Inorganic Chemistry

$_{1}$ Li₁₁Nd₁₈Fe₄O_{39- δ} Revisited

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ABSTRACT: The structure proposed for $Li_{11}Nd_{18}Fe_4O_{39-\delta}$ (Chen et al. *Inorg.* 10 Chem. 2012, 51, 8073) on the basis of diffraction and Mössbauer spectral data is 11 compared to that determined previously for Nd₁₈Li₈Fe₅O₃₉ (Dutton et al. Inorg. 12 Chem. 2008 47, 11212) using the same techniques. The Mössbauer spectrum 13 reported by Chen et al. has been reinterpreted. The newly refined spectral 14 parameters differ significantly from the published values but are similar to those 15 reported for Nd₁₈Li₈Fe₅O₃₉. The relative areas of the three components indicate 16 that iron cations occupy the 2a, 8e, and 16i sites in space group $Pm\overline{3}n$, in 17 disagreement with the model determined from neutron diffraction by Chen et al. in 18 which only the 2a and 8e sites are so occupied. The relationship between 19 $Li_{11}Nd_{18}Fe_4O_{39-\delta}$ and $Nd_{18}Li_8Fe_5O_{39}$ is discussed, and it is proposed that the 2.0 sample prepared by Dutton et al. is a kinetic product whereas the sample prepared 21 by Chen et al. is the thermodynamically preferred product. 22



23 INTRODUCTION

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 $_{24}$ The crystal structure of La₁₈Li₈Rh₅O₃₉, see Figure 1, was $_{25}$ deduced from neutron powder diffraction data in 2005.¹ Chains 26 of coordination polyhedra, in which octahedral sites alternate



Figure 1. Cubic La₁₈Li₈Rh₅O₃₉ structure (space group $Pm\overline{3}n$); gray circles represent oxygen, black circles La. The LiO₆ trigonal prisms are blue (16*i* site), the RhO₆ octahedra are green (2*a* site) and red (8*e*). A 2*a* site at the center of the unit cell is hidden in this view.

with trigonal prismatic sites, were found to occupy channels 27 within a La-O framework. The chains and channels run along 28 the $\langle 111 \rangle$ directions of the cubic unit cell, and the chains 29 intersect each other at (0, 0, 0) and (1/2, 1/2, 1/2). Equivalent 30 octahedral sites are located at these two points of intersection, 31 and a further, crystallographically distinct octahedral site is 32 located halfway between them. The two distinct types of 33 octahedral sites, both occupied by rhodium, are always 34 separated from each other by a prismatic site which is occupied 35 by lithium. The Rh³⁺ and Rh⁴⁺ cations, which occur in a 4:1 36 ratio in La₁₈Li₈Rh₅O₃₉, are found respectively on the 8e and 2a 37 sites of space group $Pm\overline{3}n$; the midchain 8e site is significantly 38 larger than the 2a site at the points of intersection, and the 39 cation ordering is thus consistent with the difference in size of 40 the sites. The Li⁺ cations are located on 16*i* sites within the 41 prisms.

Subsequent studies^{2–7} have shown that the second-row 43 transition-metal element in $Ln_{18}Li_8Rh_5O_{39}$ (Ln = La, Nd, Pr) 44 can be replaced by several combinations of first-row elements 45 to form $Ln_{18}Li_8M_{5-x}M'_xO_{39}$ (M, M' = Mn, Fe, Co). 46 Unfortunately, the samples produced have always been 47 contaminated by the excess lithium carbonate used in the 48 synthesis. X-ray diffraction patterns collected on laboratory 49 diffractometers have not always revealed the presence of the 50 impurity phase, but it has been easily identified in neutron 51



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52 diffraction patterns. However, the crystal structure of the 53 weakly diamagnetic carbonate is known, and the presence of 54 this second phase has therefore not presented a problem in the 55 analysis of diffraction or magnetometry data collected from 56 these samples.

Recently⁸ Chen et al. have carried out a phase-diagram study 57 58 of the Nd₂O₃-Fe₂O₃-Li₂O ternary system in an attempt to 59 prepare a pure sample for Ln = Nd, M = M' = Fe. They report 60 the synthesis, from a stoichiometric mixture of starting 61 materials, of a monophasic sample of composition 62 $\text{Li}_{11}\text{Nd}_{18}\text{Fe}_4\text{O}_{39-\delta}$. Dutton et al. have previously⁵ reported the 63 synthesis of Nd₁₈Li₈Fe₅O₃₉ contaminated, as usual, by lithium 64 carbonate. X-ray diffraction shows that the two compounds 65 have the same unit-cell parameter. Consequently, Chen et al. 66 suggest that the composition assigned to the compound 67 prepared by Dutton et al. is incorrect. This implies that 20% 68 of the iron in the reaction mixture used by the latter is in a 69 phase other than the principal product. The fact, not 70 commented on by Chen et al., that this material was studied 71 by ⁵⁷Fe Mössbauer spectroscopy down to 4.2 K without the 72 observation of a second iron-containing phase is therefore 73 surprising, as is the fact that the magnetic susceptibility shows 74 no deviation from paramagnetic behavior above 10 K. The use 75 of a lithium-rich formulation by Chen et al. stems from the 76 analysis of their neutron diffraction data which suggested that 77 25% of the octahedral 8e sites in the structure are occupied by 78 lithium rather than iron cations. Dutton et al. found an Fe:Li 79 ratio of 85:15 on this site, but with the important difference so that the prismatic 16i site was partially occupied by iron cations so as to maintain the overall composition. This model was 81 82 consistent with both their neutron diffraction and Mössbauer spectroscopic data. The remainder of the extra lithium required 83 s4 by the composition $Li_{11}Nd_{18}Fe_4O_{39-\delta}$ is attributed by Chen et 85 al. to the partial occupation by lithium of a 24k site which was 86 unoccupied in the structural model proposed by Dutton et al. It 87 appears that this additional site lies only 1.57 Å from an oxide 88 ion, and only 1.599 Å from a lithium cation in one of the 89 polyhedral chains. The authors recognize this as an issue and 90 suggest that the two cation sites are not simultaneously 91 occupied, but the implied presence of vacancies on the 16i sites 92 and the consequences of this for the composition are not discussed further. 93

The Mössbauer spectra recorded by Chen et al. do suggest 95 the possible presence of some Fe³⁺ on the 16*i* site in their 96 sample, which would add to the uncertainty in the overall 97 composition. The full-width at half-maximum of this spectral 98 component, 0.194 mm s⁻¹, is as narrow as is permitted by the 99 Heisenberg Uncertainty Principle and is considerably narrower 100 than the other line widths reported by either Chen et al. or 101 Dutton et al. This might be considered surprising in view of the 102 structural disorder around the 16*i* site, although Chen et al. 103 comment that the inclusion of this component was necessary to 104 obtain satisfactory fits. In this paper we report our analysis of 105 the spectrum published by Chen et al.

106 **RESULTS**

107 No further experimental work has been carried out. Rather we 108 have digitized and refitted the experimental data shown in 109 Figure 5 from reference 8. Unfortunately, Chen et al. failed to 110 report⁸ the percentage transmission in their Mössbauer 111 spectrum. We have therefore assumed that the percentage 112 transmission was similar to that observed⁵ earlier, and we have 113 fixed the minimum transmission at 98%. First, we fitted the experimental data with the spectral parameters constrained to 114 be identical to the published fit parameters given in Table 4 of 115 reference 8; we refined only the baseline and the total spectral 116 absorption area. As expected, this leads to a fit that is virtually 117 identical to that shown in Figure 5 of reference 8. Second, we 118 refined the isomer shift, δ , the quadrupole splitting, ΔE_{Q_2} the 119 full width at half-maximum, Γ , and the percentage area of each 120 of the three doublets, as well as the baseline, and the total 121 absorption area, that is, a total of 14 parameters. This 122 refinement used the highly robust and efficient Levenberg– 123 Marquardt algorithm. The result of this fit is shown in the lower 124 portion of Figure 2; the fit previously reported in reference 5 125 f2



Figure 2. Room temperature Mössbauer spectra of $Nd_{18}Li_8Fe_5O_{39}$ previously published by Dutton, et al. in reference 5, top, and of $Li_{11}Nd_{18}Fe_4O_{39}$ obtained from Chen, et al. in reference 8 and fitted with three unconstrained symmetric quadrupole doublets, bottom. The vertical scale of the lower spectrum is an approximation because no scale is given in reference 8. The colors used to distinguish the three components in this figure correspond to those used for the different sites in Figure 1; the 2*a*, 8*e*, and 16*i* sites are in green, red, and blue, respectively.

for $Nd_{18}Li_8Fe_5O_{39}$ is shown for comparison in the upper 126 portion of this figure. In this refinement, rather surprisingly, we 127 obtained a lower χ^2 value than that obtained in the first step, an 128 indication that the fit shown in Figure 5 of reference 8 is not 129 the best possible fit. More significantly, all the line widths 130 refined to normal values that are similar for the three 131 components. Further, this fit avoids the significant misfitting 132 present at both -0.2 and +0.6 mm/s in Figure 5 of Chen et al.; 133 the misfitting is primarily associated with the artificially narrow 134 line width of the doublet tentatively assigned by them to the 1*6i* 135 site. Our refined parameters are given at the bottom of Table 1; 136 t1 the parameters published in references 5 and 8 are given for 137 comparison at the top and center of this table, respectively. 138

DISCUSSION

It is quite clear that the fit published by Chen et al.⁵ for 140 $\text{Li}_{11}\text{Nd}_{18}\text{Fe}_4\text{O}_{39}$ had not properly converged. This failure to 141 converge may be a consequence of the extensive correlations 142 between the refined spectral parameters, as is shown by our 143 correlation matrix. Further, the reported spectral parameters 144 were given with more significant figures than is justified by the 145 estimated, but not statistically obtained, accuracies. The refined 146 fit obtained herein for the $\text{Li}_{11}\text{Nd}_{18}\text{Fe}_4\text{O}_{39}$ spectrum yields 147 spectral parameters that are very similar to those reported in 148

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Table 1. Mössbauer Spectral Parameters for Nd₁₈Li₈Fe₅O₃₉ and Li₁₁Nd₁₈Fe₄O₃₉

compound	site	δ , ^{<i>a</i>} mm/s	$\Delta E_{\rm Q}$, mm/s	Γ, mm/s	area, %	assignment	ref.
Nd ₁₈ Li ₈ Fe ₅ O ₃₉ ^b	2 <i>a</i>	-0.229(9)	0.30(2)	0.33(2)	19.99	iron(IV)	5
	8e	0.306(1)	0.149(8)	0.332(7)	68.17	iron(III)	5
	16 <i>i</i>	0.27(1)	0.88(3)	0.33(3)	11.84	iron(III)	5
Li ₁₁ Nd ₁₈ Fe ₄ O ₃₉ ^c	2 <i>a</i>	-0.296(20)	0.334(20)	0.360(20)	21.8(2.0)	iron(IV)	8
	8e	0.294(20)	0.041(20)	0.418(20)	71.6(2.0)	iron(III)	8
	16i ^d	0.077(20)	0.948(20)	0.194(20)	6.7(2.0)	iron(III)	8
Li ₁₁ Nd ₁₈ Fe ₄ O ₃₉ ^e	2 <i>a</i>	-0.28(3)	0.35(5)	0.34(4)	20(5)	iron(IV)	е
	8e	0.282(4)	0.08(3)	0.38(2)	65(2)	iron(III)	е
	16 <i>i</i>	0.06(3)	0.92(7)	0.35(5)	15(3)	iron(III)	е

^{*a*}The isomer shift is given relative to α -iron at room temperature. ^{*b*}The relative spectral areas of the three components have been constrained to agree exactly with the iron occupancies obtained from neutron diffraction refinements reported by Dutton et al. in reference 5. ^{*c*}The proposed stoichiometry and the Mössbauer spectral parameters reported by Chen et al. in reference 8. ^{*d*}This component was assigned by Chen et al. to a partial iron(III) occupancy of the 16*i* site. ^{*e*}The best fit refinement obtained herein for the experimental Mössbauer spectral data reported by Chen et al. in Figure 5 of reference 8.

149 reference 5 for Nd₁₈Li₈Fe₅O₃₉. The major difference is the 150 spectral area assigned to the 16*i* site that is larger herein than in $_{151}$ reference 5 and accounts for the increased absorption at -0.4152 mm/s in the lower spectrum in Figure 2 as compared to the 153 upper spectrum. Consequently, the Mössbauer spectrum published by Chen et al.8 does not support their neutron 154 155 diffraction refinement, in which iron cations occupy only the 2a156 and 8e sites in a ratio of 1:3. In contrast, the Mössbauer spectrum agrees very well with the iron occupancies of the 2a, 157 8e, and 16i sites proposed in reference 5 for Nd₁₈Li₈Fe₅O₃₉, but 158 159 with a slightly larger occupancy of the 16i sites. Alternatively, 160 the third doublet in blue in the lower spectrum in Figure 2 with $_{161}$ a relative area of 15(3) % could be assigned to an undetermined 162 iron-cation containing impurity and then the green and red doublets assigned to the 2a and 8e sites have an area ratio of 163 164 1:3.3 close to the 1:3 expected from the neutron diffraction 165 refinement. In this case, the presence of three doublets in the 166 Mössbauer spectrum certainly does not support the claim of purity of the samples prepared⁸ by Chen et al. 167

There is one further aspect of the model proposed by Chen 168 169 et al. that should be considered carefully. Although they claim 170 to have prepared monophasic samples, they state that the sample used in their neutron diffraction experiment contained 171 1.9 wt % LiFeO₂. As a result of the large difference in relative 172 molecular mass between the impurity and the principal phase, 173 174 this corresponds to a molar ratio of \sim 1:1.38. The composition of the sample from which the structural model is derived 175 176 therefore differs significantly from the ideal formulation. 177 Unfortunately, the impurity was not detected in their preliminary X-ray studies. The neutron diffraction pattern 178 shown by Dutton et al. reveals no such impurity, and the 179 180 Mössbauer spectra they present do not show a component that 181 is consistent with the hyperfine parameters reported 182 previously^{9,10} for LiFeO₂ or NdFeO₃, the two most likely iron-containing impurities. We estimate the detection limit of 183 184 these spectra to be $\leq 3\%$. Furthermore, when Dutton et al. 185 attempted to prepare Nd₁₈Li₈Co₅O₃₉ they detected a 2.6 wt % (1:1 molar) $LiCoO_2$ impurity by in-house X-ray diffraction and 186 subsequently prepared a monophasic sample of Nd₁₈Li₈Co₄O₃₉. 187 This suggests that the samples of Dutton et al. were subjected 188 189 to a particularly rigorous check for impurities, and that had 20% 190 of the iron in their reaction mixture been in a second phase it is 191 likely that it would have been identified.

CONCLUSION

Although it is easy to accept that the samples prepared by the 193 two groups have different compositions, it is not obvious that 194 the composition assigned by Dutton et al. to their sample is 195 incorrect. The samples prepared by Chen et al. were heated for 196 12-24 h at 950 °C, whereas Dutton et al. synthesized their 197 samples by heating at 950 °C for only 1 h. It is thus possible 198 that the sample prepared by Dutton et al. is a kinetic rather 199 than thermodynamic product. 200

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Notes	204
The authors declare no competing financial interest.	205

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