

# **Synthetic oligorotaxanes exert high forces when folding under mechanical load**

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1 **Combining the exquisite complexity of folded molecular architectures<sup>1,2</sup> and**  
2 **mechanically interlocked molecules<sup>3-5</sup>, donor-acceptor oligorotaxane foldamers were**  
3 **recently synthesized<sup>6</sup>. So far, their mechanochemical properties, and thus their potential**  
4 **as pieces of advanced molecular machinery, are unknown. Here, using AFM-based**  
5 **single-molecule force spectroscopy, we mechanically unfolded oligorotaxanes, made of**  
6 **oligomeric dumbbells incorporating 1,5-dioxynaphthalene units encircled by**  
7 **cyclobis(paraquat-*p*-phenylene) rings. Real-time capture of fluctuations between**  
8 **unfolded and folded states reveals that the molecules exert forces of up to 50 pN against**  
9 **a mechanical load of up to 150 pN, and displays transition times of less than 10  $\mu$ s.**  
10 **While the folding is as fast as that observed in proteins, it is remarkably more robust,**  
11 **thanks to the mechanically interlocked structure. Our results show that oligorotaxanes**  
12 **have the potential to exceed the performance of natural foldamers-based machines.**  
13

14 Folding is a ubiquitous process that nature uses to control the conformations of its molecular  
15 machines so as to perform chemical and mechanical tasks. Over the years, chemists have  
16 synthesized foldamers that adopt well-defined and stable folded architectures, mimicking the  
17 control expressed by natural systems in their three-dimensional arrangements<sup>1,2</sup>. Mechanically  
18 interlocked molecules, such as rotaxanes and catenanes, are prototypical molecular machines  
19 that enable the controlled movement and positioning of their component parts<sup>3-5</sup>. The  
20 mechanical bonds maintain precise co-conformations that confer upon molecules specific  
21 properties and functionality, such as movement, transport, and catalysis, to name but a few.  
22 The concept of simple [2]rotaxanes—a dumbbell encircled by a ring— has evolved in the  
23 past 10 years into more complex structures<sup>3</sup>. Recently, combining the exquisite complexity of  
24 these two classes of molecules, the syntheses of well-defined and function-designed  
25 compounds have emerged<sup>6-10</sup>. Among these, donor-acceptor oligorotaxane foldamers, in  
26 which interactions between the mechanically interlocked component parts dictate the single-  
27 molecule assembly of a folded secondary structure, have been proposed<sup>6,8,9</sup>. The unique  
28 folding motif is based on flexible poly(ethylene oxide) (PEO) dumbbells bearing 1,5-  
29 dioxynaphthalene (DNP) donors, which fold their way through a series of cyclobis(paraquat-  
30 *p*-phenylene) (CBPQT<sup>4+</sup>) acceptor rings, also known as the little Blue Boxes<sup>11</sup>, in a  
31 serpentine-like fashion in order to enable extended donor–acceptor (D–A) stacking  
32 interactions between the DNP units (red in Fig. 1) and the electron-poor 4,4'-bipyridinium  
33 (BIPY<sup>2+</sup>) units in the CBPQT<sup>4+</sup> rings (blue in Fig. 1). The large number of  $\pi$ -stacks occurring  
34 between the donor and acceptor groups confers a compact folded secondary structure upon  
35 the molecule, as shown by X-ray crystallographic studies, <sup>1</sup>H NMR spectroscopy in solution,  
36 and computational modelling studies<sup>6,8,9,12,13</sup>. PF<sub>6</sub><sup>-</sup> counterions also intercalate<sup>12</sup> in these  
37 folded (co)-conformations to neutralize the four positive charges on each ring and counteract  
38 the Coulomb repulsion, further stabilizing the structure. Molecular dynamics simulations have

39 shown that the folding is energetically favoured but entropically penalized, with the energetic  
40 contributions overcoming the entropy penalty and effectively driving the folding<sup>13</sup>.

41 Much of the exquisite and detailed information on the structure, dynamics and operation  
42 of natural complex functional molecules and machines have been obtained from single-  
43 molecule force spectroscopy<sup>14-19</sup>. During these experiments, the molecule of interest is  
44 trapped and pulled between a substrate and a microscopic probe, and the restoring force is  
45 measured. AFM-based force spectroscopy is able to probe the mechanical properties of  
46 molecules with sub-nanometer resolution<sup>14,16,17,20</sup>. It has been widely used to elucidate the  
47 conformations of (bio)macromolecules<sup>14</sup>, to study the mechanochemistry of biological  
48 machines<sup>17</sup>, to understand the binding process between molecular partners<sup>21,22</sup>, to investigate  
49 the mechanochemical processes in rotaxanes, catenanes and other mechanically interlocked  
50 compounds<sup>23-25</sup>, etc.

51 Here, we describe how we have used AFM-based force spectroscopy to investigate the  
52 folding propensity of oligorotaxanes under mechanical load at the single-molecule level and  
53 provide information on their mechanochemical properties. The oligorotaxanes are terminated  
54 by bulky stoppers which prevent the rings from dethreading and 1,2-dithiolane rings at both  
55 their ends in order to allow strong interactions with the gold-coated substrate and tip. They  
56 were grafted in solution onto a gold-coated silicon surface, following our previously  
57 published strategy<sup>24</sup> to obtain a sparse distribution of the molecules on the surface (see  
58 Supplementary information) and optimize the interactions of single molecules with the tip.  
59 During the force spectroscopy measurements, a gold-coated silicon AFM tip approaches the  
60 oligorotaxane-functionalized substrate and strong Au-S interactions with the free end of the  
61 oligorotaxane are created. Then, increasing the distance between the tip and the surface, the  
62 behaviour of the molecule under extension is monitored by the deflection of the flexible tip-  
63 bearing cantilever (Fig. 2a).

64 Force-extensions profiles were obtained in *N,N*-dimethylformamide (DMF). A  
65 characteristic and reproducible sawtooth profile (Fig. 2b) with equally separated peaks was  
66 observed. Interestingly, the pattern is made of an alternation of higher and smaller peaks. We  
67 attribute this pattern to the sequential rupture of the intramolecular interactions between each  
68 CBPQT<sup>4+</sup> ring and the unoccupied DNP units on both of its sides. The alternation of higher  
69 and smaller peaks can be explained by the unfolding cooperativity: when an interaction is  
70 broken, the symmetrical one at the other side of the CBPQT<sup>4+</sup> ring is weakened and can be  
71 broken more easily. The mean rupture forces, between 100 and 150 pN (Fig. S3), are  
72 consistent with the breaking of weak noncovalent interactions like hydrogen bonds,  
73 electrostatic interactions,  $\pi$ -stacking and C–H– $\pi$  interactions<sup>26</sup>. The last peak on the curves  
74 displays a higher force, of up to 300 pN, an observation that is characteristic of an S–S  
75 interaction with gold<sup>27</sup>.

76 To support our hypothesis, we measured the length increments between successive  
77 rupture peaks (Fig. 2b) with the use of the worm-like chain (WLC) model<sup>28</sup>, an entropic  
78 elasticity model that predicts the relationship between the extension of a flexible polymer and  
79 its entropic restoring force. Interestingly, the repetition of rupture peaks equally separated by  
80 a contour length increment ( $\Delta L_c$ ) of about 1.2 nm was observed, as depicted by the main  
81 population in the  $\Delta L_c$  distribution (Fig. 2c). Considering the X-ray crystallographic data  
82 obtained for crystals grown from a solution of oligorotaxanes<sup>6,9,12</sup>, it seems reasonable to  
83 attribute the  $\Delta L_c$  of 1.2 nm to the breaking of the interaction linking one side of a CBPQT<sup>4+</sup>  
84 ring to the neighbouring DNP unit. Indeed, the theoretical distance between two subsequent  
85 DNP units in a locally folded conformation is about 0.7 nm, whereas this distance rises to  
86 almost 1.9 nm for the extended PEO segment in the locally unfolded conformation (see  
87 Supplementary information). Therefore, the expected difference in length between locally  
88 folded and unfolded states is 1.2 nm (Fig. 2a), which nicely matches the value measured here.

89 In less frequent cases, the disappearance of the small peaks (Fig. 2d) gives rise to higher  
90 contour length increments ( $\Delta L_c$  centered at 2.3 nm, the second population in Fig. 2c), which  
91 can be attributed to the unfolding of two symmetrical repeating units almost simultaneously.  
92 This observation of missing peaks in force curves, and thus a  $\Delta L_c$  twice (or more) as high, has  
93 already been observed by others in proteins containing repeating units separated by a few  
94 nanometers<sup>29</sup>. The role of unfolding cooperativity in such systems with repeating segments  
95 close to each other has been proposed to explain the populations of  $\Delta L_c$  at double values.  
96 When an interaction is broken, the next one is weakened and can be broken easily and rapidly,  
97 causing a missing peak in the curves, and thus a doubling of  $\Delta L_c$ . Here, the existing symmetry  
98 on both sides of the CBPQT<sup>4+</sup> ring, and the very short distance ( $\sim 1$  nm) involved, is consistent  
99 with a local cooperativity that makes the probability of nearly simultaneous breaking of the  
100 symmetrical interactions rather likely. On the basis of the highly defined  $\Delta L_c$  populations  
101 observed in characteristic reproducible sawtooth patterns, we can also exclude the attribution  
102 of the multi-peak pattern to the detachment of multiple interactions (randomly distributed) of  
103 the molecule with the tip. The unfolding pattern is thus in good agreement with the previously  
104 proposed folded (co)-conformation in solution<sup>6,8,9</sup>.

105 Pulling-relaxing experiments were then performed (Fig. 3). A single molecule was  
106 trapped between the tip and the substrate, and retraction-approach cycles were performed  
107 repeatedly. The force *versus* z-displacement curves show sequential breaking and reforming  
108 of a single interaction within the foldamer against the mechanical load. These four successive  
109 cycles performed on a single interaction demonstrate the reversibility of the unfolding and the  
110 robustness of the system. Moreover, rapid fluctuations between folded and unfolded states  
111 were detected, both in the pulling and relaxing curves (Figs. 3 and 4a). By inspecting the  
112 standard pulling curves at higher loading rates (Fig. S4), rapid transitions between two peaks  
113 were also observed. These quick fluctuations between two subsequent unfolding peaks

114 provide evidence for the thermodynamic equilibrium between locally folded and unfolded  
115 states when the molecule is under an external load. Very few examples of observing  
116 fluctuations in real time have been described in the literature. Fluctuations were observed for  
117 the first time in RNA molecules under an external load applied with optical tweezers<sup>30</sup>. The  
118 RNA molecules were shown to fluctuate back and forth between two conformations when the  
119 loading rate was small enough to approach thermal equilibrium conditions or when kept under  
120 a constant low force. These fluctuations can be explained by the reforming of an interaction a  
121 short time after being broken. This phenomenon, called *hopping*, has also been observed<sup>31</sup> in  
122 RNase H. Later, real-time binding-unbinding transitions of calmodulin<sup>32,33</sup> and reversible  
123 unfolding-refolding of an  $\alpha/\beta$  protein<sup>34</sup> were observed by AFM-based force spectroscopy.  
124 Here, we observe, for the first time we believe, such fluctuations in a significantly smaller  
125 synthetic molecule and at much higher velocities. The distributions of the external load ( $F_{app}$ )  
126 applied to the locally unfolded oligorotaxane and the force exerted ( $\Delta F$ ) by the molecule to  
127 regain its locally folded state are shown in Fig. 4b and 4c, respectively. These distributions  
128 show that the molecule is able to exert a force ( $\Delta F$ ) of 25 pN on average, up to 50 pN, against  
129 an external load ( $F_{app}$ ) of 60 pN, to refold and remake the broken interactions during the  
130 pulling experiment. In an attempt to obtain more details on the kinetics of these fluctuations,  
131 we focused (Fig. 4d) on the force *versus* time curves during the hopping events. At the  
132 highest loading rate ( $10^5$  pN·s<sup>-1</sup>), we measured a fluctuation rate of more than 4300 per  
133 second (Fig. S5). If we compare to the fluctuation rate measured for natural folding proteins  
134 like calmodulin<sup>33</sup>, it is more than 300 times higher. One fluctuation between partially  
135 unfolded and folded states happens in less than 10  $\mu$ s, which makes the oligorotaxane the  
136 fastest folding system ever measured at such high loading rates.

137 As these transitions can, in principle, be observed near the equilibrium, i.e., when the  
138 speed of the experiment is lower than the binding-unbinding kinetics, we can compare

139 qualitatively the results for the synthetic oligorotaxanes with those obtained previously for  
140 **folding proteins** under the same conditions. Since we have observed the hopping phenomenon  
141 on the full range of loading rates — $10^3$  to  $10^5$  pN·s<sup>-1</sup>, which is 100 to 1000 times higher than  
142 those previously tested for **natural folding proteins**<sup>30-34</sup>—, we can conclude with good reason  
143 that the folding motifs made of mechanically interlocked components are much more robust  
144 than the classical ones in the proteins investigated previously by AFM or by employing  
145 optical tweezers. The robustness also manifests itself in the intensity of the mechanical load  
146 against which the molecule can refold. **The maximum mechanical load that folding proteins**  
147 **can accommodate when fluctuating between unfolded and folded states during the pulling**  
148 **experiment is typically in the order of 10 pN**<sup>30-34</sup>. Although several proteins are able to refold  
149 **during relaxing experiments, when the tension exerted by the probe is released, the refolding**  
150 **during the pulling regime is a much rarer event and, in the few examples described in the**  
151 **literature, the mechanical load was limited to 10 pN**. Here, the oligorotaxane can refold  
152 during the force increasing regime against a load of 60 pN on the average, up to 150 pN. This  
153 remarkably robust behaviour can be explained by the mechanically interlocked components  
154 which restrict the degrees of freedom and stay in close proximity after the interaction breaks.  
155 The mechanically interlocked structure constrains the rings around the DNP recognition sites  
156 and favours a defined geometry that pre-orient the partners, decreasing the degrees of freedom  
157 and thus facilitating the refolding. The proximity ensures that the entropic penalty is low. It is  
158 thus possible, even under a high external load, to counteract the small increase of entropy and  
159 reform the interactions rapidly. Larger natural biological folds rely upon weak interactions  
160 amongst a significant number of constituents, all covalently bound in a single molecule.  
161 These folding interactions are opposed by the folding entropy which roughly scales as the  
162 number of constituents involved in the fold. When a load is applied, a cascade of disrupted  
163 interactions occurs, driven by a high gain in entropy. Compared to biological counterparts, the

164 refolding of oligorotaxanes is associated with a low entropy decrease that can be opposed by  
165 the enthalpic gain on account of the strong electron donor-acceptor interactions. The entropy  
166 drive to distort the fold is not so great, as the fold consists of only four ethylene oxide units  
167 and four C–C/C–O bonds. The fold in the oligorotaxanes can thus sustain a large load. The  
168 entropy plays a greater role in the unfolding of a protein, lowering the force against which the  
169 fold can be sustained.

170 Our results show that donor-acceptor oligorotaxanes, wherein the dumbbells are  
171 composed of oligoether chains interrupted by 1,5-dioxynaphthalene units, half of which are  
172 encircled by cyclobis(paraquat-*p*-phenylene) rings<sup>6</sup>, are able to exert significant forces in  
173 refolding against a high external load. The observation of fluctuations, even at high loading  
174 rates, implies that the thermal dynamics of these molecules are very high, making them the  
175 fastest ever measured at such high loading rates. The mechanically interlocked structure of the  
176 oligorotaxanes is most likely responsible for this remarkably robust and highly dynamic  
177 folding. Taking inspiration from nature's molecular machinery that makes use of well-  
178 defined, stable, but highly dynamic systems, the emerging class of synthetic oligorotaxane  
179 foldamers reported here reveal themselves to be efficient, robust and intricately regulated by  
180 their dynamics. Our results show that oligorotaxanes have the potential to exceed the  
181 performance of natural foldamers.

182

## 183 **Methods**

184 **Oligorotaxane synthesis.** The [5]rotaxanes were synthesized according to a protocol  
185 described recently<sup>6</sup>. Briefly, this one-pot synthesis uses a copper-catalyzed azide-alkyne  
186 cycloaddition to thioctic ester-functionalized stoppers at both ends of the DNP-derived  
187 polyether chains in the presence of CBPQT<sup>4+</sup> rings. Using conventional chromatographic

188 techniques, oligorotaxanes with half the DNP units encircled by a ring were isolated. The  
189 molecules were characterized by NMR spectroscopy and mass spectrometry<sup>6</sup>.

190 **Immobilization of the oligorotaxanes molecules on surfaces.** The oligorotaxanes were  
191 grafted onto gold-coated silicon substrates using our previously established method<sup>24</sup> to  
192 obtain a sparse regime of the molecule of interest. The substrates were dipped for 1 h in a  
193 solution of the oligorotaxane ( $2 \cdot 10^{-8}$  mol) containing 90 mol% of dodecyl disulfide in Me<sub>2</sub>CO  
194 at room temperature. The role of the dodecyl disulfide is to ensure a dilute distribution of the  
195 oligorotaxanes on the surface and passivate the substrate. All the details are given in SI.

196 **Single-molecule force spectroscopy experiments.** Experiments were carried out with a  
197 PicoPlus 5500 microscope (Agilent Technologies) equipped with a closed-loop scanner.  
198 Gold-coated tips (OBL-10 Biolever, Bruker; nominal spring constant  $k=0.009-0.1 \text{ N}\cdot\text{m}^{-1}$ )  
199 were used for all the force experiments. The spring constant of each cantilever was calibrated  
200 by the thermal noise and Sader methods.<sup>35,36</sup> The molecules were picked up for force  
201 experiments by gently pressing the AFM tip against the substrate. Force measurements were  
202 performed at loading rates between  $10^3$  and  $10^5 \text{ pN}\cdot\text{s}^{-1}$ . Pulling-relaxing cycles were realized  
203 using a custom-made routine to guide the tip. Each force curve contains 20,000 data points.  
204 All the details are given in SI.

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300

#### 301 **Author contributions**

302 D. S. and S. H. performed the AFM experiments and analysed the data. C. B. and Z. Z. carried out the  
303 oligorotaxane synthesis and characterization studies. A.-S.D. and J.F.S. designed the experiments. A.-S. D. and D.  
304 S. prepared the manuscript. All the authors discussed the data and commented on the manuscript.

305

#### 306 **Additional information**

307 Supplementary information is available on the online version of the paper. Reprints and permission information  
308 is available online at <http://www.nature.com/reprints>. Correspondence and requests for materials should be  
309 addressed to A.-S.D.

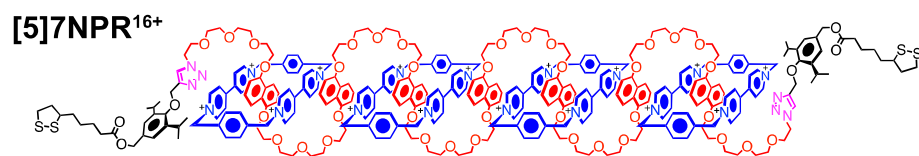
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#### 311 **Competing financial interests**

312 The authors declare no competing financial interests.

313

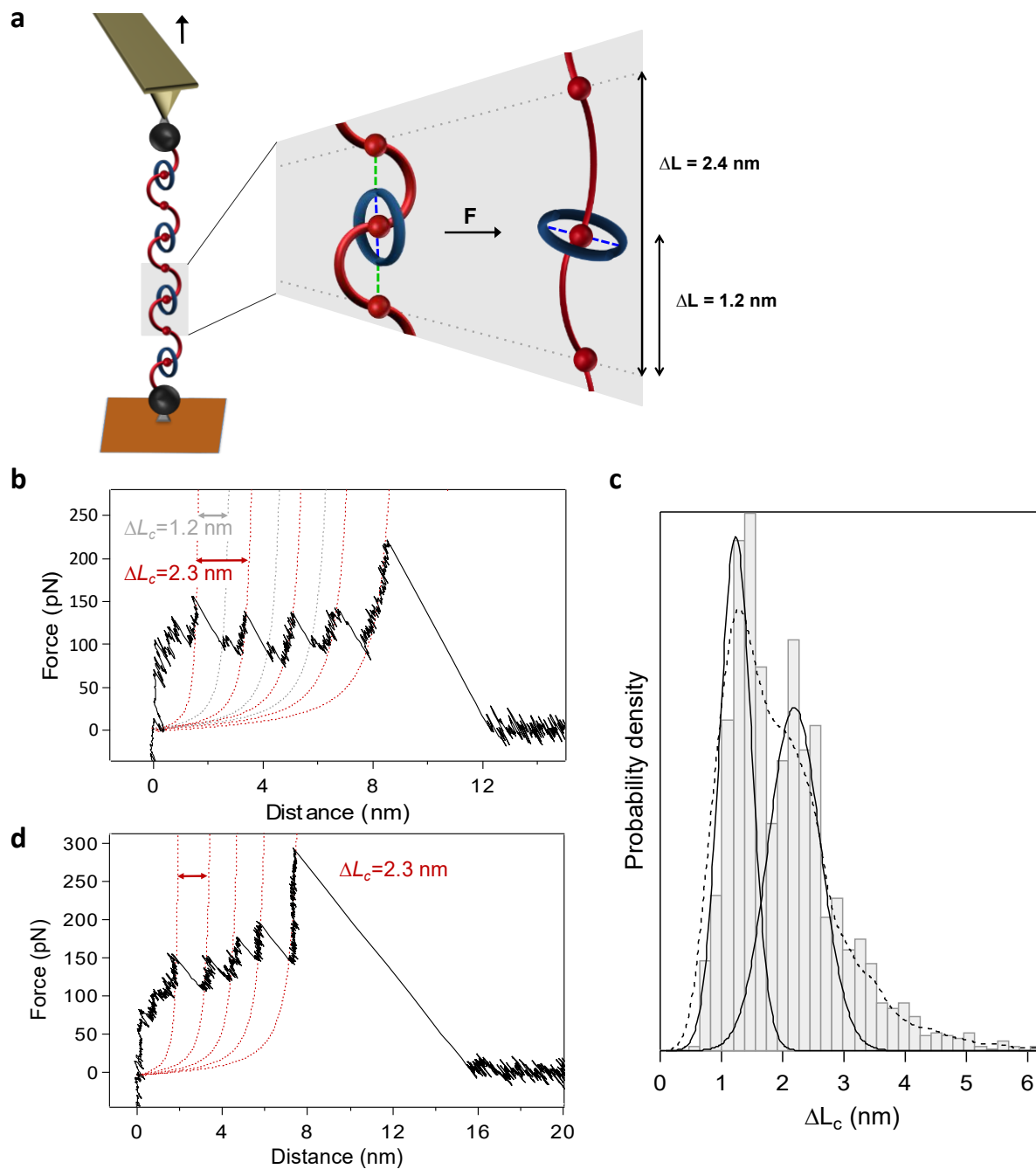
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315

316 **Figure 1. Structure formula and co-conformation of the [5]oligorotaxane ([5]7NPR<sup>16+</sup>) of**  
317 **the [0.5(*n*-1)+2] family.** The terms in square brackets denote the total number of interlocked  
318 components and *n* is the number of DNP units in the dumbbell.<sup>6</sup> In this co-conformation, half  
319 of the dioxynaphthalene (DNP) units (in red) are encircled by cyclobis(paraquat-*p*-phenylene)  
320 (CBPQT<sup>4+</sup>) rings (in blue). The serpentine-like folding arises from  $\pi$ -stacks between DNP  
321 units and bipyridinium charged rings, hydrogen bonds linking alpha-hydrogen atoms of the  
322 rings with poly(ethylene oxide) of the dumbbells, and Coulombic interactions between the  
323 PF<sub>6</sub><sup>-</sup> counterions and the rings. PF<sub>6</sub><sup>-</sup> counterions are not shown.

324



325

326 **Figure 2. AFM-Based mechanical unfolding of the [5]oligorotaxane in DMF. a.**

327 Illustration of the pulling experiment. Mechanical unfolding occurs by retracting the

328 cantilever at a constant velocity. The theoretical variation of length after the breaking of one

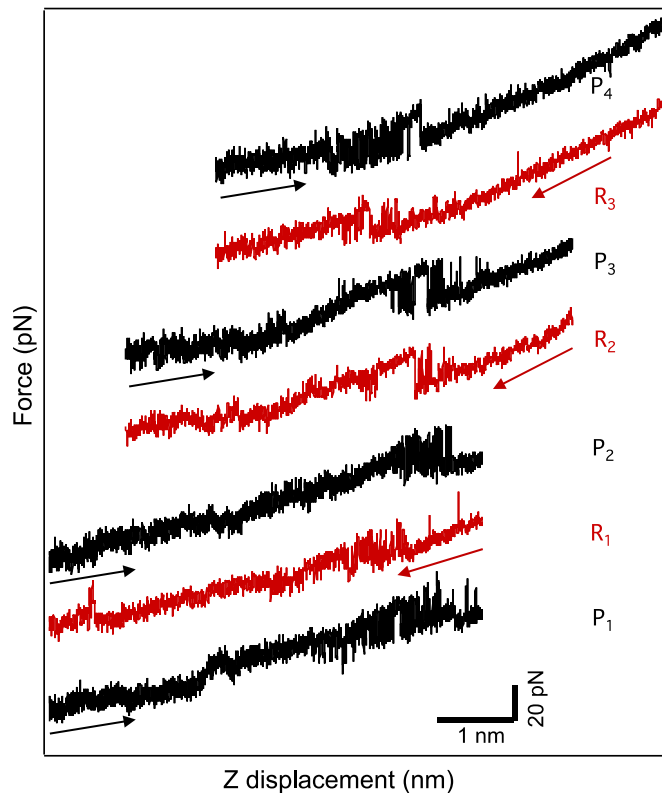
329 interaction between a free DNP and the contiguous ring is about 1.2 nm, on the basis of

330 crystallographic data (see the details in SI). **b.** Example of a force-distance curve obtained at

331  $40 \text{ nm} \cdot \text{s}^{-1}$  ( $10^3 \text{ pN} \cdot \text{s}^{-1}$ ) showing a characteristic reproducible saw-tooth profile. Each peak is

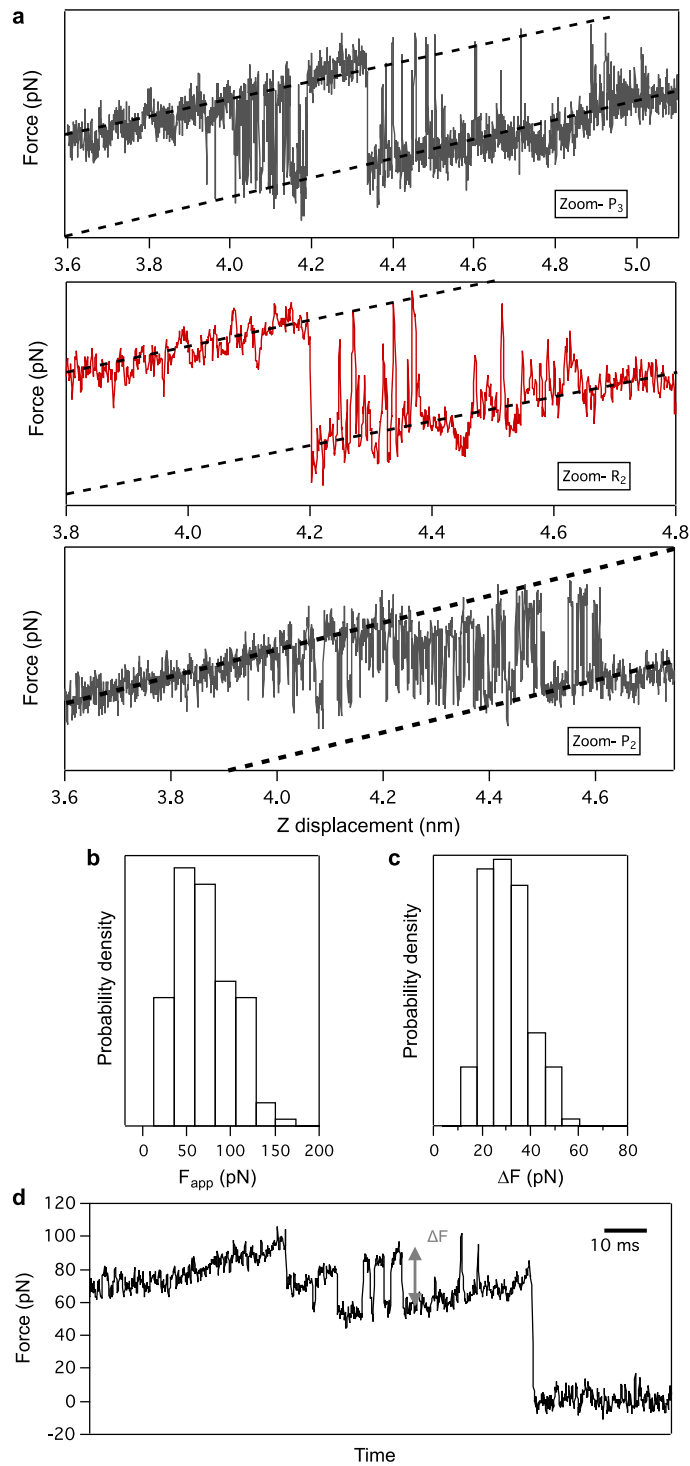
332 fitted with the WLC model to obtain contour length increments ( $\Delta L_c$ ) values. **c.** Distribution

333 of the contour lengths increments ( $\Delta L_c$ ) between successive peaks ( $N = 1712$ ). Histograms,  
334 Gaussian mixture model (GMM) multiple populations, and probability density function (PDF)  
335 are shown. **d.** Example of a force-distance curve obtained at  $350 \text{ nm}\cdot\text{s}^{-1}$  ( $10^4 \text{ pN}\cdot\text{s}^{-1}$ ) showing  
336 a characteristic reproducible saw-tooth profile with missing every second peak.  
337



338

339 **Figure 3. Pulling-relaxing cycles showing the reformation of one interaction under**  
340 **mechanical load.** Pulling(P)-relaxing(R) curves (down-top) showing sequential folding and  
341 unfolding of a single interaction in an individual [5]oligorotaxane molecule in DMF. Hopping  
342 from one co-conformation to the other is observable during pulling and relaxing. (Pulling at  
343  $10 \text{ nm}\cdot\text{s}^{-1}$  ( $300 \text{ pN}\cdot\text{s}^{-1}$ ) and relaxing at  $4 \text{ nm}\cdot\text{s}^{-1}$  ( $120 \text{ pN}\cdot\text{s}^{-1}$ )).



344

345 **Figure 4. Pulling-relaxing experiments showing the numerous fluctuations between**  
 346 **folded and unfolded states during the breaking and reformation of one interaction.**

347 **a.** Pulling-relaxing cycles obtained on an individual [5]oligorotaxane molecule in DMF. The  
 348 figures are magnifications of the force curves shown in Fig. 3 (P3, R2 and P2). **b.** Distribution  
 349 of the external force applied before the fluctuations from locally unfolded to locally folded

350 states ( $N = 115$ ). The histogram shows the maximum external force against which the  
351 oligorotaxane is able to refold. **c.** Distribution of the force increment during fluctuations  
352 ( $N = 115$ ). During the local folding, the oligorotaxane exerts a force of 25 pN on the average,  
353 up to 50 pN (loading rate of  $10^4 \text{ pN}\cdot\text{s}^{-1}$ ). **d.** Force-time curve (loading rate of  $10^4 \text{ pN}\cdot\text{s}^{-1}$   
354 ( $350 \text{ nm}\cdot\text{s}^{-1}$ )).