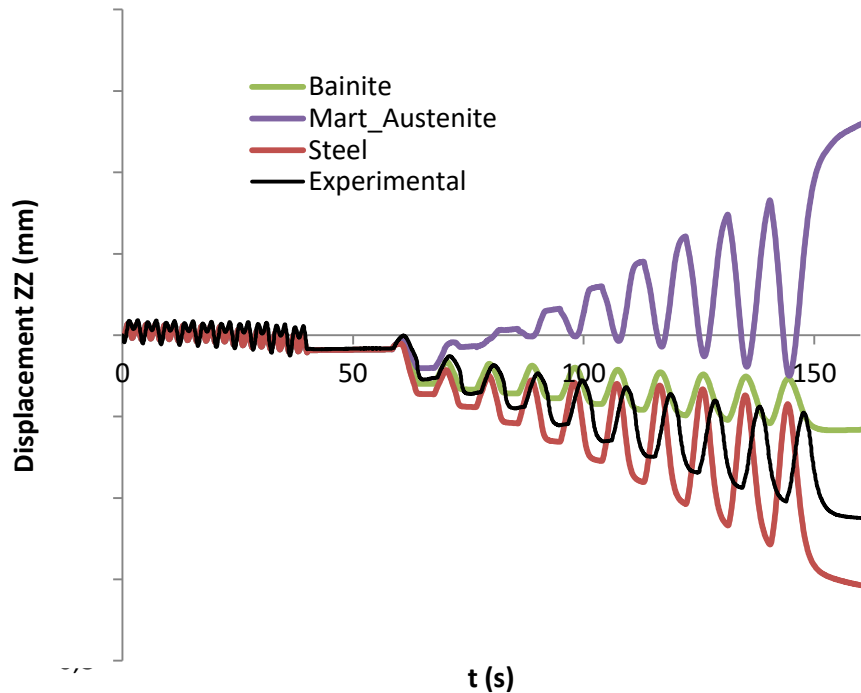
Temperature history



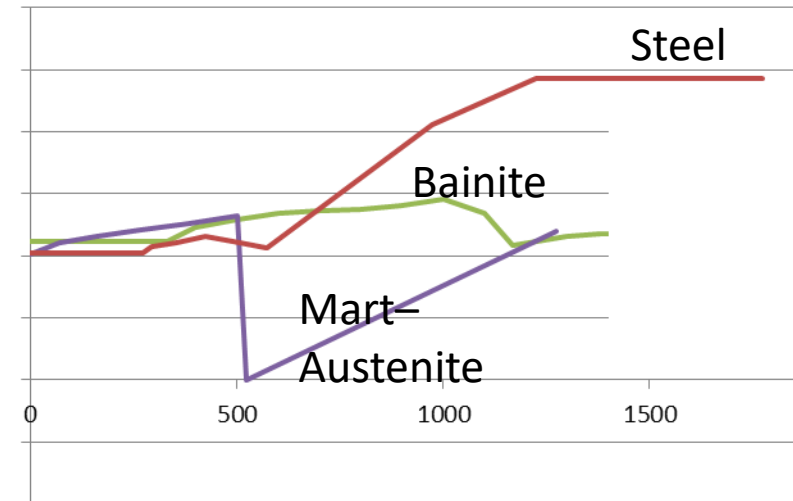
Vertical displacement at the middle



“3D” thermo-mechanical data analysis - thin walls - validation?



Dilatation coefficient of the clad



No effect of annealing temperature
Strong sensitivity to the dilatation coef of the clad

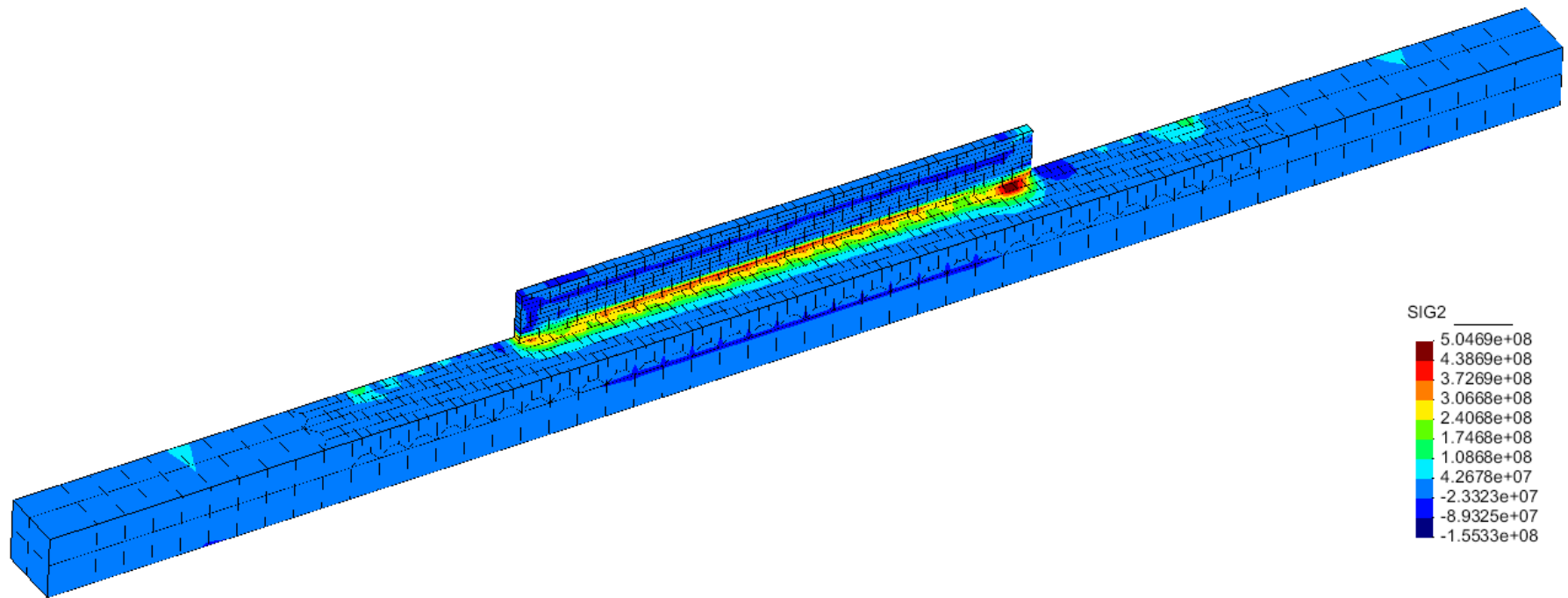
- Sensitivity to dilatation of substrate, HAZ, have to be done
- Metallography on the thin wall is on going

“3D” thermo-mechanical data analysis - thin walls - validation?

Transversal stress σ_{yy} along thin wall

At the end of cooling

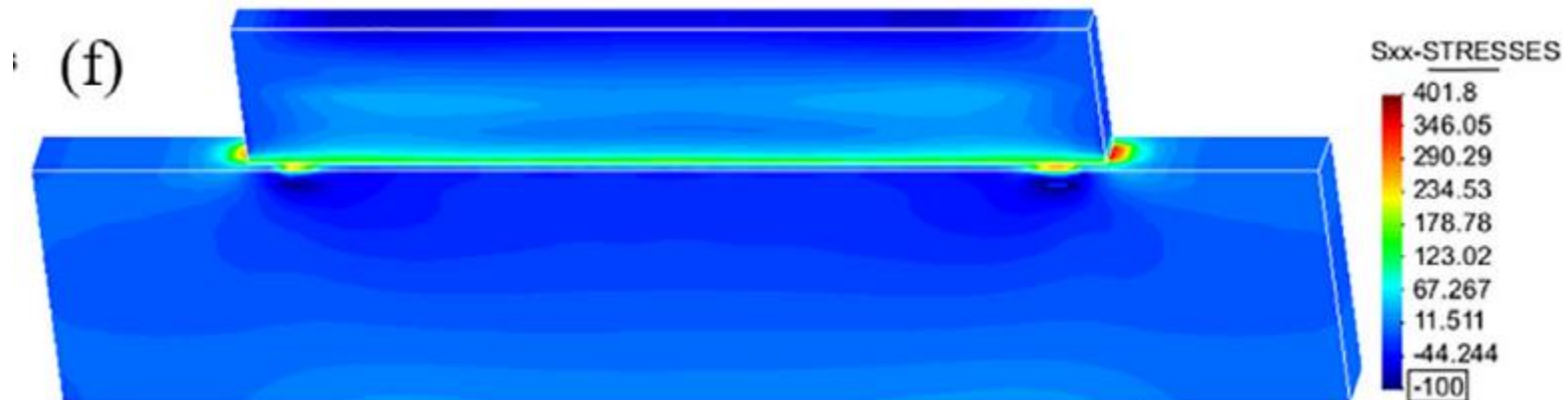
Dilatation Case « steel »



“3D” thermo-mechanical data analysis - thin walls - validation?

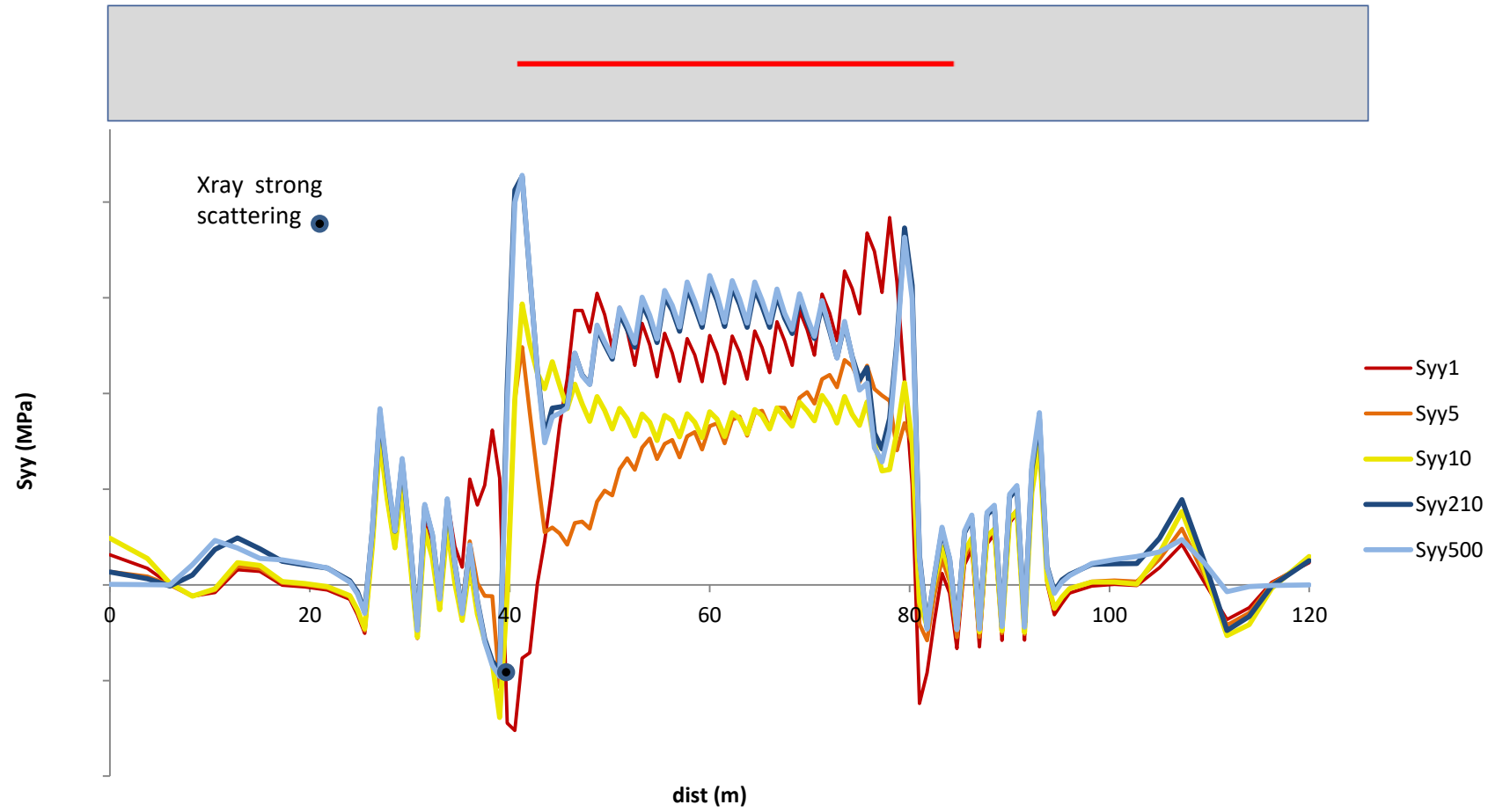
Transversal stress σ_{yy} along thin wall from literature

X. Lu, et al. , *In situ measurements and thermo-mechanical simulation of Ti – 6Al – 4V laser solid forming processes*, Int. J. Mech. Sci. 153–154 (2019) 119–130..



“3D” thermo-mechanical data analysis - thin walls - validation?

Dilatation Case « steel »



Conclusions

FE thermo-mechanical model available,
Solid latent heat and dilatation of a single phase
No activation of phenomenological solid phase transformation model

Annealing t_p° effect depends on the shape of hardening curves
No effect on residual stress or displacement for the correct stress-strain curves

Validation by temperature, melt pool size, displacement, residual stress, microstructure of thin wall still ongoing ...

X Ray measurements provide quite scattered data
Complex microstructure justifies scattering + Laser cladding
experiment repeatability

Perspectives

Systematic sensitivity approach (clad and substrate properties)
→ Identify what should be improved to reach validation

Use of different experimental conditions
crack and no cracks cases + hot tensile rupture value
for further FE validation method

Thank you

Jardin R.T., et al. (2019)
Materials Letters. 236:42-45